

Data-Driven Design for Metamaterials and Multiscale Systems: A Review

*Doksoo Lee, Wei (Wayne) Chen, Liwei Wang, Yu-Chin Chan, and Wei Chen**

Metamaterials are artificial materials designed to exhibit effective material parameters that go beyond those found in nature. Composed of unit cells with rich designability that are assembled into multiscale systems, they hold great promise for realizing next-generation devices with exceptional, often exotic, functionalities. However, the vast design space and intricate structure–property relationships pose significant challenges in their design. A compelling paradigm that could bring the full potential of metamaterials to fruition is emerging: data-driven design. This review provides a holistic overview of this rapidly evolving field, emphasizing the general methodology instead of specific domains and deployment contexts. Existing research is organized into data-driven modules, encompassing data acquisition, machine learning-based unit cell design, and data-driven multiscale optimization. The approaches are further categorized within each module based on shared principles, analyze and compare strengths and applicability, explore connections between different modules, and identify open research questions and opportunities.

1. Introduction

Engineered material structures generally benefit from some extreme or spatially varying material properties to achieve higher performance or complex functionalities. $[1-4]$ Typically, tuning

D. Lee, L. Wang, W. Chen Dept. of Mechanical Engineering Northwestern University Evanston, IL 60208, USA E-mail: [liwei.wang@northwestern.edu;](mailto:liwei.wang@northwestern.edu)weichen@northwestern.edu W. (Wayne) Chen J. Mike Walker '66 Department of Mechanical Engineering Texas A&M University College Station, TX 77840, USA Y.-C. Chan Siemens Corporation Technology Princeton, NJ 08540, USA

The ORCID identification number(s) for the author(s) of this article $\left(\mathbf{D}\right)$ can be found under <https://doi.org/10.1002/adma.202305254> © 2023 The Authors. Advanced Materials published by Wiley-VCH

GmbH. This is an open access article under the terms of the [Creative](http://creativecommons.org/licenses/by-nc-nd/4.0/) [Commons Attribution-NonCommercial-NoDerivs](http://creativecommons.org/licenses/by-nc-nd/4.0/) License, which permits use and distribution in any medium, provided the original work is properly cited, the use is non-commercial and no modifications or adaptations are made.

DOI: 10.1002/adma.202305254

material properties involves precise control of material composition and processing conditions, which can be both technically challenging and cost-prohibitive, particularly when aiming for spatially varying properties. In contrast, metamaterials are engineered architectural materials that can reach a broad range of properties by carefully designing their architectures or microstructures rather than altering the material composition itself.^[5-8] Along with the recent enhancements in manufacturing capabilities, metamaterials are emerging as a new paradigmatic material system to enable unprecedented design flexibility in properties. They can advance applications in a wide range of fields, including optics, $[9-11]$ electromagnetics, $[12,13]$ thermology, $[14-17]$ $acoustics,$ ^{[\[18–22\]](#page-38-0)} and mechanics.^{[\[5,23–27\]](#page-38-0)}

Nevertheless, the design of metamaterials and their multiscale structures proves to be a complex process that involves

navigating an infinite-dimensional topological design space, mapping microstructures to their effective properties across multiple scales, dealing with numerous local optima in design optimization, the absence of analytical gradient information, and expensive property or performance evaluation. Most existing metamaterial designs adopt trial-and-error and heuristic methods, $[8,24,27-31]$ which rely heavily on the experience of a designer, or simple parameter optimization methods, [\[32,33\]](#page-39-0) confining the metamaterials to a restricted selection of properties. In some specific cases with relatively simple and differentiable physical models, gradient-based topology optimization (TO) has been utilized to facilitate the automatic design of metamaterials. Nonetheless, these methods are generally not scalable to the design of multiscale systems of metamaterials that features highdimensional design space, nested optimization loops, numerous property/performance evaluations at different scales, and location-specific microscale designs.

The emergence of data-driven methods has provided solutions to these challenges by enabling high-throughput property or response prediction, reducing the dimensionality of complex problems, accelerating design space exploration and design optimization, and allowing fast solutions to ill-posed inverse design problems. Data-driven metamaterials design typically includes three modules, 1) data acquisition: acquiring a precomputed dataset of unit cells; 2) machine learning (ML) based unit cell design: using ML to extract information from data and help unit cell designs; 3)

CIENCE NEWS www.advancedsciencenews.com www.advmat.de

IDVANCEI

multiscale design: utilizing unit-cell database and ML models for design synthesis at the system level. In practical applications, it is possible to integrate all of these modules into a unified framework or selectively focus on specific modules based on the design requirements. However, the core idea underlying data-driven design remains consistent across these modules, which is to extract meaningful patterns from data that are unavailable or difficult to obtain with physical models, and incorporate them into the design process to simplify the complexity in traditional design approaches. Although these capabilities come with the cost of data collection and model training, deploying the trained model has the benefit of negligible inference time, which can significantly speed up the design process. Data-driven design methods are especially useful in scenarios where the design problems are highdimensional or the governing physics is unknown. They can also achieve unprecedented design performance owing to their capability of encapsulating higher design freedom (e.g., heterogeneous metamaterials system design) compared to conventional design methods.

Motivated by the transformative shift of the design paradigm, the last several years have seen the emergence and growth of datadriven design reported from a variety of communities. To our best knowledge, the first implementations of data-driven methods in the design of multiscale structures date back to 2015, when Panetta et al.^{[\[34\]](#page-39-0)} and Schumacher et al.^{[\[35\]](#page-39-0)} proposed to optimize multiscale structures with spatially varying material properties by conceptualizing and harnessing metamaterial databases. Starting in 2018, researchers began to incorporate machine learning into the design for mechanical and optical metamaterials.^{[\[36–42\]](#page-39-0)} After 2020, machine learning methods expanded to additional metamaterials domains including acoustic, elastic, thermal, and magneto-mechanical.[\[43–45\]](#page-39-0)

As evidence of growing attention to data-driven multiscale design, with or without the use of data, a suite of literature reviews from different communities have been published in recent years. Each review is centered on particular topics, such as design, $[46-50]$ manufacturing, $[51,52]$ mechanics, $[50,53-59]$ and specific ML methods.[\[47\]](#page-39-0) Despite the meaningful contributions of each, and of the aggregate, we recognize a lack of discussion on some points that may impede researchers from unlocking the full potential of data-driven design for multiscale architectures. First, in the corpus, we observe a disconnect between two primary lines of approaches, one being the data-driven camp that harnesses pre-existing ML tools with minimal customization, and the other being the physics-based camp that specializes in physics with limited awareness of recent advancements of data-driven methods. Second, when covering datadriven design, the prior reviews are typically dedicated to spe-cific aspects, e.g., ML methods,^{[\[47\]](#page-39-0)} physical mechanisms,^[53-55,59] individual components of data-driven design.^{[\[46,48\]](#page-39-0)} In the current state of the field, it is difficult to find a singular review that provides a holistic overview of data-driven design, a summary of the archetypal design framework, and the critical interdependencies between design components. Third, while some existing reviews^{[\[47,52,59\]](#page-39-0)} discussed data preparation as a core module, i.e., a step in data-driven design, none have systematically compared how existing datasets were created and how data quality could be ensured to better support the design of multiscale architectures.

The key contributions of our review include the following:

- 1. We adopt a design-centered perspective to examine a wide range of papers, categorizing existing methods from various domains into three modules within the data-driven design framework. This synthesis of different studies allows us to present a clear and cohesive picture of "how" data-driven design has been practiced from data acquisition to single-scale and multiscale optimization. Our key focus is on the methodological aspects of design without specific deployment goals, i.e., without filtering based on the underlying physical mechanisms, geometric families of unit cells, and fabrication methods. This holistic approach allows us to uncover the common threads and overarching principles that have driven the field of data-driven design.
- 2. We review the common practices of current data generation strategies for metamaterial design through a standardized taxonomy, discuss key concerns in depth, and attempt to raise awareness on certain issues that are crucial yet underestimated, or even overlooked, in data acquisition.
- 3. We review prior data-driven design methods that can be broadly applied to metamterial design under different physics (i.e., optical, acoustic, mechanical, thermal, etc.) and raise critical concerns that are generalizable to all types of metamaterial design problems addressed by data-driven methods.
- We investigate the role of data-driven models and methods in the context of multiscale system design. Unlike previous studies that treat data-driven models as isolated solutions, we highlight their integration into a complete design workflow and their scalability in handling large databases, multiple scales, and combinatorial design spaces. By examining these tools within the broader design process, we aim to offer insights into how data-driven approaches can be effectively utilized to enhance the efficiency and effectiveness of metamaterial design.

We specify our scope as follows:

- 1. We limit our scope to only the ML-based design methods for metamaterials and their multiscale systems.
- 2. We consider a wide range of physics including optical, acoustic, mechanical, thermal, and magneto-mechanical, because the ML methods are usually applicable regardless of the physics that governs the problem.
- 3. The reviewed ML methods do not necessarily require prior training data (i.e., we included past works using methods such as reinforcement learning (RL) or physics-informed neural networks).
- 4. We encompass design for multiscale systems that are built on either a unit-cell database of metamaterials or ML models. It includes the use of descriptors of unit cells, surrogate models for constitutive laws, efficient simulation via ML, optimization, and assembly of unit cells based on the database.

This paper is structured as follows: Section [2](#page-2-0) gives a brief definition of key concepts underpinning our review and major ML methods utilized in the works we will cover. In Section [3,](#page-4-0) isolating data acquisition from the pipeline of data-driven multiscale design, we anatomize the common practices of data acquisition

Figure 1. Hierarchical terminology system in this review. A metamaterial are created by assembling multiple microstructures to achieve properties on the macroscale. A microstructure is an assembly of multiple unit cells. A unit cell is the smallest representative unit of a material in the microscale to control its properties.

strategies, with particular attention to the methodological procedures of shape generation and property-aware sampling. Following this, we provide a brief review of the current practices of data assessment that aim to ensure data quality and often shed light on how it can propagate into downstream tasks. Section [4](#page-16-0) reviews prior works using ML methods for single-scale metamaterials design, discusses some critical considerations when evaluating these methods, and proposes promising future directions. Section [5](#page-26-0) explores data-driven design methods for multiscale systems that are built on either a unit-cell database or ML models. It includes the use of descriptors of unit cells, surrogate models for constitutive laws, efficient simulation via ML, optimization, and assembly of unit cells based on the database.

2. Preliminaries

This section defines key terminologies and concepts used throughout this paper as well as in other data-driven metamaterials design literature. We also briefly introduce common ML techniques and how these techniques were applied in the context of metamaterials design.

2.1. Key Concepts

Below, we list working definitions of key concepts to be frequently used throughout the paper.

Unit Cell: The smallest representative unit of a materialto control its properties, as shown in **Figure 1**. It is often referred to by interchangeable terms such as meta-atom, meta-molecule, building block, and cell,^{[\[7\]](#page-38-0)} depending on the physical mechanism, scale, and geometry. Note that, strictly speaking, a unit cell originally refers to a repeated structure within periodic materials.[\[60\]](#page-39-0) However, in the context of this review, we expand this concept to include representative units within both periodic and aperiodic metamaterials, and representative volume element (RVE) that statistically characterize the collective properties of stochastic metamaterials.

Microstructure: An assembly of multiple unit cells arranged in a specific pattern to achieve more complex properties arising from their arrangement.

Metamaterial: An assembly of multiple microstructures, often in a periodic pattern, to achieve unique and tailored properties that cannot be found in natural materials.

Homogenization: A process to bypass or average the details in the microscale, either deterministic microstructures in periodic structures or stochastic microstructures with random distribution, to obtain effective material parameters that could characterize the overall structural or material response in the macroscale.[\[59\]](#page-39-0)

Multiscale Design: The process of designing metamaterials with desired properties at multiple length scales, from the microscopic level of a single unit cell to the macroscopic level of the metamaterial structure.

Module: An independent and reusable unit or component within the data-driven multiscale design. It provides a specific functionality that is required for design. An ordered sequence of modules forms a data-driven multiscale design framework. Modules of primary interest in this review include data acquisition (Section [3\)](#page-4-0), ML-based unit cell design (Section [4\)](#page-16-0), and data-driven mulitscale optimization (Section [5\)](#page-26-0).

Effective Properties: The macroscopic property of a metamaterial that arises from the collective behavior of its microstructures.[\[60\]](#page-39-0) These properties are often different from the intrinsic properties of the constituent materials, and can be tailored through design and optimization of the microstructure.

Class: A group of unit cells that can be generated from the same geometric motif or design parameterization.^{[\[61\]](#page-39-0)}

Representation: A set of parameters, or models, used to di-rectly characterize unit cells.^{[\[61\]](#page-39-0)} Representations often involve the projection of high-dimensional instances into a lowerdimensional space.

Design Space: The space of all possible design solutions. It contains all the combinations of design variables. In the context of metamaterials design, design variables usually refer to geometric or material design variables.

Shape Space: The geometric design space of unit cells.

Property Space: The response space of unit cells.

Evaluation: To obtain system responses of concern given an architecture and loading conditions, typically through numerical analyses such as finite element methods (FEM). In ML literature, the evaluation process is similar to "labeling", which refers to the

IDVANO **CIENCE NEWS www.advancedsciencenews.com www.advmat.de**

process of adding attributes of interest (i.e., labels) to raw data. Throughout this review, we will assume that the term evaluation is interchangeable with labeling.

Shape–Property Mapping: A directional mapping from instances in shape space to property space. It is typically learned through ML using labeled data.

Compatibility: Capability of neighboring unit cells to possess seamless geometric/mechanical connections, or a lack of geometric frustration. It is often measured through geometric/mechanical similarities at the interface of neighboring unit cells.[\[62,63\]](#page-39-0)

2.2. Machine Learning

Data-driven metamaterial design processes usually include the use of ML models to extract useful information from data. The extracted information can then help the design process in different ways. Machine learning approaches used in this context were mainly from four categories—supervised learning, unsupervised learning, semi-supervised learning, and RL. We briefly introduce these categories in this section.

2.2.1. Supervised Learning

In a supervised learning task, we train an ML model to predict attributes of interest (i.e., labels). The ML model is trained on a collection of data-label pairs. The data can take various forms, e.g., vectors, images, and graphs. Depending on the type of labels, supervised learning can be divided into a broad dichotomy: regression with continuous outputs and classification with discrete ones. Commonly used ML models include neural networks, kernel machines, and decision trees. Selecting the type of model is generally done before training, and could be contingent upon the end task (e.g., regression/classification), data volume, need to predict uncertainty, and design applications.

Within the context of data-driven metamaterials design (DMD), supervised learning is most widely-used for creating shape-property relations. A data pair typically describes a shape, represented by its parameterization (e.g., parameterized lattice), and its quantities of interest (e.g., elasticity components or response spectra). A key motivation for using supervised learning has been to replace the resource-intensive evaluation of unit cells with a faster surrogate model. The types of models have been based dominantly on neural networks^{[\[62,64,65\]](#page-39-0)} and sometimes on Gaussian processes (GPs).^{[\[62\]](#page-39-0)} Once trained on a large volume of data, a data-driven model offers on-the-fly predictions of the (effective) properties of unseen unit cells. Sometimes such a surrogate is incorporated into a larger network architecture and trained end-to-end together with other components.^{[\[63,66\]](#page-39-0)}

2.2.2. Unsupervised Learning

In contrast to supervised learning, unsupervised learning extracts information from unlabeled data. More specifically, it addresses tasks such as clustering, anomaly detection, and dimensionality reduction. In metamaterials design, it is mainly applied to learning the representation or the distribution of complex metamaterial geometries. Unsupervised learning models

commonly used in metamaterials design are autoencoders, variational autoencoders (VAE), and generative adversarial networks (GAN).

Autoencoder^{[\[67\]](#page-39-0)} is a type of neural network that uses an encoder-decoder architecture to extract lower-dimensional latent variables of input data. In metamaterials design, autoencoders were used to reduce the dimensionality of either metamate-rial geometries^{[\[43\]](#page-39-0)} or high-dimensional responses such as scattering parameters.[\[68\]](#page-39-0) In 2013, Kingma and Welling proposed the VAE.^{[\[69\]](#page-39-0)} which has a similar architecture with autoencoders, while being a type of deep generative model (DGM) that learns the distribution of data. New data can be generated by sampling latent variables that are low-dimensional and follow a welldefined distribution. Therefore, the latent representation learned by a VAE is usually more efficient and interpretable than the original metamaterial design representation, especially when consid-ering high design complexity and freedom.^{[\[66,70\]](#page-39-0)} In 2014, Goodfellow et al. proposed the GAN.[\[71,72\]](#page-39-0) Same as VAEs, GANs can also generate new metamaterial designs and learn efficient representations. A GAN models the generation as a game between its generator and discriminator. Compared to VAEs, GANs can generate higher-quality samples.[\[73\]](#page-39-0)

While VAEs and GANs have been generally used for unsupervised learning, prior works have proposed variants of these models, such as conditional generative adversarial networks (cGAN) (e.g., refs. [\[63, 64, 74\]](#page-39-0), conditional variational autoencoders (cVAE) (e.g., refs. [\[75, 76\]](#page-39-0)), and VAE-regressor models (e.g., ref. [\[66\]](#page-39-0)), that require supervised learning. These supervised learning models either enable the inverse design of metamaterials[\[63,64,74–76\]](#page-39-0) or help construct a property-related metamaterial latent representation.[\[66\]](#page-39-0)

2.2.3. Semi-Supervised Learning

Semi-supervised learning trains an ML model on partially labeled data so that the model can predict the labels of any given data. Typically it is assumed that the portion of labeled data is much smaller than the counterpart. This is a commonplace scenario in ML due to the high labeling cost. Marrying supervised and unsupervised learning, semi-supervised learning aims to improve the performances of either. It also inherits most of the ML tasks above, such as semi-supervised classification and regression, and semi-supervised generative modeling.

Within DMD, the efficacy of semi-supervised learning has been demonstrated in some works. In the design of photonic metasurfaces, Ma et al. used both labeled and unlabeled data when training the prediction network to improve the regression with a less computational overhead of data preparation.^{[\[75\]](#page-39-0)} Although the idea of semi-supervised learning could sound compelling, the efficacy comes under some conditions; in-depth discussion on this point can be found in some reviews dedicated to semi-supervised learning.^{[\[77,78\]](#page-39-0)}

2.2.4. Reinforcement Learning

Reinforcement learning (RL) is another category of ML methods used in metamaterials design. It is modeled as a Markov decision

IDVANCED **SCIENCE NEWS www.advancedsciencenews.com www.advmat.de**

process,[\[79\]](#page-39-0) where an agent takes actions in an environment to maximize a reward. The goal is to learn an optimal policy that guides the action-taking strategy. Different from supervised, unsupervised, and semi-supervised learning, RL does not require an initial dataset to learn from. Instead, the agent in RL learns from its experience of exploring the action space and receiving rewards. There have been successful applications of RL in areas such as gaming and robotics. Reinforcement learning is usually applied to solving sequential decision-making processes. But with a proper definition of the action space and sequential decision-making setting, the design of metamaterials can also be formulated as RL problems, as shown in prior works.^{[\[44,80,81\]](#page-39-0)}

3. Data Acquisition

3.1. Overview

Creating and leveraging datasets of unit cells has been a core enabler of the recent success of DMD. Data acquisition in multiscale design is a decision-making activity that determines a finite collection of structure-property pairs, which delegates the space to be explored in downstream tasks. When one strives to exploit the power of DMD, data acquisition presents multifaceted open challenges, such as exploration of high-dimensional design space of unit cells, resource-intensive design evaluation in particular for large datasets, many-to-one mapping from shape to property, distributional bias in data, and opaque compounding effects of data quality to downstream tasks.

Generally taken as the first module of DMD, data acquisition was tackled through diverse strategies in past works. To our best knowledge, the most foundational, formal demonstration of precomputed unit cell library for multiscale design was reported in 2015.[\[34,35\]](#page-39-0) Commonly pursuing digital design for 3D printable elastic architectures, both conceptualized the notion of microstructure families and exploited their interpolation to secure decent coverage and distributional uniformity in the property space of isotropic elastic constants. While sharing a simi-lar end-goal, Panetta et al.^{[\[34\]](#page-39-0)} advocated for truss-like symmetric structures endowed with explicit parameterization, whereas Schumacher et al.^{[\[35\]](#page-39-0)} harnessed freeform microstructures, generated by TO. The work was followed by Zhu et al., $[82]$ where a more advanced TO-based approach was proposed to iteratively expand the property space of voxelated, freeform unit cells. Foundational works involving data acquisition in plasmonic metasurfaces include Malkiel et al.,[\[83\]](#page-39-0) where unit cells were synthesized based on geometric primitives and operations, both of which were explicitly parameterized. Pursuing a large-scale shape dataset with more geometric variability, Liu et al.^{[\[42\]](#page-39-0)} conceived pixelated unit cells that were crafted from a group of selected canonical classes, e.g., cross, bow-tie, V-shape, split ring resonator, using a variety of operations, e.g., scaling, rotation, union. To support building blocks with wide topological diversity, Whiting et al.^{[\[84\]](#page-39-0)} proposed a representation based on high-dimensional embedding using control points. Meanwhile, recent years have also seen efforts toward method development dedicated to data acquisition of metamaterial unit cells beyond the widely used classic space-filling sampling.^{[\[85,86\]](#page-39-0)}

Despite the diversity of strategies seen across different communities, it is difficult to draw connections among them due to

the absence of attempts to build a common context that bridges different strategies. To this end, we present a review of data acquisition for DMD with a particular focus on the methodological perspectives of prior works. We propose a standardized taxonomy to organize the literature in a relatable and easy-to-compare manner. Based on our observations of the current research trend, our review of data acquisition consists of two parts: Shape-Centric Data Generation Method (Section 3.2) and Property-Aware Data Acquisition Strategy (Section [3.3\)](#page-9-0). In the corpus, we recognize that many demonstrations of data acquisition adopted shape generation heuristics, which do not necessarily consider property, in order to incorporate domain knowledge into datasets and avoid handcrafting a large number of shape instances one by one. Section 3.2 offers a detailed review of these shape-centric data acquisition methods. In contrast, Section [3.3](#page-9-0) reviews acquisition strategies that take property into account. These methods could boost sampling efficiency and facilitate data customization for specific design tasks.

Following the current research trend, the scope of this section is centered on data acquisition, but we also briefly introduce exemplar demonstrations of data assessment in Section [3.4](#page-11-0) that is key to underpinning data quality assurance and data sharing practice. Each subsection includes a discussion specific to its topic, while Section [3.5](#page-15-0) offers a more general and comprehensive discussion that covers multiple themes in data acquisition.

3.2. Shape-Centric Data Generation Method

Data acquisition for DMD often entails a large collection of unit cell shapes. Handcrafting individual shape instances one by one is intractable for large data, as is running inverse optimization to obtain all unit cells corresponding to a massive set of prespecified target properties. To create large data in an effective, systematic manner, past works presented a diverse array of methodological procedures. We recognize that individual approaches commonly involve two research questions in shape generation: 1) How to specify a group of unit cells? and 2) How to grow sparse data to large data? Answering the first essentially entails a representation of unit cells (Section 3.2.1), which was usually preselected at the early stages of data acquisition and often justified based on criteria such as domain knowledge and fabrication methods. On the other hand, answering the second question involves the reproduction of unit cells (Section [3.2.2\)](#page-7-0), which facilitates the collection of a large enough dataset from sparse data without both handcrafting and optimizing a large volume of individual samples. We remark that in literature the two questions were addressed primarily based on generation strategies driven by shape rather than property; we call these types of approaches shape-centric generation methods. Given this taxonomy, we can bridge diverse data generation approaches scattered over different communities in order to offer a comparative review.

3.2.1. Representation of Unit Cells

A representation refers to a set of parameters, or models, used to directly characterize unit cells, typically by projecting

Figure 2. Representations of building blocks. a) and b): Parametric Multiclass. a) The mixed-variable multiclass lattice representation of mechanical metamaterials. Reproduced with permission.[\[87\]](#page-39-0) Copyright 2021, ASME. b) The parametric representation of 3D multiclass building blocks of mechanical metamaterials. Reproduced with permission.[\[88\]](#page-39-0) Copyright 2022, American Association for the Advancement of Science. c,d): Implicit Function. c) The representation based on Triply Minimal Periodic Surfaces of mechanical metamaterials. Reproduced with permission.^{[\[89\]](#page-39-0)} Copyright 2019, ASME. d) The spinodoid representation of mechanical metamaterials. Reproduced with permission.^{[\[90\]](#page-39-0)} Copyright 2020, Creative Commons CC BY. e,f): Pixel/Voxel. e) 3D voxelated representation of mechanical metamaterials. Reproduced with permission.[\[82\]](#page-39-0) Copyright 2017, Association for Computing Machinery. f) 2D pixelated representation. Reproduced with permission.^{[\[66\]](#page-39-0)} Copyright 2020, Elsevier B.V. g,h) Parametric Curve/Surface. g) The parametric surface repre-sentation of photonic metasurfaces. Reproduced with permission.^{[\[84\]](#page-39-0)} Copyright 2020, Optica Publishing Group. h) The parametric curve representation of photonic metasurfaces. Reproduced with permission.[\[40\]](#page-39-0) Copyright 2018, AIP Publishing. i,j): Constructive Solid Geometry. i) The representation based on primitive superposition and four-fold symmetry of dielectric metasurfaces. Reproduced with permission.[\[91\]](#page-39-0) Copyright 2021, Wiley-VCH GmbH. j) The union-based representation of plasmonic metasurfaces. Reproduced with permission under the terms of the Creative Commons Attribution 4.0 license.^{[\[83\]](#page-39-0)} Copyright 2018, the Authors. Springer Nature.

high-dimensional instances into a lower-dimensional space (Section [2.1\)](#page-2-0). Determining what representation to use, specifically for the unit cells in this subsection, is a key decision that should be made at the early stages of data acquisition, as it dictates the nature of resulting data distributions.

Figure 2 depicts widely used representations reported to date, organized based on our literature survey. The list might not be exhaustive, but we believe that it provides a sufficient overview of common practices in data generation. For each category, we discuss the definition, hallmarks, and relevant works. A comparative, multifaceted discussion across all the representations covered herein can be found in Section [3.2.3.](#page-8-0)

Parametric Multiclass: An intuitive way to encode domain knowledge into a dataset is to directly include a set of geometric classes (Section [2.1\)](#page-2-0) intensely studied in the literature. Herein, the seed classes are expected to serve as "pivots" of the shape generation. Each class is typically endowed with some lowdimensional explicit parameterization, e.g., the length/thickness **www.advancedsciencenews.com www.advmat.de** of a bar entity,[\[41,80,87,92,93\]](#page-39-0) volume fraction,[\[87\]](#page-39-0) angle between ge-

SCIENCE NEWS

ometric entities,^{[\[94\]](#page-39-0)} or rotational angle,^{[\[88,93\]](#page-39-0)} which supports di-rect design exploration either within a class^{[\[87,88,94\]](#page-39-0)} or across classes.[\[93\]](#page-39-0)

In literature, some advocated for a mixed-variable representation where qualitative (e.g., building block type) and quantitative (e.g., scaling factor) variables were used together as an explicit multiclass representation for TO (Figure [2b\)](#page-5-0).^{[\[95\]](#page-39-0)} Conceptual generalization was proposed by Liu et al., showcased by lattice-like building blocks, spinodal pattern-like ones, and multimaterial composites (Figure [2a\)](#page-5-0), each of which involved an explicit, mixed-variable representation with ensured compatibility.^{[\[88\]](#page-39-0)} Meanwhile, others conceived a versatile parametric representation able to generate multiple classes without explicitly defining classes.[\[96\]](#page-39-0) To avoid geometric frustration when assembling building blocks in the multiscale design of mechanical metamaterials, the compatibility among neighboring unit cells based on geometric and/or mechanical factors often served as a primary criterion for choosing classes.[\[87,88,96,97\]](#page-39-0)

Implicit Function: It has been widely adopted to represent unit cells using implicit functions that can generate geometric families. Therein, a shape instance is implicitly represented through a surface function.

Demonstrations in literature have been centered on functions that enjoy smooth topological variations, as opposed to lattice representations, and which are tunable by a small number of parameters. The most widely used families in DMD are Triply Periodic Minimal Surfaces (TPMS) that feature zero mean curvature and large surface areas (Figure [2c\)](#page-5-0).^{[\[63,89\]](#page-39-0)} Another isosurface representation based on linear combinations of analytical crystallographic symmetry functions was implemented by Chan et al.^{[\[85\]](#page-39-0)} Boddapati et al. proposed another representation that can synthesize diverse classes of quasi-free unit cells of mechanical metamaterials by a linear superposition of periodic cosine functions.[\[98\]](#page-39-0) Inspired by the phase separation process described by the Cahn-Hilliard equation, Kumar et al.^{[\[90\]](#page-39-0)} reported a spinodoid representation (Figure [2d\)](#page-5-0), which features smooth, aperiodic variations of complex topologies and tunable anisotropy.

A variant under this branch harnesses spectral decomposition. A manifestation of this for photonic metasurfaces was shown by Liu et al.,^{[\[99\]](#page-39-0)} where Fourier transform and level-set function of shapes served as key pillars for the new representation. The spectral representation enjoys topologically rich unit cells, reconstruction capability supported by inverse Fourier Transform, and efficient symmetry handling. Another example in this line was used by Wang et al.,^{[\[62\]](#page-39-0)} where the Laplace-Beltrami operation was employed for dimension reduction of the freeform unit cells.

Pixel/Voxel: Pixel/voxel representation builds on the assumption that any shape instance can be viewed as a spatial aggregate of solid/void elements. They are typically freeform. Distinct from other representations, this approach offers a direct connection with inverse TO.

As an early demonstration in DMD, Zhu et al. employed the voxelated representation with TO (Figure $2e$).^{[\[82\]](#page-39-0)} Wang et al. implemented inverse TO to find hundreds of freeform unit cells to start with, each of which closely matches the target effective property specified a priori (Figure [2f\)](#page-5-0).[\[62,66\]](#page-39-0) Li et al. used multimaterial TO to systemically construct a library of freeform unit cells, each of whose response was programmed to exhibit a prescribed target force-displacement behavior.^{[\[100\]](#page-39-0)} TO was also used for a thermal emitter design that aims for frequency selective re-flectivity when generating seed instances.^{[\[101\]](#page-40-0)} Harnessing interpretable ML for band gap engineering, Chen et al. adopted the 2D pixel representation to generate unit cell templates of phononic metasurfaces.^{[\[102\]](#page-40-0)}

In the literature, we also observe another subcategory that advocated the pixel-/voxelated representation while considering user-defined classes. For the design of photonic metasurfaces, many built datasets spanning from a group of canonical classes, or meta-atoms, such as a cross, bow-tie, V-shape, I-beam, split ring resonator, and others.[\[64,75,92,103,104\]](#page-39-0) This allows one to encode the data generation procedure with domain knowledge in contrast to the optimization-based pixel representation introduced above.

Parametric Curve/Surface: Boundary-based representations, also referred to as contour-based shape descriptors, [\[105\]](#page-40-0) have been commonly used to describe shapes as well. In these approaches, a shape is represented by an ordered sequence of control points on curves or surfaces.

Within DMD, this approach has been primarily favored in the design of wave-based metamaterials that pursue design exploration beyond canonical families. For metagrating design, Inampudi et al. employed a boundary representation that specified shape instances with 16 boundary control points (Figure $2h$).^{[\[40\]](#page-39-0)} Li et al. used the four-order formulation of trigonometric functions with tunable parameters to explicitly represent a boundary curve of scattering inclusions of phononic crystals.[\[43\]](#page-39-0) Tanriover et al. also harnessed such a representation to construct a shape dataset not restricted to the canonical meta-atoms in the liter-ature under curvature constraints.^{[\[106\]](#page-40-0)} As an extension, it was also shown that a higher dimensional embedding of parametric curves/surfaces can be used as an implicit representation of 2D boundaries. For example, in photonic metasurfaces, Whiting et al. conceived a representation that offers topologically diverse instances in order to generate quasi-free building blocks (Figure $2g$).^{[\[84\]](#page-39-0)} A key distinction between this and the Implicit Function method above is that the 3D embedding here is fully governed by control points, which are explicitly defined. As an example in mechanical metamaterials, Wang et al. adopted the Cassini oval curve to represent the proposed auxetic planar meta-surfaces with oval holes.^{[\[107\]](#page-40-0)}

Constructive Solid Geometry: Constructive Solid Geometry (CSG) is a geometric modeling method to create a solid ob-ject in a syntactic manner.^{[\[108\]](#page-40-0)} Its underlying concept is to compose an instance by following a sequence of set-theoretic operations (e.g., union; intersection) acting on primitives (e.g., rectangle, cylinder, sphere). The semantic nature makes instances of CSG highly interpretable, and offers a seamless connection with Computer-Aided Design. Capitalizing on explicit parameterization, a similar approach, the so-called moving morphable components,[\[109,110\]](#page-40-0) has been developed in the TO community.

Within the context of DMD, CSG has been utilized in some works, in particular for the design of photonic metasurfaces. Malkiel et al.^{[\[83\]](#page-39-0)} applied the primitive rectangle, whose presence, length, and angle were parameterized, along with the union op-eration to synthesize plasmonic nanostructures (Figure [2j\)](#page-5-0). A recent work that builds on further design freedom was reported by An et al.,^{[\[91,92\]](#page-39-0)} where a heuristic shape composition strategy,

Figure 3. Reproduction strategies applied to sparse data to generate large data. White arrows indicate the direction from existing shapes to new ones. a) Parametric Sweep applied to Wang et al., where six lattice classes were explored by varying their volume fraction. Reproduced with permission.[87] Copyright 2021, ASME. b) Multiclass Blending employed in Chan et al. where the unit cells at the leftmost and rightmost columns serve as seed classes. The proposed class remixing generates inter-class instances with ensured connectivity. Reproduced with permission.[65] Copyright 2022, Springer-Verlag GmbH Germany. c) Perturbation implemented in Wang et al., where radial distortion was harnessed as the perturbation method to recursively expand the coverage in property space while preserving the topology as well as axial symmetry. Reproduced with permission.^[86] Copyright 2023,ASME.

named as the needle-drop approach by the authors, was employed to produce a large volume of quasi-freeform unit cells as unions of rectangle primitives (Figure [2i\)](#page-5-0).

3.2.2. Reproduction from Sparse Data to Large Data

Once a representation is determined, a typical next step that follows is to use it to grow a large-scale shape collection, with an optional target dataset size. We will call this task reproduction throughout this review. A reproduction strategy dictates the way of producing generic instances in a shape set and the distributional nature of resulting data, thus significantly affecting the quality of downstream tasks of DMD. Through effective reproduction, DMD can enjoy a quality dataset that represents the property distribution with space-filling samples and wide coverage. Based on our survey, we observe three primary lines of reproduction methods: i) Parametric Sweep, ii) Multiclass Blending, and iii) Perturbation. **Figure 3** illustrates each with an example in literature.

Parametric Sweep: Given a representation of unit cells that spans a descriptor space with low/moderate dimensionality, Parametric Sweep is often applied to explore the descriptor space as uniformly as possible with a finite number of samples. Spacefilling sampling that is effective in low-dimensional spaces has been intensely studied for a long time. Readers interested in the topic are referred to reviews.^{[\[111,112\]](#page-40-0)}

We observe that, perhaps not surprisingly, Parametric Sweep is the most widely-used reproduction strategy.^[40,63,80,87,88,90,92-94,113] It has been combined with diverse types of unit cell representations, especially low-dimensional ones such as the mixedvariable lattice representation (Figure $3a$),^{[\[95\]](#page-39-0)} the six-bar lattice representation,[\[96\]](#page-39-0) the H-shape meta-atom parameterized with six variables,^{[\[92\]](#page-39-0)} the CSG-based representation,^{[\[91\]](#page-39-0)} and the pixel/voxel dataset with canonical classes,^{[\[103\]](#page-40-0)} to mention a few.

Despite its wide use, three key drawbacks are that i) the approaches offer little design freedom, as the sweep takes place only within a selected pivot class and hence cannot bridge multiple unit cell classes; ii) as the dimensionality gets larger, the density of space-filing sampling drastically drops due to "the curse of dimensionality"^{[\[114\]](#page-40-0)}; iii) sampling only in shape space, even if done well, typically leads to huge bias in property space (discussed in detail in Section [3.4.3\)](#page-13-0).

Multiclass Blending: Blending, or interpolating, across classes offers an avenue to grow a large shape library from a few initial seed classes. This approach differs from Parametric Sweep in that multiple classes jointly contribute to a new instance. Hence, this line of reproduction approaches could be powerful for merging different classes into a unified landscape that includes unseen inter-class instances.

In DMD, some works addressing photonic metasurfaces also utilized image transformations based on Boolean operations (i.e., union, intersection, complement) among canonical classes (e.g., cross, split ring resonator, I-beam) to synthesize freeform inter-class instances.^{[\[64,75\]](#page-39-0)} In the corpus, such class blending for DMD is often followed by deep generative modeling, $[69,72]$ which distills a continuous shape manifold of multiclass unit cells.[\[63,64,75\]](#page-39-0) For a recent method developed in this branch, Chan et al. proposed a versatile Multiclass Blending scheme for functionally graded structures.[\[61,65\]](#page-39-0) The entire scheme combines a weighted sum of seed classes and an activated soft union with lower feasible bounds. This enables the method to be directly integrated into multiscale TO with assurances on both structural integrity (i.e., connectivity within a unit cell) and feasibility (i.e., connectivity among neighbors) simultaneously, while avoiding the restriction of the unit cell to predefined shapes (Figure 3b).

While feasible blending operations are not tied only to settheoretic Boolean operations (e.g., union, complement), relevant works seem to have reported either simple unions $[63]$ or their variants.[\[65\]](#page-39-0) For implementation one needs to specify i) the type of operations (e.g., union; intersection); ii) the number of classes; and iii) rules for weighting factors. We point out that either justifications or analyses on the impact of the choices have been rarely reported in the literature, with a possible exception of Chan et al. $[65]$

Perturbation: Perturbation-based reproduction utilizes heuristics that also allow data expansion. The core idea is to 1) either look up near-boundary or on-boundary instances in the property space of a given dataset at the current iteration, 2) apply geometric perturbations on them in the ambient shape space, often beyond the given representation space, and 3) iterate the perturbation to drive the data acquisition for on-demand purposes (e.g., coverage expansion).

An example of this approach is the iterative database expansion proposed by Wang et al., $[62,66]$ where the radial distortion was recursively applied to sampled freeform building blocks for progressive growth of the property coverage. This reproduction strategy enables extensible data acquisition, which can possibly go beyond user-defined seed instances and unit cell representation, contrary to the two aforementioned strategies. At the heart of its implementation are sampling strategies that enable efficient, property-aware exploration of the shape space.^{[\[62,82\]](#page-39-0)}

Perturbation has been mostly applied to the Pixel/Voxel representations; yet it could be even more effective for other lowerdimensional representations, e.g., a lattice representation using the Parametric Multiclass method (Section [3.2.1\)](#page-4-0), to explore new instances beyond them, as depicted in Figure [3c.](#page-7-0)

3.2.3. Perspectives on Shape Generation

Hinged on the taxonomy presented, we relate individual representations and reproduction strategies, and share our perspectives in multiple aspects.

Shape Dataset as a Design Element: It is generally affordable to produce a bounty of unit cell shapes for DMD without obtaining their properties. Nevertheless, the shape collection needs to be judiciously prepared since i) it primarily determines the landscape to be explored by the downstream tasks; and ii) its utility (which we delineate later in Section [3.4\)](#page-11-0) is related to the resulting property distribution, which is, in general, initially unknown and resource intensive to obtain. Thus, we argue shape data for DMD is a critical design element.

A Trade-Off Between Dimensional Compactness and Expressivity: Any representation is subject to the trade-off between dimensional compactness and expressivity. For example, both the Parametric Multiclass and Implicit Function representations enjoy dimensional compactness. Combined with Parametric Sweep, performing data generation with these is relatively straightforward. However, design exploration could be restricted unless supported by effective reproduction strategies. In contrast, the Pixel/Voxel representation supports freeform topologies without restrictions; yet in practice, the advantage comes with the challenges of: 1) large computational overhead for both design evaluation and machine learning due to poor scalability with respect to the resolution, particularly in 3D metamaterials, as evidenced by coarse resolutions in the relevant literature; 2) efficient design exploration in the vast space, possibly with constraints associated with desirable design attributes, such as manufacturability.[\[75,101\]](#page-39-0) Parametric Curve/Surface allows for free boundary variations but suffers from topological restriction. As another moderate-dimensional representation, CSG offers topologically quasi-free instances, yet its coverage of possible shapes highly depends on the shape generation heuristics that are agnostic to property; hence, it is prone to distributional bias in property space. This property bias could be a hurdle for a data-driven model to accurately learn and perform inference, and trigger the compounding effects of data quality issues in downstream tasks,[\[85,86\]](#page-39-0) or, as it is known in the ML community, "Data Cascade."[\[115\]](#page-40-0) Relevant works that attempted to tackle the property bias are reviewed in Section [3.3.1.](#page-9-0)

Class-Centric vs Class-Free: Depending on the presence of user-defined classes, the representations introduced above can be divided into two groups: class-centric approaches that typically include predefined classes (such as Parametric Multiclass and Implicit Function), and class-free (such as Pixel/Voxel, which is generally the case). By pre-specifying seed classes, class-centric approaches enjoy a database that can take advantage of desirable features inherited from the user-defined unit cell templates, i.e., it can include domain expertise. However, resorting to a particular set of user-defined classes tends to restrict the design freedom early in the procedure of DMD and bias the resulting data distribution in undesirable ways. A workaround applicable during reproduction is to consider Multiclass Blending (Section [3.2.2\)](#page-7-0) introduced earlier, which offers seamless connection across seed classes.[\[65\]](#page-39-0)

Choosing Seed Classes: A crucial step for approaches under the class-centric umbrella is to choose, and justify the choice of, the classes with which to start. They can be chosen based on attributes related to shapes (e.g., topological features, mass/volume, smoothness, manufacturability) or their properties (e.g., elastic anisotropy, performance-to-mass ratio, broadband response). It is also important to ensure shape diversity among the classes, since it secures broad coverage of shape space. Last not but least, it has been pointed out that diversity in the shapes of unit cell data barely contributes to property diversity or task-awareness.[\[85,86\]](#page-39-0) When wider coverage and better uniformity are sought, property diversity, in addition to shape diversity, could serve as a selection criterion of seed classes.^{[\[87\]](#page-39-0)}

Inspiration-Based Data Acquisition: Natural materials that exhibit outstanding properties supported by complicated structures evolved over a long time have been a great inspiration for innovation in design-by-analogy.[\[116,117\]](#page-40-0) Some works dedicated to engineering metamaterials have espoused motifs from biosystems.[\[118–121\]](#page-40-0) Although biologically inspired design provides a compelling avenue to concept generation, relevant works seem to mainly focus on proof-of-concept demonstrations with little design exploration due to grand design challenges such as scalability and repeatability.^{[\[122,123\]](#page-40-0)} We believe that DMD can tackle the challenges by marrying the inspiration from biosystems and data-driven exploration, especially in combination with effective reproduction strategies (Section [3.2.2\)](#page-7-0).

Deterministic vs Stochastic: To date, most datasets prepared for multiscale architectural systems were created based on deterministic representations. On the other hand, a huge line of work addresses multiscale systems whose microstructures are either intrinsically random or subject to irreducible uncertainties associated with system deployment, e.g., material, operating conditions, and fabrications. Such scenarios can be better addressed via stochastic representations. Examples in the literature include the spectral density function approach^{[\[124,125\]](#page-40-0)} proposed for quasirandom nanophotonic structures and photovoltaic cells, and the spinodoid representation,^{[\[90\]](#page-39-0)} which was claimed to be more robust to fabrication imperfection compared to deterministic counterparts. We believe that a comparative study between deterministic and stochastic representations is yet to be explored. Relevant discussion with more focus on the trustworthiness of DMD can be found in Section [4.4.2.](#page-25-0)

Handling of On-Demand Attributes: In choosing a representation, the capability of handling desirable attributes could be decisive. Attributes of potential concern in DMD include symmetry, periodicity, invariance, volume constraint, manufacturability, connectivity among unit cells, and others. In general, the handling of those attributes is easier for explicit, lowdimensional representations, namely the Parametric Multiclass,

Implicit Function, and CSG types under the proposed taxonomy. On the other hand, the Pixel/Voxel representations tend to need special techniques to enforce those attributes, typically with the aid of constraints^{[\[62,82,106\]](#page-39-0)} or data augmentation.^{[\[101,106\]](#page-40-0)}

3.3. Property-Aware Data Acquisition Strategy

Once a shape collection is prepared, design evaluation of all or some of the individual shapes usually follows in order to build training data for semi-/supervised learning (Section [2.2\)](#page-3-0). In literature, the exhaustive evaluation of a large amount of data has been widely employed by means of space-filling sampling, such as Latin hypercube sampling.^{[\[126\]](#page-40-0)} Exam-ples in DMD include the 6D lattice representation,^{[\[93\]](#page-39-0)} the mixedvariable multiclass representation, $[87]$ and the Fourier transform based representation.^{[\[99\]](#page-39-0)} It has been widely used in a parametric space associated with reproduction (Section [3.2.2\)](#page-7-0) as well, such as for exploration of the weight space of some isosurface representations.[\[63,85\]](#page-39-0) The popular use of space-filing design is perhaps attributable to its simplicity and generality of implementation.

However, exhaustive sampling could become intractable when i) the relevant simulation is time-consuming (e.g., high resolution; 3D simulation), ii) the on-demand data size is too large (e.g., more than 100k^{[\[66\]](#page-39-0)}), or iii) the sampling space is too high dimensional (e.g., more than 50D). It could also be the case that one wishes to acquire a data distribution with particular characteristics related to downstream tasks (e.g., negative Poisson's ratio or strong elastic anisotropy). Under such scenarios, it is warranted to exploit the acquisition strategies that take (estimated) property into account as a complement to the aforementioned shape generation heuristics (Section [3.2\)](#page-4-0). Compared to a plethora of works on sampling in the small data regime, not many methods dedicated to DMD with large data have been reported. We introduce some within the context of data acquisition for DMD.

3.3.1. Sequential Acquisition with Active Learning

Active learning^{[\[127–130\]](#page-40-0)} refers to ML approaches that iteratively guide the locations of the next samples. In DMD, it is common to obtain a large pool of shape instances, but have labels for none or only a small portion of it. Under such a case, active learning offers a systematic, efficient, and general route to acquire evaluated samples, and thus could help prepare on-demand data.

In DMD literature, a few works specifically harnessed sequential, heuristic sampling as part of their data acquisition. In an early demonstration, Zhu et al. used a sequential sampling score that aims at property boundary expansion through randomly flipping voxels of near-boundary instances.[\[82\]](#page-39-0) The estimated data density and distance to the boundary were two key criteria that constitute the sampling score. Inspired by that work, Wang et al. developed an iterative stochastic data expansion scheme that builds on a sampling rule accommodating both infilling and gamut growth in the property space.^{[\[62\]](#page-39-0)}

With the aim to develop a method that is widely applicable to data acquisition in DMD, Lee et al. proposed a diversity-based active learning framework specialized in customizing metamaterial datasets with respect to design tasks (**Figure 4**[d,e\)](#page-10-0).[\[86\]](#page-39-0) As opposed to one-shot sampling where the whole samples are collected in a single iteration, sequential data acquisition powered by active learning uses metrics to monitor the growth of a dataset, thus offering a potential answer to a pressing research question in DMD: "How much data?".[\[86\]](#page-39-0) In addition to progressive dataset generation, active learning can serve as a key component for other tasks of data management, such as domain adaptation^{[\[131\]](#page-40-0)} and bias mitigation. An example was shown by Zhang et al., $[132]$ where the proposed entropy-based active learning was demonstrated to substantially reduce the structure-stability bias of two public crystal datasets (Figure [4a–c\)](#page-10-0).

3.3.2. Downsampling Representative Subsets

All our discussion above can be summarized as how to grow a sparse, existing dataset into a large one. Some downstream tasks of data acquisition, however, do not always benefit from using an entire, massive dataset. A prevalent issue of large datasets in DMD is distributional biases. They typically present as containing more of certain shapes or properties, typically in undesirable ways, hosting the issue known as "learning under data imbal-ance" during the downstream tasks.^{[[134\]](#page-40-0)} To this end, Chan et al. proposed a diversity-based subset selection framework built on Determinantal Point Processes,^{[\[135\]](#page-40-0)} a probabilistic way of modeling diversity in relation to the determinant of a similarity matrix. The key idea is to find small yet representative subsets, whose diversity in terms of shape, property, or a joint of them is tunable.[\[85\]](#page-39-0)

Such downsampling could also be useful for training the models that do not scale gracefully to large datasets, such as vanilla $GPs^{[136]}$ $GPs^{[136]}$ $GPs^{[136]}$ that scale cubically with respect to data size for training and inference. When large databases (e.g., more than 20k) are to be used as a ground set of downsampling based on pairwise metrics, e.g., diversity based on Euclidean distance, scalability of the downsampling algorithms becomes critical. It is often resolved through special schemes related to large-scale kernel learning.[\[137–139\]](#page-40-0) Lastly, downsampling can be harnessed for determining a set of initial shapes to serve as seeds during shape generation, as shown by Chan et al., who leveraged Multiclass Blending (Section Multiclass Blending) as their data reproduc-tion strategy.^{[\[65\]](#page-39-0)}

3.3.3. Perspectives on Acquisition Strategy

Sequential Acquisition for Generic Use: Uncertainty vs Diversity: Sequential acquisition can be thought of as designing the rules with which to query an existing pool of unlabeled data, which, in our review, is typically a large number of unit cells with unknown properties. Here, we discuss and compare two key approaches to acquisition for DMD: uncertainty-based sampling, $[140,141]$ and diversity-based sampling.^{[\[65,85,86,132\]](#page-39-0)} Uncertainty-based sampling is centered on improving the prediction confidence of a model, typically resulting in a distributional imbalance that poorly represents the distribution of unlabeled data. Meanwhile, diversitybased sampling focuses on identifying a finite number of landmark data points to combat the distributional bias. Practical implementation of either approach entails deciding the input dimensionality of data and the computational cost of the sampling algorithms.

www.advancedsciencenews.com www.advmat.de

Figure 4. Data acquisition for DMD through active learning. a–c) Entropy-based active learning^{[\[132\]](#page-40-0)} demonstrated by the J-CFID dataset.^{[\[133\]](#page-40-0)} The dataset includes 10 898 instances with seven types of crystal symmetries. a) Kernel density estimation plot of t-distributed stochastic embeddings (t-SNE), where regions with light colors are covered by sparse data. b,c) t-SNE plots of graph embeddings of the materials selected by entropy-based active learning and random sampling, respectively. The proposed entropy-based active learning better covers the sparse regions, hence mitigating bias in the multiclass crystal dataset. Reproduced with permission.^[132] Copyright 2023, AIP Publishing. d,e) Task-aware diversity-based active learning demonstrated by purposefully preferring data with strong anisotropic elasticity.[\[86\]](#page-39-0) The test dataset includes 88,180 instances of freeform pixelated unit cells of orthotropic mechanical metamaterials.[\[62\]](#page-39-0) The resulting property distributions of 3k datapoints in the *C*11-*C*²² space obtained by random sampling and the proposed task-aware active learning, respectively. Reproduced with permission.^[86] Copyright 2023, ASME.

Uncertainty is typically formulated as a model-specific, pointwise function that takes a query point as the input. Including uncertainty within sampling is effective for directly improving the predictive performance of models, whereas the computational complexity escalates as the input dimensionality increases. Within DMD, acquisition methods utilizing uncertainty could be useful when fitting ML models that offer uncertainty quantification, e.g., GPs or Bayesian linear regression, with a low-dimensional representation relative to a large number of data (say $> O(10⁴)$). In general, uncertainty-based acquisition can be conducted based on either frequentist approaches, e.g., random forests and deep neural networks, or Bayesian approaches, e.g., GPs and generalized linear models. For practical guidance on which to employ, readers are referred to Zhang et al.^{[\[141\]](#page-40-0)}

Meanwhile, diversity is frequently modeled as a pair-wise, model-agnostic metric that involves a mapping from a pair of instances to a scalar similarity.^{[\[135\]](#page-40-0)} By harnessing the pair-wise kernel trick, the diversity-based acquisition is capable of handling high-dimensional input instances. However, the acquisition does not gracefully scale with respect to data size due to large storage requirement $(O(N^2)$ where *N* is the data size) and matrix inversions that involve time complexity of $O(N^3)$ unless any large-scale kernel approximations^{[\[137,139\]](#page-40-0)} are employed.

Tailoring Property Distributions: Data acquisition methods that are agnostic to property can lead to datasets that are prone to property bias. For example, **Figure 5**[a,b](#page-11-0) shows the highly biased distributions of formation energy in public crystal datasets. A plethora of sampling methods in low-dimensional input domains have secured datasets with decent coverage as well as **www.advancedsciencenews.com www.advmat.de**

IENCE NEWS

Figure 5. Illustration on distributional bias in the property space of existing DMD datasets. a,b) Stability distributions of two different crystal datasets, 2,953-size OQMD-8^{[\[146\]](#page-40-0)} and 10,898-size CFID,^{[\[133\]](#page-40-0)} whose instances are categorized based on symmetry. Reproduced with permission.^{[\[132\]](#page-40-0)} Copyright 2023, AIP Publishing. c–e) Visualization of the six-bar parametric lattice dataset.^{[\[86,93\]](#page-39-0)} c) Conceptual illustration of the unit cell shape generation based on Parametric Multiclass. d) The near-uniform space-filling design in the projected w_1-w_2 shape descriptor space. e) The resulting property distribution in the C_{11} -*C*₁₃ space. The near uniformity in the weight space leads to a strong bias in the C_{11} -*C*₁₃ space. f,g) The near-zero correlation between shape diversity and property diversity observed during downsampling of the isosurface dataset and active learning applied to the orthotropic freeform dataset, [\[62,86\]](#page-39-0) respectively. f) Reproduced with permission.[\[85\]](#page-39-0) Copyright 2020, ASME. g) Reproduced with permission.[\[86\]](#page-39-0) Copyright 2023, ASME.

uniformity.^{[\[111,142–145\]](#page-40-0)} The quality in the input domain, however, has been found to barely transfer to that in the output domain. Figure 5c–e shows such an example in a mechanical metamaterial dataset. Within DMD literature, this point was observed through the near-zero correlation between shape diversity and property diversity depicted in Figure 5f during downsampling and in Figure $5g$ under active learning.^{[\[85,86\]](#page-39-0)} The implication is that property distributions are likely to be highly imbalanced even though the design in shape space is space-filling, a problem that is currently overlooked in DMD.

Furthermore, for design purposes, addressing property imbalance is only part of data quality assurance. Not all data are equally useful because users frequently have certain preferences in terms of shape, property, or both. For example, a user might wish to collect unit cells of mechanical metamaterials with negative Poisson's ratios, with packaging or shock absorption applications in mind. During data acquisition of photonic metasurfaces, broadband reflectivity might be preferred over narrowband for some defense applications. Therefore, data acquisition for DMD should, ideally, be task-aware so that more resources can be invested in the region of central interest. Figure [4d,e](#page-10-0) illustrates a methodology where batch sequential data acquisition is encouraged to favor samples with high elastic anisotropy.

It is difficult to tailor the property distribution during data acquisition without supervising the properties of interest. This supervision is non-trivial in that 1) properties of unseen unit cells are unknown before evaluation; 2) the evaluation is typically resource-intensive, particularly for large datasets; and 3) distributional control in regression tasks is more challenging and has been under-explored compared to that in classification.[\[134\]](#page-40-0) We believe this topic calls for more research attention.

3.4. Data Assessment

Data assessment in DMD entails quality quantification across candidate sets with respect to either general use or specific

IDVANCED **CIENCE NEWS www.advancedsciencenews.com www.advmat.de**

design tasks. Large sizes of data render the close inspection of individual samples intractable. Thus, their assessment is often conducted through proxy measures such as a quantitative summary of distributional characteristics, or through data visualization for qualitative interpretation. Subjectivity perhaps cannot be totally excluded in data assessment, particularly for DMD applications that involve multiple assessment criteria, e.g., data size, distributional uniformity of data, property coverage, target design tasks, and manufacturability of unit cells. Nevertheless, protocols could help to compare datasets and make decisions on which set is best suited for the intended goal.

Below, we share feasible ideas and scenarios for data assessment that have been drawn from our survey of a large volume of existing metamaterial datasets, and a sparse number of exemplar quantitative/qualitative assessment methods. As reviewers, we hope our discussion will contribute to the establishment of assessment protocols on metamaterial datasets.

3.4.1. Quantitative Assessment with Metrics

Space-Filling Metrics: In general, data acquisition aims to make the data distribution as uniform as possible to ensure any local region of potential interest is equally covered by the dataset. In doing so, some works in DMD employed sequential sampling that included the density, or concentration, of data in certain regions of the design space as part of the sampling utility function.[\[62,66,82\]](#page-39-0) The density in these works was not used to assess dataset quality, but can possibly be used to do so.[\[86\]](#page-39-0) As an alternative to point-wise density, the alternative of set-wise diversity, or pairwise dissimilarity, $[135]$ can also serve as a sampling criterion to suppress distributional biases in both shape and property space, as shown in some DMD works.^{[\[65,85,86\]](#page-39-0)} Point-wise diversity can be measured through information entropy.^{[\[147\]](#page-40-0)} A recent demonstration in literature was performed by Zhang et al., $[132]$ where the coverage imbalance of formation energy across seven crystal systems within the dataset was quantified through point-wise entropy and mitigated by the proposed entropy-based active learning (see Figure [5a,b\)](#page-11-0).

Task-Related Metrics: In DMD, a dataset often ends up being used for a particular design scenario. In such cases, distributional metrics alone may not ensure the on-demand deployment of DMD. Even with a dataset that exhibits perfect uniformity, it could be the case that the region associated with a given design task (e.g., high performance-to-mass ratio, high stiffness anisotropy, broadband reflectivity) happens to be covered by only a few, or even none, of the datapoints. This implies that, if design tasks of interest are given, the assessment on a given dataset must vary. The task specificity of data assessment indicates that if a new task is given, data assessment must realign according to it. In a relevant work, Lee et al. specified a couple of design scenarios involving different design tasks given a shape-only dataset and showed that the resulting property distribution for each case can be tailored through diversity-driven active learning.^{[\[86\]](#page-39-0)}

Within-Dataset Assessment: In case multiple metamaterial datasets share the same input space, i.e., the representation space of building blocks, the quality of each can be comparatively measured based on metrics. Such comparisons could be useful for assessing across multiple data acquisition methods.

For example, Chan et al.^{[\[85\]](#page-39-0)} validated the proposed diversitydriven subset selection method for the isosurface representation by showing larger shape diversity of subsets against that of random, independent sampling. Assuming a similar setting but with more focus on sequential acquisition, Lee et al.^{[\[86\]](#page-39-0)} conceived a measure of diversity gain, which quantifies a relative ratio between the diversity of selected subsets and that of independent and identically distributed (*i.i.d*.) samples with the same data size, to quantify the increase of shape diversity enabled by the proposed sequential sampling strategy. The authors also demonstrated better task-awareness in the two representation spaces, both of which were latent spaces distilled by training generative models.

3.4.2. Qualitative Assessment Through Visualization

Property Coverage: Property coverage offers an intuitive, relative criterion to comparatively gauge utility across datasets, similar to how the Ashby chart visualizes a modulus-density space for disparate materials.[\[148\]](#page-40-0) Upon valid normalization across datasets, data assessment in property space is usually less subjective compared to that in shape space due to lower dimensionality. Intuitive examples in the literature include elastic-ity components,^{[\[62,93,149\]](#page-39-0)} transmission-phase delay at a single frequency,^{[\[91\]](#page-39-0)} and formation energy of crystal structures.^{[\[132\]](#page-40-0)}

Figure [6](#page-13-0) shows an example of a visual comparison between two datasets in a low-dimensional property space. Both carry 3D mechanical metamaterial instances under linear elasticity. Considering the geometrical symmetry, we only consider three components, Young's modulus (E) , Poisson's ratio (v) , and volume fraction (v_f) . The red one denotes the 924-size TPMS dataset gen-erated by Wang et al.^{[\[63\]](#page-39-0)} using Implicit Function and Parametric Sweep. The yellow one denotes the 21 684-size multiclass lattice dataset presented by Chan et al.,^{[\[61\]](#page-39-0)} created with Parametric Multiclass and Multiclass Blending. The pairwise plots of projected properties in Figure [6b](#page-13-0) shed light on some insightful information in a comparative sense including: i) Overall, the multiclass dataset has better data uniformity in terms of Young's modulus (*E*) and volume fraction (v_f); ii) In the *E* − v space, the TPMS dataset has some regions that are covered only by sparse data, arguably attributed to the limitation of Parametric Sweep with a few classes; iii) Although the data size of the multiclass dataset is more than 20 times larger, some property values in the $E - v_f$ space are only available in the TPMS dataset.

In the case where the dimensionality of property is highdimensional, e.g., optical spectra of transmission, dimensional reductions are necessary to visualize the data preferably in 2D space. Zandehshahvar et al.^{[\[151\]](#page-40-0)} shows such an example built on an autoencoder, where the latent space of optical spectra 1) visualizes the coverage dependent on the design complexity of unit cells (**Figure 7**[d,e\)](#page-14-0) and 2) automatically encodes the shift of resonance frequency along the circumferential directions (Figure [7f\)](#page-14-0).

Shape Manifolds: The distribution of data in shape space could give another insight into data assessment in DMD. Despite high dimensionality, ranging from several (e.g., Parametric Multiclass) to millions (e.g., Pixel/Voxel), there is an array of dimension reduction schemes developed for exploratory data analy-sis, such as principal component analysis (PCA),^{[\[152\]](#page-40-0)} t-distributed

Figure 6. An example visual comparison between two 3D mechanical metamaterial datasets. a) Examples of unit cells in the 924-size TPMS dataset.^{[\[63\]](#page-39-0)} (red, top) and the 21 684-size multiclass lattice dataset.[\[61\]](#page-39-0) (yellow, bottom). b) Pairwise plots of the properties of interest: Young's modulus (*E*), Poisson's ratio (v), and volume fraction (v_f). The effective properties are computed by energy-based homogenization.^{[\[150\]](#page-40-0)} The plots located at the diagonal depict a histogram of each component. All the off-diagonal components show scatter plots of two different properties. The upper triangle plots are shown considering the symmetry. The volume fraction ranges from 0.25 to 0.8.

stochastic neighbor embedding $(t\text{-SNE})$, $^{[153]}$ $^{[153]}$ $^{[153]}$ and uniform mani-fold approximation and projection (UMAP).^{[\[155\]](#page-40-0)} The projection is conducted preferably into 2D spaces for straightforward visualization. This can uncover the underlying characteristics of the data distribution, e.g., clusters formed by data acquisition, often dictated by reproduction strategies of unit cells (Section [3.2.2\)](#page-7-0).

For example, in DMD, Ma et al.^{[\[76\]](#page-39-0)} employed a VAE^{[\[69\]](#page-39-0)} and then visualized the latent space using t-SNE. The visualization projected in a 2D space reveals the clustering across the seed eight classes learned in an unsupervised manner (Figure [7\)](#page-14-0). Mean-while, Wang et al.^{[\[66\]](#page-39-0)} employed PCA to claim the versatility of the proposed latent representation learned by a conditional VAE. The visualization was used to delineate that the continuous, interpretable latent space offers simple interpolation across building blocks, a shape similarity measure, and intrinsic clustering of associated properties.

An even more interesting use of such high-dimensional data visualization is for comparisons across datasets. Employing the

one-class support vector machine,^{[\[154\]](#page-40-0)} Zandehshahvar et al. visually demonstrated the impact of geometric freedom in building blocks by visualizing different coverage in both shape and property space (see Figure [7d–f](#page-14-0) for illustration).

3.4.3. Perspectives on Data Assessment

Task Specificity of Data Acquisition and Assessment: In Task-Related Metrics under Section [3.4.1,](#page-12-0) we covered the task specificity of data assessment with a particular focus on metrics. We now summarize our general view on data acquisition and data assessment of DMD for commonplace scenarios as follows:

a) Provided that no target tasks have been specified, data acquisition can aim to create a dataset for generic use, i.e., focus on uniformity and wide coverage, in both the shape and property

Figure 7. Visualization of high-dimensional data. a) Visualization of the latent space in the orthotropic mechanical metamaterials dataset with the property distributions included.^{[\[62\]](#page-39-0)} The 16D latent representation distilled by VAE is visualized in 2D space using principal component analysis.^{[\[152\]](#page-40-0)} Reproduced with permission.^{[\[62\]](#page-39-0)} Copyright 2020, Elsevier B.V. b,c) Visualization of the latent space in plasmonic metasurfaces.^{[\[76\]](#page-39-0)} b) The representative images of the seed classes included in the dataset. c) The resulting data distributions in the 2D latent space using t-distributed stochastic neigh-bor embedding.^{[\[153\]](#page-40-0)} Each class forms separate clusters in the latent space, projected from 20D into 2D. Reproduced with permission.^{[\[76\]](#page-39-0)} Copyright 2020, Science China Press and Springer-Verlag GmbH Germany. d–f) The latent space representation of the resonant reflection spectra in dielectric metasurfaces.[\[151\]](#page-40-0) d) Metasurface unit cells with five different levels of geometric complexities. e) The corresponding convex hulls of shape mani-folds estimated through the one-class support vector machine.^{[\[154\]](#page-40-0)} f) The corresponding property distribution of high-dimensional optical spectra in the property manifold. It encodes the shift of resonance frequency with respect to the traversal along counter-/clockwise directions. Reproduced with permission.[\[151\]](#page-40-0) Copyright 2020, American Chemical Society.

spaces of building blocks. The data assessment can also follow the same criteria without preferring a certain region in those spaces. Ref. [\[85\]](#page-39-0) shows an example in literature.

- b) Even without being given any specific on-demand properties a priori, instance-wise preferences related to shape (e.g., fabrication feasibility), property (e.g., high physical anisotropy), or both (e.g., performance-to-mass ratio) can be enforced during data acquisition to tailor the data distribution as desired with minimal trial-and-error. The data assessment approach must address both distributional metrics and task-related metrics. Ref. [\[86\]](#page-39-0) is an example.
- c) If a target downstream task is prespcified , or a set of target tasks is given, the data acquisition and assessment can be aligned to the specified task(s), in addition to data uniformity. In these cases, data uniformity is useful only within the domains associated with the tasks. Moreover, the assessment is subject to the definition of target tasks. A concrete design example of mechanical metamaterial systems where the task

of matching target displacements at the system level is addressed in ref. [\[62\]](#page-39-0) and discussed in Section [5.4.3.](#page-36-0)

Assessment Protocols: Data assessment is essential for either diagnosing a dataset or choosing the best among competing ones. How to fairly measure the quality of metamaterial datasets is key to decision-making over competing datasets, as well as to minimization of iterations among the modules of DMD. A general guideline for the assessment of synthetic datasets for engi-neering purposes was recently proposed.^{[\[156\]](#page-40-0)} However, not many attempts that are dedicated to data assessment for DMD have been reported, compared to the rapidly growing volume of the corpus. Without agreement upon standard protocols of data assessment, it is difficult to judge the quality of individual works that include data acquisition and to draw meaningful conclusions among them. Thus, we assert the need for more research efforts centered on data assessment.

Benchmark Datasets: The easy access to public datasets has been an enabler of the recent surge of ML. Ideally, newly proposed methods on any module of DMD should be validated through diverse benchmark datasets that are suggested by the communities. In the corpus of data-driven design, however, such solid validation seems difficult to find. A profound reason that applies to general data-driven design is, arguably, a dearth of public datasets and benchmarks.[\[47,156,157\]](#page-39-0) Echoing this, we argue that securing more public datasets and setting some of them as benchmarks will be the first step towards the research practice that prompts quantitative, rigorous comparisons of relevant works and reproducible research, hence helping readers better appreciate individual works with relation to the field. In doing so, it is highly encouraged to make new datasets publicly available, preferably in online repositories that support consolidation across datasets.

3.5. Discussion

3.5.1. Limitations of Unit Cell Datasets

To date, most demonstrations of data-driven multiscale design have been built on building block datasets. Each data point, typically a structure-property pair, is generated based on homogenization under the assumptions that i) there is scale separation, and ii) the system of interest is either deterministic under periodicity (e.g., crystal) or stochastic following the ergodic principle and therefore stochastically periodic (e.g., polycrystal; spin-odoid systems^{[\[90\]](#page-39-0)}). The conditions imply that, in datasets of deterministic systems, the homogenized properties can be inaccurate, possibly substantially, to describe the unit cell behaviors in the multiscale systems. Nevertheless, some recent works, which accelerated the design of fully aperiodic systems using the homogenized properties, have claimed that the accumulated errors at the system level are acceptable under special conditions, e.g., when including geometric/mechanical compatibility under lin-ear elasticity,^{[[34,66\]](#page-39-0)} functional grading,^{[\[61,65\]](#page-39-0)} and uncoupled oper-ations in photonic metasurfaces,^{[\[91\]](#page-39-0)} where the accumulated deviation of the homogenized properties at the system level was claimed to be acceptable.

We point out that while the homogenization assumption has been a backbone of the recent progress in DMD for both periodic/non-periodic and deterministic/stochastic systems, it is also a hurdle that impedes researchers from exploring problems beyond academic benchmarks where the assumption does not hold true. There are a variety of cases where the deviation of effective properties under periodic boundary conditions at the system level is more than acceptable.^{[\[50\]](#page-39-0)} Examples include the systems under: i) large deformation, $[45,158]$ ii) strong local coupling among neighbors,^{[\[159,160\]](#page-40-0)} iii) long-range interactions,^{[\[161,162\]](#page-40-0)} and iv) heterogeneous loading conditions.[\[140,163,164\]](#page-40-0) Preparing the datasets that take a supercell (i.e., a collection of neighboring unit cells) instead of a single unit cell as a datapoint could offer a simple yet powerful extension to the unit-cell-based approaches, as shown by refs. [\[140, 160, 165\]](#page-40-0). The extension would involve increased computational cost and validating the boundary conditions applied to the supercell simulations. A relevant discussion at the system level can be found in Section [5.4.2.](#page-35-0)

3.5.2. Learning Global Responses vs Local Responses

Many works reported in DMD include a surrogate model that directly maps the parameterized unit cells to effective, homogenized properties, such as elasticity components in mechanical metamaterials $[61-63,65]$ and scattering parameters in photonic metasurfaces.[\[42,75,91\]](#page-39-0) In general, the associated mapping that returns a low dimensional output is expected to be learned within an acceptable amount of error, provided that 1) there is a good amount of training data to uncover the underlying pattern, 2) the output dimensionality is low, and 3) the tuning parameters of the model have been explored thoroughly enough. Nevertheless, this type of learning which targets global responses comes at a significant loss of information available in simulations and experiments, since the model has access only to high-level, typically condensed, quantities of specific interest, not to raw fullfield data.

A workaround that can boost the generality of the structureproperty mapping is to directly learn output physical fields, e.g., displacement, electric fields, and temperature, as a function of parameterized unit cells. The goal can be achieved through either physics-informed ML or operator learning, $[166-170]$ where an underlying physics is either explicitly imposed as a constraint or discovered by field data. Such a mapping can capture underlying spatial correlations of fields, which could be subject to strong long-range interaction across unit cells in DMD. Within DMD, an early demonstration of physics-informed ML was reported by Fang et al.^{[\[171\]](#page-41-0)} in 2019. Chen et al.^{[\[172,173\]](#page-41-0)} further investigated the potential of physics-informed ML for DMD under more challenging scenarios that include noise field data and high-index ma-terial. Lim et al.^{[\[174\]](#page-41-0)} followed with a particular focus on modeling and inverse design of aspheric micro-lenses. Recent years have seen growing interest in operator learning, the other major branch, as well. In 2022, Lu et al.^{[\[175\]](#page-41-0)} proposed a multi-fidelity neural operator for efficient sampling for modeling nanoscale heat transport. Augenstein et al.^{[\[176\]](#page-41-0)} harnessed a Fourier Neural Operator for field-to-field modeling and inverse design of 3D quasi-free scatters specified by the VAE latent representation. In the meantime, Zhelyeznyakov et al.^{[\[159\]](#page-40-0)} employed dynamic mode decomposition to predict the full near-field response of an electromagnetic metasurface array using a sparse representation.

These field-to-field modeling approaches feature decent model transparency, generality, and sample efficiency. In practice, however, their implementation involves non-trivial challenges that include 1) how to effectively regularize the learning with regard to high-dimensional output fields (e.g., adding a sparsity penalty to avoid overfitting^{[\[166\]](#page-41-0)}) and 2) how to impose priors associated with physics (e.g., smoothness of fields $[177]$). The extrapolation capability of field-to-field modeling has been an open research question. Limiting the scope to operator learning, Zhu et al.^{[\[178\]](#page-41-0)} shared some insightful discussions under two scenarios for extrapolation, each of which assumes either governing equations are known or sparse data in the extrapolation region of interest is given.

Pursuing a more direct approach toward one-to-many inverse design, some recent efforts have been dedicated to constructing a spatiotemporal mapping from on-demand nonlinear properties to associated physical fields through denoising diffusion models.^{[\[179,180\]](#page-41-0)} Examples include Vlassis et al.^{[\[181\]](#page-41-0)} and Bastek

CIENCE NEWS www.advancedsciencenews.com www.advmat.de

et al.,^{[\[182\]](#page-41-0)} where a probabilistic inverse map from a target nonlinear stress-strain behavior (e.g., large deformation) to a full-field response (e.g., stress fields in deformed configurations) is constructed to accelerate on-the-fly, one-to-many inverse design. To date, the relevant reports seem sparse, particularly for such nonlinear cases where naively feeding undeformed configurations to ML models may not suffice to capture the spatiotemporal causal relation. We believe that there is room to place more attention on both field-to-field modeling and property-to-structure modeling. Details of this topic are reviewed in Section. [4.2.4.](#page-20-0)

3.5.3. Determining Data Size

In one form or another, "How much data?" has been a key research question in data-driven approaches. Within the scope of DMD, this question affects the following aspects: 1) model com-plexity (e.g., neural networks vs GPs^{[\[136\]](#page-40-0)}), 2) unit cell representation (e.g., high-dimensional vs low-dimensional), and 3) simulation cost/fidelity, among others. Due to the multifaceted nature of DMD problems, it is difficult to predict an "optimal" data size a priori, especially via one-shot sampling. The data sizes reported in the literature could be a good starting point, provided that a new design task of interest shares some attributes, e.g., model complexity and unit cell representation, with those of the reported works. Integrating active learning with data acquisition (Section [3.3.1\)](#page-9-0) can be a more rigorous, general approach to determining data size since it offers metrics related to either the data themselves or model performance. Lee et al.^{[\[86\]](#page-39-0)} claimed that diversity metrics can be monitored to gauge the relative utility of incoming data, hence serving as a proxy to determine data size based on the gamut growth in property space. For generic scenarios, a guideline on the size of engineering datasets was proposed by Picard et al.^{[\[156\]](#page-40-0)}

In the corpus, we observe most prior efforts were hinged upon large data. It is equally worth investigating in the small data regime, $[183]$ as doing so will help tackle design problems that involve expensive simulations and limited computational resources. This call resonates with data-centric artificial intelligence, $[184-186]$ an initiative that propels a paradigm shift of data acquisition from "more data" to "better data."[\[187\]](#page-41-0)

3.5.4. Data Sharing Practice and Reusability

The surge of DMD has inspired the emergence of open-source data-sharing platforms, such as *NanoMine*, [\[188–190\]](#page-41-0) which pays special attention to polymer nanocomposites. *MetaMine*, its sister platform, currently stores 300k structure–property data of metamaterials with a diverse array of unit cell representations. Building a common platform and knowledge representation presents immense challenges. The endeavor will facilitate consolidating datasets that were acquired independently, and therefore enhance the potential of DMD beyond what is achievable by an individual dataset.

A prerequisite to building a user-interactive data platform that supports reusability and reproducibility is sharing protocols. For example, the FAIR (Findability, Accessibility, Interop-erability, and Reusability) principles^{[\[191\]](#page-41-0)} is a concise, domainindependent, and thus generic, guide for data sharing. Exercising the FAIR principles is built upon core elements such as standardized vocabularies, ontologies, and data formats. Despite the foundational role that such general guidelines have played in existing data-sharing platforms, they tend to only specify broad guidelines of data quality assessment at a high level, leaving the needs of discipline-specific standards unaddressed.^{[\[192\]](#page-41-0)} For DMD, the proper format of (meta)data could differ wildly across different domains (e.g., nanocomposites vs mechanical/photonic metamaterials). Even within a given domain, it could be difficult to define a set of commonly structured vocabularies or knowledge representations that accommodate all datasets submitted by users. In this regard, we advocate for an extensible, dynamic platform, which starts with initial vocabularies and schema defined by humans, as showcased by *Nanomine* for poly-mer nanocomposites,^{[\[190\]](#page-41-0)} and which is then allowed to evolve without supervision as more data is ingested.

Despite the advent of such data-sharing platforms, we point out that the current practice in data-driven multiscale architectural design typically starts with creating one's own dataset, rather than with existing ones; arguably, the practice is what has led to "per-task" datasets involving a similar/equivalent end-use. To avoid the trial-and-error and computational resources that data creation from scratch demands, new works can be conducted by either i) creating a versatile, high-quality dataset that can address multiple, disparate design tasks, or ii) customizing a public dataset with respect to new design tasks. We believe that the reusability of datasets is by no means trivial to achieve, and thus deserves further research attention.

3.5.5. Other Tasks Involving Data

So far we have primarily covered data acquisition. Other possible tasks include data augmentation, data consolidation, bias reduction, problem/domain adaptation, and exploratory data analysis. Each task plays a unique role that cannot be fully addressed by data acquisition itself. For example, it is highly recommended to perform data augmentation because 1) it increases the amount of training data without further evaluations; 2) it helps an ML model to be encoded with operational invariances, such as rotation, scaling, translation; and 3) it tends to mitigate overfitting by serving as a regularizer of model training. Within DMD the efficacy of augmentation has been demonstrated in some works.^{[\[101,106\]](#page-40-0)} We believe that further research on data-related tasks other than data acquisition will underpin the future success of DMD by enhancing the generality, customizability, and reusability of data.

3.5.6. Public Resources

We share a link¹ to an online webpage of public resources that are associated with data-driven design.

4. Data-Driven Unit Cell Design of Metamaterials

In Section [3,](#page-4-0) we reviewed past works related to metamaterials data acquisition and discussed some challenges at the data

¹ https://github.com/ideal-nu/Data-Driven-Design-for-Metamaterialsand-Multiscale-Systems-Status-and-Opportunities

IDVANO **CIENCE NEWS www.advancedsciencenews.com www.advmat.de**

acquisition stage. In this section, we introduce how past works used data-driven methods to solve unit-cell-level metamaterials design problems.

4.1. Overview

The advancement of ML has motivated researchers to seek datadriven solutions to many real-world design challenges. These challenges mainly originate from the following factors:

- 1) Analysis using physical experiments or high-fidelity simulations usually requires high cost. For example, numerical nanophotonic simulations can take hours or even days for complex systems.[\[193\]](#page-41-0)
- 2) Advanced fabrication technology (e.g., additive manufacturing, micro/nanofabrication) enables high degrees of design freedom, but exploring a high-dimensional design space to find optimal solutions is challenging.
- 3) While methods, like adjoint-based shape optimization and TO, can address high-dimensional design problems by using sensitivities to guide optimization, sensitivities are usually expensive to numerically derive when the physics governing the problem is non-differentiable with respect to design variables.

In this section, we will introduce how past work used datadriven methods to solve these challenges, particularly in the domain of unit-cell-level metamaterials design. Note that datadriven methods have been gaining increasing attention in engineering design, mainly for shape and topological designs, to address the challenge brought by their high degrees of design freedom. For shape optimization, a large body of work looked at datadriven aerodynamic shape optimization, which mainly focused on dimensionality reduction or representation learning (e.g., Refs. [\[194–198\]](#page-41-0)) and inverse design (e.g., Refs. [\[199–201\]](#page-41-0)). Compared to only considering shape variation in shape design, metamaterials design can usually accommodate topological changes depending on their functional requirements. These extra degrees of freedom make it more difficult for traditional design methods to solve metamaterials design problems. Many works on MLassisted TO serve various purposes such as reparameterization, objective function prediction, sensitivity prediction, direct prediction of TO solutions, and enhancing diversity for generative design. We refer interested readers to ref. [\[48\]](#page-39-0) for a summary of past contributions of neural network-based TO methods. These TO methods were usually applied to structural design problems, where well-established physics and sensitivity analysis are available. In this section, we will cover metamaterials design under different physics (e.g., mechanical, optical, acoustic/elastic, thermal) scenarios, for some of which sensitivity analysis is either unavailable or too difficult. Due to that reason, traditional gradientbased TO methods might not be applicable to certain types of metamaterials design.

The benefits of data-driven methods highly depend on 1) the cost of data collection and learning and 2) the acceleration contributed by the data-driven model. To make such methods costeffective, the trained model needs to be reusable for different tasks. Problems like structural optimization are usually subject to task-dependent objectives and constraints. In this case, it is

difficult to train a single ML model that can be reused in different tasks. In contrast, unit-cell-level metamaterial design problems under the same physics usually share common properties of interest (e.g., elasticity properties in mechanical metamaterials and transmission and phase delay in optical metamaterials), regardless of different functional goals (e.g., design for compliant mechanism, energy absorption, or noise reduction). Thus, a data-driven model trained on the same properties of interest can be reused in different metamaterials design tasks.

This section covers five types of metamaterials—optical, acoustic/elastic, mechanical, thermal, and magneto-mechanical. The trend of applying data-driven methods to unit-cell-level design began around 2018. In the domain of mechanical metamate-rials, to our best knowledge, Chen et al.^{[\[36\]](#page-39-0)} and Gu et al.^{[\[37,38\]](#page-39-0)} first proposed new data-driven methods to discover auxetic metamaterials with extreme properties and novel composite designs in 2018. In the same year, Liu et al.,^{[\[39\]](#page-39-0)} Inampudi and Mosallaei,^{[\[40\]](#page-39-0)} Ma et al.,^{[\[41\]](#page-39-0)} Malkiel et al.,^{[\[83\]](#page-39-0)} and Liu et al.^{[\[42\]](#page-39-0)} were among the first who applied data-driven methods to assist optical metamaterial design. After 2020, data-driven methods were incorporated into more domains including the design of acoustic/elastic, thermal, and magneto-mechanical metamaterials.^[43-45] Despite differences in physics, they share similar design scenarios: 1) design allows high-degree of geometric freedom; 2) physical properties are usually the target quantities of interest; 3) physical properties depend on design geometry; 4) generating random design geometries is cheap whereas computing their physics properties is expensive. For this reason, data-driven methods developed for metamaterials are usually applicable under different types of physics. Thus, we focus on the goals of data-driven methods instead of metamaterial types when characterizing past work. Specifically, we categorize the goals into two major types — iterative design optimization (Section 4.2) and iteration-free inverse design (Section [4.3\)](#page-20-0). **Figure [8](#page-18-0)** illustrates the relationship between the goals of proposed data-driven methods, the physics being considered, and the ML models, extracted from 56 representative prior studies. These prior studies were selected from publications from 2018 to 2023, with the aim to use machine learning to assist metamaterials design. We focus on only single-scale metamaterials design in this section, where unit cell designs are arranged periodically in space. In Section [5,](#page-26-0) we will introduce multiscale design, where aperiodic unit cell designs are considered to achieve spatially-varying material properties. Note that since machine learning applied to metamaterial design has gained popularity, the reviewed publications represent merely a subset of the extensive body of prior research in this field. We select the 56 publications based on their impact (as evidenced by citation counts and journal impact).

4.2. Iterative Design Optimization

In this section, we review prior works that employed ML methods for metamaterials design optimization. In particular, ML commonly plays roles in accelerating property evaluation (Section [4.2.1\)](#page-18-0), learning more efficient design representation (Section [4.2.2\)](#page-19-0), sequential decision making (Section [4.2.3\)](#page-19-0), and physics-informed solution generation (Section [4.2.4\)](#page-20-0).

CIFNCF NFWS **www.advancedsciencenews.com www.advmat.de**

Figure 8. Categorization of physics, goals, machine learning models, and their relations based on 56 representative prior works. (LR: logistic regression; RL: reinforcement learning; PCR: principal component regression; DT: decision tree; DGM: deep generative model; GP: Gaussian process; MLP: multilayer perception; CNN: convolutional neural network; RNN: recurrent neural network; GNN: graph neural network; PINN: physics-informed neural network.)

4.2.1. Accelerated Optimization via Data-Driven Property Prediction

In unit-cell-level metamaterials design, material properties are usually the design target. For example, absorption spectra and dispersion relations can be properties of interest for optical and acoustic metamaterials design, respectively;^{[\[202,203\]](#page-41-0)} for mechanical metamaterials design, properties of interest can be Young's modulus, Poisson's ratio, and volume fraction.^{[\[63\]](#page-39-0)} We can perform numerical simulations or experiments to evaluate material properties. However, depending on different physics properties, the computational cost can be prohibitive, especially when iterative design optimization is required. Datadriven models can learn complex structure-property relations and hence surrogate time-consuming simulations or experiments, allowing high-throughput property evaluation. This can accelerate the design process when combined with downstream design space exploration methods (e.g., sampling, screening, and optimization).

Based on the reviewed literature (Figure 8), most MLaccelerated design optimization works were based on this approach, among which the most commonly used ML models were convolutional neural networks $(CNN)^{[37,45,204-208]}$ and multilayer perceptron (MLP).^{[\[40,159,209–212\]](#page-39-0)} Typically, CNNs were used for pixelated design representations with high geometric freedom or design dimensionality (allowing more complex designs), while MLPs were used for parametric or shape designs with lower design dimensionality. These two models were mostly used when the training data size was larger than 700. One special case is ef. [\[159\]](#page-40-0), where the local design variables of only 10 metasurfaces along with the local patches of their electromagnetic (EM) fields response were used as training data of a property-prediction MLP, resulting in 123 210 actual training samples.

While the training of most neural network-based models normally requires large datasets, GP was employed when there was a relatively small amount of training data.^{[\[87,213\]](#page-39-0)} GP's ability to estimate uncertainty makes it well-suited for adaptive sampling and Bayesian optimization. These are useful design techniques especially when the computation of responses is time-consuming. However, one limitation of standard GP models is the difficulty in handling large datasets due to its $O(N^3)$ time complexity and $O(N^2)$ memory complexit'y, where *N* is the training data size. This also limits the dimensionality of design problems to be considered, because the required amount of data scales exponentially with the dimension due to the curse of dimensionality. Past work proposed ways to make GP more scalable with larger datasets,[\[95,214–218\]](#page-39-0) which can potentially expand the use cases of GP to larger datasets and higher problem dimensions in metamaterials design.

IDVANO **CIENCE NEWS www.advancedsciencenews.com www.advmat.de**

Dimensionality reduction was also applied to reduce the dimensionality of the original design or property space before applying regression models for property prediction. Wang et al.[\[213\]](#page-41-0) used a Gaussian mixture beta variational autoencoder (GM- β VAE) to reduce the dimension of the pixelated metamaterials design representation. Chen et al.^{[\[36\]](#page-39-0)} employed principal component regression (PCR) to reduce the 3D metamaterial design parameters, where the PCA was followed by linear regression to predict the elastic material properties of mechani-cal metamaterials. Zhelyeznyakov et al.^{[\[159\]](#page-40-0)} reduced the dimension of the near-field response of the metasurface using singular value decomposition (SVD) before applying property prediction models.

In some works, decision trees (DTs) were used as the property prediction model due to their interpretability and flexibility in learning non-linear structure-property relations. Elzouka et al.[\[219\]](#page-41-0) predicted the spectral emissivity of dielectric and metal-lic particles. Chen et al.^{[\[102\]](#page-40-0)} used Generalized and Scalable Opti-mal Sparse Decision Trees (GOSDT)^{[\[220\]](#page-41-0)} to predict band gaps in different frequency ranges based on shape-frequency features extracted from acoustic metamaterial geometries. Particularly, the interpretability of DTs allows us to extract explicit design rules that can guide inverse design, which we will elaborate on in Section [4.3.4.](#page-23-0)

In most prior work, metamaterials design variables were either represented as a vector of design parameters or a tensor representing pixelated designs, while the properties of interest normally consisted of single or multiple scalars. Beyond these com-mon representations, Yang et al.^{[\[221\]](#page-41-0)} studied 3D graphene metamaterials with a graph representation and used a graph neural network (GNN) to predict the local atomic stress distributions. Sajedian et al.^{[\[204\]](#page-41-0)} aimed at predicting the absorption curve of plasmonic structures and solved this problem with a recurrent neural network (RNN).

To obtain the metamaterial design solution, a common approach is to use the property prediction model as a surrogate design evaluation model and incorporate it into any iterative optimization loop.[\[40,45,159,208,209,212,213\]](#page-39-0) Besides iterative op-timization, past works also employed virtual screening^{[\[221\]](#page-41-0)} and sampling^{[\[37,102,210\]](#page-39-0)} to select design solutions, which also took advantage of the fast design evaluation capability of property prediction models.

4.2.2. Accelerated Optimization via More Efficient Design Representation

Another way to accelerate design optimization is through learning an efficient design representation (i.e., latent representation) that is more compact than the original representation but still as expressive, thereby covering the same design space with fewer design variables. This benefits in three ways: 1) it mitigates the issue caused by the curse of dimensionality when training property prediction models, thus lowering the requirement for training data size and model complexity; 2) it enables more efficient optimization, since searching for global optimal solutions in a lower-dimensional latent space is easier and faster; 3) it allows easier downstream analysis such as data visualization, clustering, and arithmetic operations in the latent space. Since we introduced past work on using dimensionality reduction for property prediction in Section [4.2.1,](#page-18-0) this section will focus on the other two benefits.

As shown in Figure [8,](#page-18-0) most prior works under the category of "representation learning" used DGMs. Liu et al.[\[70\]](#page-39-0) employed a VAE to learn a lower-dimensional latent space of pixelated optical metasurface designs and leveraged an evolutionary algorithm to optimize designs over the latent space. Wang et al.[\[66\]](#page-39-0) constructed an end-to-end model combining a VAE with a mechanical property regressor, and learned a lower-dimensional, structured latent space organized by mechanical properties of pixelated metamaterial designs. Semantic operations (e.g., moving from "low stiffness" to "high stiffness") can be achieved by simply moving in certain directions of the resulting latent space. Chen et al.^{[\[222\]](#page-41-0)} proposed a GAN with hierarchical latent spaces to simultaneously represent the "as-designed" and "as-fabricated" optical metasurfaces. The model not only learned a compact latent representation for "as-designed" metasurfaces, but also captured the geometric uncertainty of "as-fabricated" designs, which enables efficient and robust design optimization under fabrication uncer-tainty. Shen and Buehler^{[\[223\]](#page-41-0)} used StyleGAN^{[\[224\]](#page-41-0)} to learn disentangled latent spaces that capture attributes and variations at different levels. Design optimization and geometric manipulations (i.e., projection, encoding, and mixing) can be achieved by optimizing the latent vectors.

Autoencoders were also used for compressing the dimension of the metamaterials design space.[\[81,225\]](#page-39-0) Compared to DGMs like VAEs and GANs, autoencoders only minimize the reconstruction errors of training samples without considering the continuity of the latent space, which may reduce the efficiency of design optimization and latent space analysis.

4.2.3. Design as Sequential Decision Making via Reinforcement Learning

The design problem can also be modeled as a sequential decisionmaking process, which can be solved by reinforcement learning (RL). As introduced in Section [2.2.4,](#page-3-0) the main components to be defined in any RL tasks are state, action, and reward. Defining these three concepts — especially the action — in a design problem needs extra consideration, unlike canonical RL problems such as gameplay and adaptive control, since it is more intuitive to treat metamaterials design as a standard optimization problem rather than a dynamic problem that requires sequential decision-making. Nonetheless, Sajedian^{[\[80\]](#page-39-0)} employed the Double Deep Q-Network (DDQN)^{[\[226\]](#page-41-0)} to design both the material type and the geometry of the optical metasurface that maximizes hologram efficiency. The actions were defined as increasing or decreasing each design parameter by a fixed amount; the state is design parameters; the reward considers both the phase-generating capability and the efficiency. Similarly, Liu et al.^{[\[44\]](#page-39-0)} used the DDON to design a periodic lattice system that achieves thermal transparency. The action was defined similarly to the one in ref. [\[80\]](#page-39-0); the state is the combination of design parameters and the response of heat fluxes; the reward is represented by how far the simulated system is away from achieving perfect thermal transparency. Sui et al.[\[81\]](#page-39-0) proposed a collaborative deep Q network (DQN) that designs mechanical metamaterials (voxelated designs

 Q_{net2} value

Max

Figure 9. "Flipping process" of the collaborative DQN method in ref. [\[81\]](#page-39-0): one agent selects a soft voxel and turns it into a stiff voxel, then the other agent does the opposite. Reproduced with permission.[\[81\]](#page-39-0) Copyright 2021, American Chemical Society.

Min

representing the arrangement of soft and stiff materials). The action is the "flipping process" by two agents: one agent selects a soft voxel and turns it into a stiff voxel, while the other agent does the opposite (**Figure 9**); the state is represented by the voxelated design; the reward is the change of the averaged equivalent modulus.

Soft

Voxel

Min

 a_l

 Q_{net1} value

Max

 s_k

Sui et al.^{[\[81\]](#page-39-0)} compared RL with the genetic algorithm (GA) under different design dimensions. The results show that RL does not have an advantage over GA under small action spaces, but will outperform GA when the action space becomes larger, due to the generalization ability of deep neural networks on large action spaces. On the other hand, with a larger action space, more design evaluations will be required to sufficiently explore the action space, which can quickly lead to a prohibitive computational burden for RL. This was reflected by the design problem dimensions addressed by the reviewed past works, none of which exceeded 50 dimensions.

Based on these observations, we conclude that it is important to find the "sweet spot" of action space dimensions where RL can outperform classic optimization methods while not requiring prohibitive computational costs. When proposing RL methods for metamaterials design, one needs to compare it to optimization (either classic or ML-accelerated optimization as introduced in Sections [4.2.1](#page-18-0) and [4.2.2\)](#page-19-0) and justify the necessity of using RL instead of optimization methods that are usually more intuitive and simpler to formulate. Despite the caveats, RL can still be a promising technique in metamaterials design because 1) compared to data-driven design optimization, RL does not require prior data and hence is not limited by the boundary of existing designs, and 2) it is easier to formulate the design problem with RL when the design has an unstructured representation (e.g., irregular truss structure represented as graphs^{[\[227\]](#page-41-0)}).

4.2.4. Design via Physics-Based Learning

While physics-informed ML has drawn huge attention in recent years, its application in metamaterials design is relatively limited. Physics-based learning in design tasks usually uses governing equations to guide the training of ML models which produce optimized design solutions. Jiang and $Fan^{[228,229]}$ $Fan^{[228,229]}$ $Fan^{[228,229]}$ proposed a generative neural network to generate high-performance dielectric metasurfaces, where the generator training was guided by the

gradients from the adjoint electromagnetic simulations of gener-ated designs. Lu et al.^{[\[230\]](#page-41-0)} proposed physics-informed neural networks with hard constraints (hPINNs) to solve the TO problem in metamaterials design. The method builds on physics-informed neural networks (\overline{PINNs} ^{[\[167\]](#page-41-0)} but further allows optimizing a design objective function as well as the governing partial differential equations (PDEs) being modeled as hard constraints. The hPINNs method was demonstrated on design problems in optics and fluids. Compared to PDE-constrained adjoint-based optimization methods, the hPINNs method achieved the same objective value, but obtained a simpler and smoother solution with faster convergence (**Figure [10](#page-21-0)**).

Unlike classic ML or data-driven methods, these physics-based learning methods require no training data and are less susceptible to the curse of dimensionality. The optimization can be guided by gradients and hence there is no need to explore the entire solution space. Therefore, design via physics-based learning can be a promising future research direction. On the other hand, we should also note that common physics-based learning methods such as PINNs may have stability issues in training, due to their complicated loss functions and the need to solve highly non-convex optimization problems.^{[\[231\]](#page-42-0)} Besides, most studies only contain validation on simple academic problems, and lack evidence of efficiency in more complex and practical use cases. These gaps need to be addressed in future research.

4.3. Iteration-Free Inverse Design

While the reviewed works in Section [4.2](#page-17-0) used ML methods to accelerate design optimization, they still need an iterative optimization process to obtain the final solution. In this section, we review past works that used ML to achieve iteration-free inverse design. There are mainly two application scenarios: 1) obtaining designs that meet target properties or responses; 2) obtaining near-optimal solutions under certain constraints or operating conditions. Most prior works addressed the first scenario. Existing ML methods for iteration-free inverse design primarily belong to three categories: one-to-one mapping from target to design (Section [4.3.1\)](#page-21-0), cascaded neural networks (Section [4.3.2\)](#page-21-0), and conditional generative models (Section [4.3.3\)](#page-23-0). We will also introduce a few other works that do not fall into these three categories (Section [4.3.4\)](#page-23-0).

Figure 10. Designs of permittivity obtained by a) the hPINNs method, b) PDE-constrained adjoint-based optimization with the finite-difference frequency-domain (FDFD) method as the numerical PDE solvers, and c) PDE-constrained adjoint-based optimization with the FEM as the numeri-cal PDE solvers. The hPINNs method achieved a simpler and smoother solution.^{[\[230\]](#page-41-0)} Reproduced with permission.^[230] Copyright 2021, Society for Industrial and Applied Mathematics.

4.3.1. One-to-One Direct Mapping from Target to Design

Owing to neural networks' capability of approximating any continuous functions, it is possible and straightforward to learn a direct mapping from target quantities of interest (e.g., properties or responses) to design solutions using neural networks. Past work has used MLPs and CNNs to learn such mappings — for example, mapping TO settings (i.e., filter radius, volume fraction, and the type of design objective) to corresponding 2D density maps of mechanical metamaterials,[\[232\]](#page-42-0) target local sound fields to 1D acoustic metasurface designs,[\[233\]](#page-42-0) sets of target scattering parameters to optical metasurface patterns,[\[68\]](#page-39-0) transmission spectrum to photonic nanostructure geometry,^{[\[83\]](#page-39-0)} and target band gaps to phononic crystal designs.[\[43\]](#page-39-0) To train the neural network models, most of these studies used the difference between the predicted and the "ground-truth" design solutions as the training loss, quantified by metrics such as the mean squared error (MSE), mean absolute error (MAE), or binary cross entropy. This poses a problem for the faithfulness of the predicted solutions in terms of meeting the target, because even structurally similar solutions can result in very different quantities of interest (**Figure 11**a), especially when the structures are in pixelated or voxelated repre-sentations, as also illustrated in Woldseth et al.^{[\[48\]](#page-39-0)}

There are works that may avoid this issue by comparing the target quantities rather than the design solutions (i.e., measuring e_2 instead of e_1 in Figure 11b). Malkiel et al.^{[\[83\]](#page-39-0)} first trained an inverse design network to predict the photonic nanostructure geometry based on the transmission spectrum, and then fine-tuned the inverse network by combining it with a forward response pre-diction network. Liu et al.^{[\[234\]](#page-42-0)} proposed a conditional GAN-based model to map a target holographic image to a corresponding optical metasurface design. The physical operation mechanism between the electric-field distribution and the metasurface was used to reconstruct the target image. Both the MSE and an adversarial loss between the original and the reconstructed target images were employed during training. Jiang et al.^{[\[203\]](#page-41-0)} used a conditional GAN to map the target dispersion curves to the structural design of elastic metamaterials. A CNN-based dispersion relation prediction model was used for the fast screening of generated designs based on the predicted dispersion curves. Instead of measuring "how well the predicted design meets the ground-truth solution", these works focused on "how well the predicted design meets the target quantities", thus can lead to the prediction of designs that better match the target quantities.

 x

4.3.2. Avoiding Nonuniqueness Issue via Cascaded Neural Networks

Despite the simplicity of learning direct target-to-design mapping for inverse design, the underlying assumption of one-to-one mapping from target to design can be problematic and may lead to convergence issues during ML model training. Because the nonunique solutions will produce conflicting training instances where the same input is associated with different outputs.[\[39\]](#page-39-0) Taking mechanical metamaterials as an example, this nonuniqueness is either owing to the fact that multiple equivalent structures exist under periodic boundary conditions (top of **Figure [12](#page-22-0)**), or because there are parts of the structure that do not contribute to the properties (bottom of Figure [12\)](#page-22-0). Liu et al.^{[\[39\]](#page-39-0)} and An et al.^{[\[91\]](#page-39-0)} also showed a similar phenomenon for 1D nanophotonic structures and 2D optical metasurfaces, respectively. To overcome this nonuniqueness issue, past work proposed the tandem neural network (T-NN) that cascades an inverse-design network with a forward-modeling network (**Figure [13](#page-22-0)**).[\[39,90,92,235–237\]](#page-39-0) The model training is separated into two phases: 1) training of the forward-modeling network, where each design corresponds to a unique property or response, and 2) training of the cascaded network by fixing the pretrained

Figure 11. The issue of measuring the error in design solutions: a) Structurally similar design solutions with very different Poisson's ratios. There is 0.16% difference in the pixelated design but 100% relative difference between their homogenized Poisson's ratios. b) The small error *e*¹ between the predicted design and the true solution compared to the large error *e*² between the resulting Poisson's ratio and the target Poisson's ratio.

Figure 12. Multiple mechanical metamaterial designs correspond to the same properties. This nonuniqueness is either owing to the fact that multiple equivalent structures exist under periodic boundary conditions (top), or because there are parts of the structure that do not contribute to the properties (bottom).

forward-modeling network, where the design produced at the intermediate layer does not necessarily belong to training data so that the model is not trained with conflicting designs.

Besides the work using T-NN, there were other model variants with a similar idea of using cascaded neural networks to solve the nonunique mapping problem. For instance, Ma et al. $[41]$ combined two bidirectional neural networks (each of which resembles a T-NN) to learn the relation between optical metamaterial design parameters, reflection spectra, and circular dichroism (CD) spectra, aiming for on-demand inverse design of chiral metamaterials given either the full reflection spectra or the CD spectra.

All the surveyed studies using the aforementioned cascade neural networks were applied to optical metamaterial design with dimensions of design variables not higher than 25. With higher design dimensions, it is more likely that the designs produced in the intermediate layer easily fall out of the training data distribution. When this happens, the forward modeling network is not reliable anymore since it has not seen such out-of-distribution designs. Thus, the cascaded network can still have low error while producing designs that are irrelevant to the target. Besides, given one target, cascade neural networks can only predict one design solution, although there are multiple potential solutions. These limitations motivate the use of generative models (Section [4.3.3\)](#page-23-0).

Figure 13. Tandem neural network.^{[39,90,92,235-237] The model training is separated into two phases: 1) training of the forward-modeling network, where} each design corresponds to a unique property or response, and 2) training of the cascaded network by fixing the pretrained forward-modeling network, where the design produced at the intermediate layer does not necessarily belong to training data so that the model is not trained with conflicting designs.

4.3.3. One-to-Many Mapping via Conditional Generative Models

Conditional generative models' ability to learn a distribution of designs conditioned on any target quantities of interest makes them the perfect candidates for learning one-to-many mappings in inverse design applications. The conditional generative models also explicitly model the relationship between the target and the designs and thus will not produce designs irrelevant to the target. Most prior works in this direction used conditional GANs[\[238\]](#page-42-0) and conditional VAEs.[\[239\]](#page-42-0)

Conditional GANs are the primary model for achieving oneto-many mapping in past inverse metamaterial design studies. The original conditional GAN relies on the adversarial loss to ensure the generated designs possess properties or produce responses that match the given target, or show optimality under the given condition.^{[\[104,240\]](#page-40-0)} However, with the purpose of reducing the distance between two distributions, the adversarial loss alone cannot promote high-accuracy matching between an individually generated design and the corresponding target or condition. To overcome this issue, prior works mainly take three measures: 1) using a separate prediction network for fast screening of unqualified generated metamaterials design, $[91]$ 2) adding a prediction loss to implicitly maximize the property/response accuracy of generated designs,[\[42,63,241\]](#page-39-0) and 3) progressively updating the training data by adding high-performance generated metamaterial designs and removing low-performance designs.[\[242\]](#page-42-0)

Conditional VAEs were also employed to achieve the same purpose.[\[75,76\]](#page-39-0) While GANs have shown superior performance on approximating high-dimensional and complicated data distributions, VAEs have the advantage of stable training and are able to extract an interpretable latent space from data. Ma et al.^{[\[75,76\]](#page-39-0)} showed that the latent space from the conditional VAE automatically learns to distinguish metamaterial geometries from different classes.

Note that although some works mentioned in Section [4.3.1](#page-21-0) used conditional GANs to generate designs based on target properties, the mapping between target properties and designs is still one-to-one.[\[203,234\]](#page-41-0) Because these works used a generator without random noise as its input, the generator can only produce a unique design given a fixed target.

4.3.4. Other Approaches

In addition to the three main approaches mentioned above, there are other prior works aiming to achieve iteration-free inverse de-sign. Luo et al.^{[\[243\]](#page-42-0)} proposed a probability-density-based neural network that predicts the acoustic metastructure design in the form of Gaussian mixture model parameters given the target transmission spectrum. By sampling from the Gaussian mixture, this approach can generate one-to-many mappings between target responses and design solutions. However, it might not work on high-dimensional design problems (e.g., topological design problems) due to the need to model more complex distributions. Elzouka et al.[\[219\]](#page-41-0) proposed to use the decision tree to solve both the forward prediction and inverse design problem. After training a decision tree for forward prediction, one can trace up the tree branches from the target response (at leaf nodes) through all branch-splitting criteria. These criteria can be used as design rules to select designs satisfying the given target. This approach naturally captures the one-to-many mapping behavior of inverse design problems, and the extracted rules are interpretable. However, this approach does not suit high-dimensional design problems either, due to the high computational cost of training deci-sion trees with a large depth. Gu et al.^{[\[38\]](#page-39-0)} trained a linear model to classify metamaterial geometries into "good" and "bad" designs based on their toughness and strength. The weights of the linear model indicate how each element in the design contributes to the performances (**Figure [14](#page-24-0)**), based on which new high-performing designs can be sampled. Since the continuous property prediction task was simplified to binary classification, a linear model is sufficient to achieve high predictive accuracy while having the ex-plainability to guide the generation of new designs. Mao et al.^{[\[244\]](#page-42-0)} trained a GAN to generate metamaterials with extreme properties such as the Hashin-Shtrikman upper bounds on isotropic elasticity. While Refs. [\[38\]](#page-39-0) and [\[244\]](#page-42-0) enabled inverse design with high design freedom, they both used training datasets on the order of millions of evaluated designs, which can lead to extremely high data collection costs. Solving high-dimensional design problems with small amounts of labeled data is an open research topic and is worth studying in engineering domains, where labeling usually requires expensive simulations or experiments.

4.4. Discussion and Future Opportunities

Based on Figure [8,](#page-18-0) the most frequently used ML models for metamaterials design are DGMs, MLPs, and CNNs, all of which are based on neural networks. Also, almost all ML models for iteration-free inverse design are based on neural networks. The flexibility and scalability of neural networks give them the versatility to address various types of problems with high complexity and ill-posedness (e.g., inverse design). On the other hand, the requirement of large datasets and low interpretability have limited the performance and applications of neural networks. To illustrate and summarize the advantages and disadvantages of different methods, in this section, we compare existing works in terms of their cost-benefit and trustworthiness.

4.4.1. Cost-Benefit

The cost of using ML models mainly exists in three stages: data collection, training, and inference. The inference stage is normally much cheaper than the other two stages and hence its cost can be negligible. In most data-driven metamaterials design works, data collection includes expensive physics-based simulations. Thus, the data collection cost usually contributes to most of the total cost and highly depends on the size of the training dataset (with the exception of semi-supervised or unsupervised learning). The training cost also depends on training size, among other factors such as model complexity and the number of training epochs. Therefore, we focus on training size when analyzing the cost.

The usefulness of ML models also depends on the benefits they can provide. Since ML usually aims at accelerating the design process, we can use the reduction of computational cost as a criterion to evaluate benefits (i.e., the computational cost difference between conventional methods and data-driven methods

Figure 14. Weights outputted from the linear model show how much each element contributes to a) toughness and b) strength.[\[38\]](#page-39-0) Colors represent the weight of each element: blue represents negative weights and red represents positive weights. Numbers on the elements represent the ranks in terms of weight. Reproduced with permission.^[38] Copyright 2018, Elsevier Ltd.

at inference). However, many prior studies did not include such information. Another important factor that reflects the benefits of ML models is the complexity of the design problem they can address. The complexity highly depends on the dimensionality of the design space. Therefore, we consider the design dimension as an important factor when evaluating benefits.

Figure 15 shows the ML models employed in each prior work, in relation to the training data size and the design dimension used for demonstration. Note that the training size and design dimension are extracted from the experimental settings in prior works and do not necessarily indicate any strict model requirements. This figure shows that when the design problem has a dimension of less than 200, MLPs are the most commonly used ML model, usually with the existence of relatively large training datasets. An exception is Zhelyeznyakov et al., $[159]$ $[159]$ where the accelerated design of high-dimensional dielectric metasurfaces was

Figure 15. Methods proposed in prior work, in relation to the training data size and the design dimension used for demonstration. The publications considered here are the articles shown in Figure [8,](#page-18-0) excluding those without the information of training data sizes or design dimensions. Please see Section S1, Supporting Information for the complete list of publications used to plot this figure. Note that the training size of RL represents the number of design evaluations performed during training. (LR: logistic regression; DR: dimensionality reduction; RL: reinforcement learning; PCR: principal component regression; DT: decision tree; DGM: deep generative model; GP: Gaussian process; CNN: convolutional neural network; MLP: multi-layer perceptron.)

achieved by using an MLP and the data of only 10 designs. In that work, instead of treating each entire design as a training sample, local patches of the design geometry and the electromagnetic field were used. The trained MLP can then be applied to predict the EM field of the entire metasurface. Other models including GP, RL, and DT were also applied in low-dimensional cases. By combining with dimensionality reduction methods such as autoencoders[\[43,68\]](#page-39-0) and VAEs,[\[213\]](#page-41-0) both MLP and GP show the capability of addressing problems in much higher dimensions. Prior works using CNNs were applied to solving problems with a wide spectrum of dimensions, with training sizes ranging from 1000 to 200 000.

For problems with over 1000 dimensions, DGMs were the most commonly used models. With the capability of representation learning and modeling one-to-many mapping, DGMs are applicable to both iterative design optimization and iteration-free inverse design (Figure [8\)](#page-18-0). When used for representation learning, DGMs can be trained with unlabeled data, such that a more compact design representation is learned from only geometric data,^{[\[70,222,223\]](#page-39-0)} which avoids time-consuming simulations when preparing training data and produces reusable design representations for problems with different properties or responses of interest. When using DGMs for iteration-free inverse design, labeled data are normally required for the DGMs to learn the mapping from properties or responses to metamaterial designs. As labeled data are usually acquired by expensive simulations or experiments, we need a careful cost-benefit analysis to determine whether the acceleration brought by a DGM is worth the cost of data collection and training (as in all supervised learning tasks). There are exceptions where the requirement was re-duced to partially labeled or even unlabeled data. Ma et al.^{[\[75,76\]](#page-39-0)} proposed a framework that can use both labeled and unlabeled data for data-driven inverse design of optical metasurfaces, where adding unlabeled data was shown to improve model performance. For the same purpose of optical metasurface inverse de-sign, Liu et al.^{[\[234\]](#page-42-0)} incorporated the physics-based operation between the electric-field distribution and the metasurface design into the decoder of the conditional generative model, which eliminates the need for design evaluation when preparing training data.

By infusing physics into ML models, we can even eliminate the need for training data. As discussed in Section [4.2.4,](#page-20-0) Jiang and Fan^{[\[228,229\]](#page-41-0)} proposed a generative model whose training was guided by the gradient from the adjoint simulation, so that highperformance dielectric metasurface designs can be generated without using training data. Lu et al.^{[\[230\]](#page-41-0)} proposed hPINNs that can optimize a design objective under constraints of governing PDEs^{[\[245\]](#page-42-0)}. However, as mentioned in Section [4.2.4,](#page-20-0) while these physics-informed methods can have zero data collection cost, their training can be unstable due to highly non-convex loss functions[\[231\]](#page-42-0) and hence may bring extra costs (e.g., from exhaustive hyperparameter tuning).

Overall, compared to iterative design optimization, iterationfree inverse design trades off accuracy (i.e., how well the solution matches the true target) for time. Nonetheless, to improve accuracy while still keeping a low computation time, we can use inverse design methods to generate near-optimal solutions as warm starts and further refine the solutions by using optimiza-tion with a relatively small number of iterations.^{[\[240\]](#page-42-0)}

Besides design dimensionality, we need to consider another important factor in evaluating the benefits of machine-learning methods — whether the trained model is applicable to sufficiently many scenarios. To quantify this generality, Woldseth et al.[\[48\]](#page-39-0) proposed a generality score for neural network-based methods used in TO, which accounted for the required similarity of test and training problems (i.e., higher similarity indicates lower generality), in addition to other TO-related criteria. This test-training similarity is also transferable to measuring generality in ML-based metamaterial design. Among the reviewed past works, the majority require the training and test problems to be similar (i.e., having the same design representation and the same properties or responses of interest). One exception is the works on representation learning, where the learned representation can be applied to design problems with different properties or re-sponses of interest. Another exception is Zhelyeznyakov et al.,^{[\[159\]](#page-40-0)} where the design dimension of test problems can vary since the ML model only cares about the local patches of the design geometry.

Note that although the aforementioned physics-based models have the benefit of requiring no training data, the fact that they need fully specified problem settings (e.g., design constraints, boundary conditions, and operating conditions) for training makes the trained models difficult to generalize beyond the problem specification considered during training, i.e., we need to retrain the model for different problem settings.

4.4.2. Trustworthiness

The trustworthiness of ML-based design methods is important in many engineering problems, especially safety-critical and risk-sensitive ones such as metamaterials for blood vessel stents^{[\[246\]](#page-42-0)} and medical imaging.^{[\[43\]](#page-39-0)} One important aspect of trustworthiness is quantifying the uncertainty of metamaterial designs to obtain robust or reliable solutions. This uncertainty quantification, however, is understudied in past literature. Uncertainty comes from sources including operating conditions and the fabrication process. Machine learning models can also have uncertainty due to insufficient data. Data-driven design methods considering these uncertainties can make more informed decisions and generate more trustworthy solutions.

Particularly, due to fabrication uncertainties, the properties or responses of as-fabricated metamaterials can largely deviate from the as-designed ones. **Figure [16](#page-26-0)**a shows an example of geometric deviation of fabricated metasurface patterns. Figure [16b](#page-26-0) shows how geometric deviation can lead to response changes. The nominal design is represented as 64×64 binary pixelated images where 1 (yellow) represents material and 0 (dark blue) represents void. The perturbed design is obtained by slightly distorting the nominal design, which mimics the fabrication error. The figure shows that the absorbance spectrum changes significantly due to this small perturbation, indicating the necessity of quantifying fabrication uncertainty. Due to the high dimensionality of variables to be considered in geometric uncertainty quantification, previous metamaterials design work assumes uniform boundary variation where the boundary of the geometry is uniformly "eroded" (e.g., over-etched) or "dilated" (e.g., under-etched).^{[\[247\]](#page-42-0)} To avoid making simplifying assumptions on

Figure 16. Fabrication uncertainty and its effects on design performance. a) Examples of metasurface patterns fabricated through the electron-beam lithography,^{[\[248\]](#page-42-0)} where the nominal design is a nanocylinder with the circular cross-section (source: Balogun Research Group at Northwestern University). b) Effects of geometric perturbation on the absorbance profile of a metasurface, where the nominal design is the optimal solution of the deterministic optimization from Chen et al.[\[222\]](#page-41-0)

the form of uncertainty and preserve the high degrees of free-dom of geometric uncertainty, Chen et al.^{[\[222\]](#page-41-0)} proposed a DGM with hierarchical latent spaces to simultaneously model the geometric variation of nominal designs and the freeform uncertainties of fabricated designs. Chen et al. incorporated this generative model in robust design optimization and demonstrated notable improvement in as-fabricated design performances compared to only considering uniform uncertainty.

Another key aspect of trustworthiness is interpretability, where humans can understand the reasoning behind an ML model's prediction or decision. Past metamaterial design works have ei-ther used inherently interpretable models like linear models,^{[\[38\]](#page-39-0)} decision trees,^{[\[102,219\]](#page-40-0)} or physics-assisted models that infuse gov-erning PDEs^{[\[230\]](#page-41-0)} and physics-based solvers^{[\[228,229\]](#page-41-0)} into neural networks. These methods improve interpretability in different ways: inherently interpretable models such as decision trees and generalized linear models allow us to extract design rules or investigate the importance of design variables; while physics-assisted models use physics rules to constrain the search of model parameters and solutions during model training. Existing studies using inherently interpretable models (i.e., decision trees) only considered very simple (low-dimensional) designs due to the prohibitive computational cost when using these models for high-dimensional inputs. The application and development of inherently interpretable models (e.g., neural network-based models^{[\[249,250\]](#page-42-0)}) for complex, high-dimensional metamaterial designs were under-explored.

4.4.3. Novelty

Another under-studied challenge in data-driven metamaterials design is how to create novel designs beyond just interpolating existing ones to achieve unprecedented properties or responses. Reinforcement learning can discover novel solutions, but past works only used RL to address low-dimensional design problems, due to the computational cost issue (Section [4.2.3\)](#page-19-0). Data-driven methods using classic ML models are built on the *i.i.d*. assumption that training and testing data are independent and identically distributed. To extrapolate outside existing designs, the training data needs to be updated with desirable generated designs to shift the training data distribution, and the model needs to be retrained on the updated training data. One usually needs to repeat this process many times to obtain a significant improvement over originally existing designs.^{[\[37\]](#page-39-0)} Without retraining, data-driven models cannot learn useful information that generalizes to scenarios outside existing training data distribution, and therefore cannot lead to novel solutions beyond data distribution. More sophisticated methods are needed to distill generalizable information and create novel de-signs. Chen et al.^{[\[36\]](#page-39-0)} extracted parameterized templates of five 3D auxetic metamaterial families from data. The templates can then be used to generate new designs beyond training data, although these new designs have the same topologies as the five families. Future research may explore new methods to generate designs that break more limitations prescribed by existing data.

5. Data-Driven Multiscale Metamaterial System Design

Traditional structural design methods and their data-driven counterparts often focus on homogeneous material distributions. In contrast, some functional engineering structures require heterogeneous property distributions to meet spatially varying requirements, which are critical in achieving better performance and more complex functions. For instance, an invisibility cloak requires heterogeneous properties around an object to prevent it from detection with external physical fields.^{[\[9,13,251\]](#page-38-0)} Requirements on heterogeneous properties can also be found in soft robots, where the goal is to achieve local or global target postures.^[1-4] Recently, structural design methods have evolved to optimize both the structure and the distribution of multiple materials for heterogeneous property requirements. TO is the most flexible among these methods, enabling freeform changes to the structure and providing greater design freedom than traditional pa-rameter or shape design methods.^{[\[100,252\]](#page-39-0)} Despite its promise, it is still challenging to fabricate these multi-material structures

Figure 17. Metamaterials and multiscale systems. a–c) Materials exhibit different Poisson's ratios *v* derived from different microstructures, i.e., different transverse displacements given the same pressing loading on the top, with the transparent boxes showing the original shapes before distortion. d) Multiscale orthopedic implant design. Reproduced with permission.[\[254\]](#page-42-0) Copyright 2019, Elsevier B.V. e) Multiscale design for shape morphing under thermal excitations. Reproduced with permission.[\[255\]](#page-42-0) Copyright 2016, Macmillan Publishers. f) An illustration of the multiscale metamaterial system design process.

in achieving as-designed functions. This issue is caused by the narrow selection of available materials and the constraints of manufacturing processes. In contrast, complex geometries can be more easily manufactured at fine resolutions with additive manufacturing.[\[253\]](#page-42-0) This technical revolution has opened up new avenues to realize unprecedented and tailorable material properties by changing the geometry of microstructures rather than constituent materials,[\[5–8\]](#page-38-0) as shown in **Figure 17**a–c. Therefore, heterogeneous properties can be obtained by spatially varying the microstructures, instead of constituent materials, to assemble a multiscale metamaterials system for intricate structural behaviors (Figure $17d,e$).^{[\[254,255\]](#page-42-0)} In this section, we will discuss datadriven methods for designing multiscale metamaterial systems that determine architectures at both micro and macro scales to achieve the desired metamaterial behaviors. Our focus is on the heterogeneous distribution of (effective) mechanical or thermal properties that originate from the lower material scale, as these physics are involved in most existing multiscale metamaterial designs. For designs with other underlying physics, we refer readers to Refs. [\[50, 53, 56–58, 256\]](#page-39-0).

5.1. Overview

Multiscale structures with carefully tuned microstructures have been shown to have the edge over single-scale design (macroscale only with homogeneous materials or periodic unit cells discussed in Section [4\)](#page-16-0) for engineering applications involving multi-

physics or spatially varying requirements. Typical applications in-clude strain cloaking,^{[\[257,258\]](#page-42-0)} target deformation design,^{[\[28,34,35,259\]](#page-38-0)} thermal-elastic property optimization,^[260-262] dynamic behav-iors design,^{[\[22,93,263–265\]](#page-38-0)} buckling resistance,^{[\[266,267\]](#page-42-0)} and energy absorption.[\[268–270\]](#page-42-0) However, the design of multiscale structures is a complex two-scale problem, as shown in Figure 17f. At the macroscale, the topology of the structure and its mechanical properties distribution are optimized to meet performance targets, while at the microscale, unit cells need to be designed at different locations to achieve the corresponding properties. Ideally, the design process for a multiscale metamaterial system should be carried out in a way that the two scales are coupled and designed concurrently. This means that the design of the microscale and macroscale architectures should be done simultaneously and interactively, rather than separately and independently. This hierarchical nature of the system poses unique challenges in the design process compared to the single-scale TO methods illustrated in Section [4.](#page-16-0)

First, at the macroscale, property parameters required to fully describe the physical response of materials are generally highdimensional without strict bounds for the achievable properties. This leads to an ill-defined property design space and a complicated optimization process. As a result, most existing methods are subject to over-simplistic constraints on the design space of properties. Second, at the microscale, the design of microstructures is an inverse problem without much a priori design knowledge (see Section [4\)](#page-16-0). It is characterized by its infinite-dimensional geometrical design space and one-to-many mapping from

Figure 18. Illustration of bottom-up and top-down frameworks. The target displacement profile (red dashed lines) is used as an example of the systemlevel design objective.

properties to structures. This creates an irregular landscape for the design objective (macroscale properties or performance) with many local optima, making the design sensitive to the initial guess and constraints. Finally, the synthesis of micro- and macrodesigns suffers from the nested optimizations at both scales, "the curse of dimensionality" induced by the hierarchical multiscale design space, complex combinatorial search associated with the unit-cell selection, and adjacent microstructures whose shapes are incompatible at their interfaces (geometrical frustration). For instance, a typical formulation for hierarchical topology optimization involves coupling a macroscale optimization prob-lem with a set of micro-scale problems at the element level.^{[\[271,272\]](#page-42-0)} In each iteration, the macroscale problem assesses overall performance, as well as the sensitivity of macroscale properties and design variables, based on micro-scale properties homogenized at the element level. Subsequently, the performance and sensitivity information is transmitted to the micro-scale level to formulate element-wise microscale optimization problems. They aim to evolve the unit-cell geometry to achieve optimal homogenized properties. Consequently, the number of design variables in this scenario corresponds to the product of the design resolutions at both macro- and microscale levels. The hierarchical nature makes the microscale problems independent from each other while certain local expansion methods^{[\[273\]](#page-42-0)} can decouple the two scales. They both can alleviate the computational cost to some extent by allowing parallel computing. Nonetheless, the independent macroscale designs and separated scales can lead to geometric discrepancies and discontinuous load-bearing paths between cells. Such issues can undermine the manufacturability and overall performance of the multiscale structure. Due to these issues, traditional multiscale topology optimization methods are either overwhelmingly time-consuming or restrictive in design flexibility to achieve heterogeneous multiscale metamaterial systems.

Capitalizing on the growth of data resources and computational capability, data-driven design based on ML models is recognized as a promising tool to address the aforementioned challenges for multiscale systems. We underscore that data-driven multiscale designs do not supplant the use of traditional structural optimization methods within the design framework; rather, they enrich these methods by substituting specific components (such as design representation, homogenization, and nested microscale designs) with gathered data or machine learning models. The effectiveness of the data-driven approach hinges on combining rigorous and matured optimization techniques with the capabilities of data science. As depicted in **Figure 18**, we propose to divide existing data-driven multiscale design frameworks into two main categories, i.e., bottom-up and top-down frameworks, based on the relations between design variables at the two scales. The bottom-up framework directly uses the parameters at the microscale level, e.g., volume fraction and unit cell type, as design variables. Costly nested calculation of effective properties of the unit cells is replaced by a surrogate model of the structure-property mapping. In contrast, with the topdown framework, the macroscale topology and spatial distribution of homogenized material properties are concurrently optimized first. Then, to assemble a full multiscale structure, the optimized properties will serve as targets to retrieve the corresponding building blocks. To accelerate this assembly process, ML models trained at the microscale level (Section [4\)](#page-16-0) can be used to compactly represent and/or efficiently generate unit cells.

The pioneering works in data-driven multiscale metamaterial design initially emerged within the bottom-up framework, which has fewer demands on data volume, unit-cell geometry complexity, and the versatility of ML models. In fact, the foundational work by Bendsøe and Kikuchi in 1988,^{[\[274\]](#page-42-0)} which laid the ground for TO, conceived structural design as the optimization of spatially distributed infinitesimal square cells. A material model was employed to establish the connection between geometric parameters and homogenized properties. Although the data-driven model is not the focus in this homogenized method, it can be regarded as the precursor to the bottom-up framework. However, due to limitations in computational power and manufacturing **www.advancedsciencenews.com www.advmat.de**

CIENCE NEWS

Figure 19. Three bottom-up frameworks. a) Property space of single-class unit cell. b) Property space of oriented unit cell. The shaded regions are for unit cells with non-zero oriented angles. c) Property space of a diversified set of unit cells. The property curves are colored in accordance with the colors of unit cell classes. Corresponding multiscale designs with d) single-class unit cells, e) oriented unit cells, f) diversified unit cells.

capabilities of that era, subsequent endeavors within the TO community leaned toward single-scale design.[\[275–277\]](#page-42-0) These efforts modified the homogenized material model to encourage binary material distribution at the macroscale, avoiding macroelements with intermediate density corresponding to different microscale designs.[\[278–280\]](#page-42-0) Advancements in computational and manufacturing capacities eventually rekindled interest in the multiscale design and bottom-up framework. Bottom-up methods integrate diverse unit-cell parametric models and ML methods, reintroducing the microscale dimension into TO. Intriguingly, while early TO works shifted from intricate homogenized models to simpler single-scale designs, recent years have witnessed the emergence of data-driven bottom-up methods aimed at augmenting unit-cell design complexity and, consequently, design flexibility. This progression has led to divergent developments, such as unit-cell reorientation and the diversification of their topologies. As data-driven models, particularly deep learning models, continue to mature, the pursuit of heightened design flexibility has propelled the ascent of the top-down framework. This framework accommodates an even broader array of unit-cell designs, including those that are highly diverse and free-form in topology.

In the remainder of this section, we will illustrate and review the state-of-the-art of these two frameworks. This section is structured in a manner that closely follows the chronological progression, tracing the evolution toward expanding the unit-cell design space. Our approach begins with the foundational bottom-up framework, followed by the introduction of enhanced bottom-up methods that incorporate the design of unit-cell orientations and diverse configurations. Concluding this trajectory, we delve into a review of top-down methods that offer the utmost design flexibility.

5.2. Bottom-Up Framework

In this section, we will focus on the most commonly used data-driven multiscale design framework, i.e., bottom-up framework. We will start from the simplest bottom-up framework with only single-class unit cells to explain each basic component. Then we will review different modified variants that aim to increase the unit-cell diversity and complexity for higher design flexibility.

5.2.1. Data-Driven Design with Single-Class Graded Unit Cells

Most bottom-up data-driven methods assumed the same topological concept (single-class) for all the microstructures and only spatially vary, i.e., grade, the geometrical parameters as shown in **Figure 19**a,d. Lattice-based^{[\[281\]](#page-42-0)} and surface-based microstructures were commonly used due to their simplicity and good manufacturability. One could change the thickness of rods or surfaces^{[\[282\]](#page-42-0)} to obtain different geometries, with each corresponding to a specific volume fraction. The thickness or the volume fraction can be used as design variables in the optimization process. An advantage of this graded single-class assumption is that one could leave out the microscale details during the optimization and directly optimize the spatial distribution of geometrical parameters at the macroscale instead. This is also called homogenization-based design. The data-driven aspect of this framework is that the time-consuming homogenized properties evaluation in each iteration was replaced by a surrogate model, capturing the relation between geometrical parameters and the precomputed properties. Some examples of such models are exponential function,^{[\[283\]](#page-42-0)} polynomial,^{[\[281,284\]](#page-42-0)} Kriging,^{[\[282\]](#page-42-0)} diffuse

www.advancedsciencenews.com www.advmat.de

IFNCF NFWS

Figure 20. Examples of oriented two-scale designs. a) Compatible tiling without unit cell orientation design. b) Incompatible tilling with unit cell ori-entation design. c) Oriented design using conformal mapping. Reproduced with permission.^{[\[290\]](#page-42-0)} Copyright 2021, ASME. d) De-homogenized design via cosine-based mapping. Reproduced with permission.^{[\[291\]](#page-42-0)} Copyright 2018, John Wiley & Sons, Ltd. e) De-homogenized design obtained for multi-loading cases using microstructures with three different layers. Reproduced permission.^{[\[292\]](#page-42-0)} f) De-homogenized design via Fourier-series-based map-ping. Reproduced with permission.^{[\[293\]](#page-42-0)} Copyright 2020, Elsevier Ltd. g) De-homogenized design via saw tooth-function-based mapping. Reproduced with permission.[\[93\]](#page-39-0) Copyright 2022, Elsevier B.V.

approximation^{[\[285\]](#page-42-0)} and neural network.^{[\[286\]](#page-42-0)} After optimization, the corresponding multiscale structure can be obtained by filling the macroscale design with the microstructures specified by the optimal parameters, a process known as de-homogenization.[\[287\]](#page-42-0)

Despite its simplicity and high efficiency, single-class datadriven graded design usually leads to sub-optimal solutions since the microstructures belong to the same topological class with fixed unit cell orientations. For example, to optimize compliance for both single-loading and multi-loading cases, microstructures must consist of oriented two and three alternating layers of orthotropic materials, known as rank-2 and rank-3 materials, respectively.[\[288\]](#page-42-0) These microstructures are referred to as rank-2 and rank-3 materials, respectively. Achieving these designs requires spatial changes to the unit cell topology and orientation. Although most existing designs used compliance minimization as the demonstrative case, the single-class graded microstruc-tures did not meet the optimum requirement.^{[\[289\]](#page-42-0)} As a result, the performance of the multiscale design was even worse than the single-scale design. Similar observations were also reported in various applications, such as frequency response control^{[\[93\]](#page-39-0)} and natural frequencies maximization.^{[\[149\]](#page-40-0)} To increase design flexibility, many studies were dedicated to 1) increasing the diversity

of microstructures and/or 2) considering unit cell orientation designs. As shown in Figure [19c,d,f,g,](#page-29-0) these two variants of bottomup methods can both expand the property space of the database. The rest of Section [5.2](#page-29-0) reviews these approaches.

5.2.2. Variant 1: Considering Unit-Cell Orientation

The major obstacle in designing graded structures with oriented microstructures lies in the de-homogenization process. When the unit-cell orientations vary across the macroscale structure, the corresponding microstructures need to be rotated accordingly. However, neighboring microstructures might not connect with each other after rotation (**Figure 20**a,c). This causes the multiscale structure to fail to attain the designed performance and, furthermore, impossible to manufacture. The key to mitigating this issue is to construct a smooth mapping from the stand-alone regular unit cells (e.g., unit cells shown in the upper corners of Figure 20b) to an assembled tiling (e.g., the multiscale structure shown in Figure 20b), which can distort the micro-structure to ensure compatibility but at the same time retain their effective properties.

Conformal mapping is considered a powerful tool to realize this mapping since it can preserve the angle of the geometrical features and thus minimize the variation in effective properties.[\[294\]](#page-42-0) Jiang et al.[\[290\]](#page-42-0) constructed conforming mapping after homogenization-based design to morph a rectangular tiling of periodic microstructures into corresponding irregular regions in the multiscale structure (Figure [20b\)](#page-30-0). However, the design of unit-cell orientations was not considered in this study. Ma et al.[\[295\]](#page-42-0) used a linear combination of a set of basis functions to parameterize the mapping from a predefined unit cell to a multiscale oriented tiling. The coefficients of the basis functions were used as design variables, and used to train an MLP model to predict effective properties from the local Jacobian matrix of the corresponding mapping. While this method allowed the implicit design of unit-cell orientation, the design space was restricted by the form and orders of basis functions, as well as the predefined shapes of unit cells.

In a major advance in 2008, Pantz and Trabelsi^{[\[296\]](#page-42-0)} focused on square microstructures with rectangular holes and proposed a method to project a homogenized design to a multiscale structure with oriented microstructures on a high-resolution mesh. They achieved this de-homogenization process by constructing implicit mapping functions from a pair of cosine fields to approximate the oriented unit cells in different spatial locations. Later, Groen and Sigmund^{[\[291\]](#page-42-0)} simplified this de-homogenization process by introducing the connected component labeling method to obtain a consistent orientation field, and by relaxing the optimization problem for the mapping function (Figure $20d$).^{[\[291\]](#page-42-0)} This simplified method was further extended to enable the efficient design of 3D multiscale structures.[\[297\]](#page-42-0) The de-homogenization process was accelerated by training a neural network to obtain the mapping function from the optimized unit-cell orientations without any extra optimization process.^{[\[298\]](#page-42-0)} Although the dehomogenization method is appealing in terms of efficiency and performance, it is still confined to simple static compliance min-imization problems.^{[\[299,300\]](#page-43-0)} The reason is that its original version can only handle square cells with rectangular holes, simply making the unit-cell orientation align with the principal strain direction. While these designs are optimal for static compliance minimization under a single loading, they would become suboptimal for general design cases, such as multi-loading, dynamic response optimization, and multi-physics problems.

To extend the applicability of de-homogenization, various enhanced methods have been proposed to accommodate more complicated unit cells. Ladegaard et al.^{[\[292\]](#page-42-0)} combined three cosine fields to construct the implicit mapping functions for oriented rank-3 microstructures (Figure [20e\)](#page-30-0). It enables the dehomogenization to find the optimal structures for static compliance problems in multi-loading cases. To handle freeform unit cells, Tamijian et al.^{[\[293\]](#page-42-0)} represented the complex unit-cell geometries as a Fourier series and optimized the spatial distribution of each Fourier basis to orient unit cells in a compat-ible way (Figure [20f\)](#page-30-0). Groen^{[\[287\]](#page-42-0)} suggested using cosine functions to construct a similar implicit mapping from regular unit cell regions into oriented patches. Inspired by the texture map-ping in computer graphics, Kumar et al.^{[\[301\]](#page-43-0)} adopted a finiteelement mesh to parameterize this implicit mapping. The mapping was obtained by solving a set of linear equations associated with the discrete mesh. While these extensions allowed the

use of more complicated unit-cell geometries, their constructed implicit mapping was not conformal and may deteriorate the performance of the multiscale designs. In a more recent work, Wang et al.^{[\[93\]](#page-39-0)} proposed to construct a conformal mapping in the de-homogenization process by using Sawtooth function fields (Figure [20g\)](#page-30-0). Unit cells with mixed-class topologies were then used to broaden the property space, with neural networks as surrogate models.

Overall, this branch of variants enabled the dehomogenization process to obtain high-resolution multiscale structures composed of oriented unit cells with superior efficiency. It has been extended to handle complex unit-cell geometries and is emerging as an effective method for largescale structural designs. However, the works under this category all focused on static compliance minimization problems with a few exceptions. Although a microstructure comprised of multiple solid layers in de-homogenization is optimal for minimizing static compliance, determining the optimal microscale geometry becomes non-trivial and challenging for general design problems with system-level performance objectives. In the context of metamaterial system design, complex functionalities encompassing multiple physics can necessitate distinct geometric configurations and discontinuous variations in unit cells. Examples of these complexities include wave guiding^{[\[21,28\]](#page-38-0)} and energy absorption,^{[\[268,269\]](#page-42-0)} which often cannot be easily achieved through smooth mappings.

5.2.3. Variant 2: Increasing the Diversity of Unit Cells

The second branch of the variants aims to increase design flexibility by considering diverse micro-structure topologies in the optimization. As illustrated earlier, the success of data-driven graded design relies on the low-dimensional descriptor of unit cell geometries. Therefore, the key challenge addressed in this branch of variants is to represent broader sets of unit-cell topologies without significantly increasing the dimension of descriptors.

As shown in Figure [19f](#page-29-0) and **Figure [21](#page-32-0)**, a relatively straightforward idea is to include multiple unit cell classes in the optimization, each with its own low-dimensional parameterization. By considering a unit cell class as a special type of discrete material, this idea naturally fits into the discrete material optimization framework. In this framework, the unit cell class was represented by one-hot encoding and relaxed to be a continuous variable by adding penalization or constraints.[\[302\]](#page-43-0) This enabled the automatic selection of the optimal unit-cell classes for different spatial regions in the optimization. Due to its simplicity, it has become the most commonly used framework to accommodate multiple unit-cell classes in the multiscalse system design.

However, the dimension of design variables (one-hot encoding) for each microstructure grows linearly with the number of classes being considered. This will greatly increase the execution time and complexity. Meanwhile, the one-hot encoding only represents the unit-cell classes in a qualitative way without any physical meaning. As a result, the surrogate models and the optimization process cannot explicitly exploit the correlation or similarity between different classes for better performance. To address these issues, Wang et al.[\[87,95,149\]](#page-39-0) proposed a multi-response latent variable Gaussian process (MR-LVGP) and its enhanced **www.advancedsciencenews.com www.advmat.de**

CIENCE NEWS

Figure 21. Multiscale designs with diversified unit cells. a) Design with multiple unit cell classes. Reproduced with permission.[\[95,149\]](#page-39-0) top) Copyright 2022, ASME. bottom) Copyright 2022, Elsevier Ltd. b) Multiscale design with unit cells that allow a smooth transition between two classes. Reproduced with permission.^{[\[303\]](#page-43-0)} Copyright 2018, Springer-Verlag GmbH Germany. c) Multiscale design with blended unit cells. Reproduced with permission.^{[\[304\]](#page-43-0)} Copyright 2021, Springer-Verlag GmbH Germany. d) Multiscale design with concurrently optimized unit cell prototypes. Reproduced with permission. [\[264\]](#page-42-0) Copyright 2020, Elsevier Ltd. e) Smooth transition of spinodal metamaterials. Reproduced with permission.[\[90\]](#page-39-0) Copyright, 2020, Creative Commons CC BY. f) Multiscale design with spatially varying spinodal microstructures. Reproduced with permission.^{[\[305\]](#page-43-0)} Copyright 2021, Elsevier B.V. under the CC BY license. g) Multiscale design with selected symmetric types of candidate spinodal microstructures. Reproduced with permission.^{[\[306\]](#page-43-0)} Copyright 2022, Wiley-VCH GmbH.

variants to transform the discrete classes into continuous 2D latent variables through statistical inference. The constructed latent space captured the effects of classes on the mechanical properties, which induced an interpretable distance metric that reflects the similarity with respect to properties (Figure 21a). Moreover, with the nonlinear embedding, the dimension of the latent space remained when the number of unit cell classes increased. By integrating the MR-LVGP models with TO, an efficient data-driven optimization process was developed that can concurrently explore multiple classes and/or constituent materials and their associated geometric parameters for better structural performance.

To further increase the diversity in unit-cell geometries, it is desirable to enable the transition or blending between discrete unit-cell classes (Figure 21b–d). In this way, new micro-structure prototypes can be created that go beyond the geometries and properties of given classes. To achieve this, Wang et al.^{[\[303,307\]](#page-43-0)} established a sophisticated parameterization method for selected classes of microstructures formed by multiple groups of rods (Figure [21b\)](#page-32-0). By changing the relative ratio among the rod thicknesses of different groups, a smooth transition between multiple unit-cell topologies can be achieved. However, this parameterization technique is difficult to generalize to other microstruc-tures with freeform topologies. Chan et al.^{[\[65\]](#page-39-0)} proposed a more general shape blending scheme that can accommodate freeform unit cell classes with distinct and even incompatible topologies, generating smoothly graded microstructures (Figure [21c\)](#page-32-0). The interpolation scheme only had a few extra parameters, which can be directly integrated into the data-driven graded design framework with a neural network as the surrogate model for properties. While blending multiple unit classes expands the design space, it requires a set of prespecified unit-cell prototypes. How to obtain an optimal set of prototypes is problem-dependent and generally unknown beforehand. In Chan et al., the sets were selected either using domain knowledge or an autonomous set selection method that maximizes diversity metrics.[\[65\]](#page-39-0) Instead of using pre-specified prototypes, Zhang et al.^{[\[264,308,309\]](#page-42-0)} propose to simultaneously evolve the prototypes during the optimization. The surrogate model to predict the properties of interpolated unit cells, i.e., a GP model, was also updated in each iteration after evolving the prototypes (Figure [21d\)](#page-32-0). By doing so, the initial prototypes can change their shapes to better handle customized design scenarios.

Besides considering multiple classes and their interpolation, special types of materials, i.e., spinodal materials, are emerging as a promising choice in data-driven multiscale design. These stochastic self-assembled materials can easily achieve diverse microscale geometries with inherent connectivity when the volume fraction is above a given theoretical threshold.^{[\[310\]](#page-43-0)} The bottomup framework is also applicable to the design of these stochastic materials, by generalizing the unit cell concept to include RVE (see the concept definition in Section [2\)](#page-2-0). Zheng et al.^{[\[305\]](#page-43-0)} used Gaussian random fields to describe micro-structures of spinodal materials and train a fully connected neural network to associate the field parameters with the effective mechanical properties (Figure [21e,f\)](#page-32-0). By using the field parameters as design variables, a multiscale structure with spatially varying but smoothly graded microstructures can be achieved.^{[\[90\]](#page-39-0)} Senhora et al.^{[\[306\]](#page-43-0)} simplified the design of spinodal materials by focusing on selected symmetric types of candidate spinodal architected materials, extending to accommodate various complex 3D designs (Figure [21\(g\)\)](#page-32-0).

5.3. Top-Down Framework

The bottom-up framework illustrated in the last section depends on a properly parameterized unit cell model with lowdimensional variables. While this constrained design space greatly expedites the design process, it also shrinks the property space, which will fall short for advanced applications such as soft robots^{[\[2\]](#page-38-0)} and mechanical cloaking.^{[\[311\]](#page-43-0)} The top-down design framework, the second branch of multiscale system design, aims to remove the constraints imposed on the microstructure geometries to allow the use of either pre-specified or freeform unit cells in assembling the full structure (**Figure [22](#page-34-0)**).[\[34,312–316\]](#page-39-0) This framework, in its ideal form, can unleash the highest potential of metamaterials. Specifically, a large database of microstructures is first constructed, containing different geometries and precomputed properties (Figure [22a\)](#page-34-0). Since the complex unit cell geometries do not have inherent low-dimensional representations as in the bottom-up framework, the property space of the database will serve as the design space for the property distribution optimization at the macroscale. After that, the property distribution at the macroscale cascades to the microscale (Figure [18\)](#page-28-0) and, based on this, the corresponding microstructures are generated or fetched from the database to fill each element in the full structure. Therefore, there is no need to do the nested optimization and property evaluation at the microscale during the design process, which significantly improves the efficiency during the design of structures. However, to ensure compatibility between adjacent unit cells, these methods still need to force unit cells to be similar in geometries or compatible on the shared boundaries, which limits the range of achievable properties.

To address this issue, Schumacher et al.^{[\[35\]](#page-39-0)} proposed to construct a database with different metamaterial families. Unit cell geometries are similar within each family but distinctly different across families. They recognized that these families cover different regions but would overlap in the property space. The overlapped regions contain candidates with various geometries for the same properties. The best match can then be selected from those candidates for compatible boundaries. This compatible tiling keeps the design performance at the macroscale matches with the evaluation with homogenization. Later, Zhu et al.[\[36,82\]](#page-39-0) discarded the concept of families and generated a larger and richer database by stochastically adding and removing materials from microstructures in the database (Figure [22b\)](#page-34-0). In the assembly stage, a random-search-based method was used to select microstructures with compatible boundaries to realize the target mechanical properties at each point in the macroscale structure.

However, due to the large amount and diverse shapes of microstructures, an immense combinatorial space needs to be explored through stochastic methods to form compatible boundaries between adjacent unit cells. Moreover, while these studies proposed elaborate methods for database construction, they lacked an effective representation and retrieval method for microstructures. As a result, it is challenging to incorporate these large databases into the multiscale design in a scalable way. This might also be the reason that these studies only focused on unit cells with isotropic or cubic symmetry.

In contrast, Bastek et al.^{[\[317\]](#page-43-0)} focused on truss metamaterials with anisotropic stiffness tensor. The unit cells in the database are generated by randomly superimposing and tessellating unique topology types, followed by affine deformations to break the symmetry and enhance anisotropy. Using a modified tandem neural network, they established an inverse mapping from target properties to corresponding geometry parameters. While this study does not involve multiscale design, it introduced an interpolation scheme in topology and reciprocal lattice space (similar to de-homogenization), enabling graded transitions between distinct unit cells (Figure [22c\)](#page-34-0). Although the interpolation scheme can improve the compatibility between microstructures, it requires an extra user-defined transition layer between every pair of unit cells. Additionally, the interpolation in geometries does not guarantee smoothly varying properties across transitions. Wang et al.^{[\[66,257\]](#page-39-0)} also focused on anisotropic metamaterials but with free-formed microstructures (Figure [22d\)](#page-34-0). To tackle the

Figure 22. Top-down methods. a) Design framework using discrete unit cells with cubic symmetry and predefined compatibility. The top row shows two pairs of compatibly connected isotropic unit cells in the database. Reproduced with permission from the authors.^{[\[34\]](#page-39-0)} b) Design framework using randomly generated isotropic unit cells with two constituents. Reproduced with permission.^{[\[82\]](#page-39-0)} Copyright 2017, Association for Computing Machinery. c) Functionally graded structure by interpolating between different anisotropic truss metamaterials.^{[\[317\]](#page-43-0)} d) Design framework using randomly generated anisotropic unit cells. Reproduced with permission.^{[\[66\]](#page-39-0)} Copyright 2020, Elsevier B.V. d) Multiscale design with stochastic growth rules. Reproduced with permission.[\[88\]](#page-39-0) Copyright 2022, American Association for the Advancement of Science.

aforementioned challenges associated with the multiscale design, they simultaneously trained a VAE and an NN-based property predictor to map complex microstructures into a lowdimensional, continuous, and organized latent space. They found that the latent space of VAE provided a distance metric to measure shape similarity, enabled interpolation between microstructures, and encoded meaningful patterns of variation

in geometries and properties. These characteristics enabled an effective selection of diverse unit cell candidate sets from the database to increase the chance of compatible assembly. The shape similarity metric was also utilized as a metric for geometrical compatibility between unit cells. By combining both geometrical and mechanical compatibility measures, the assembly process was formulated as an energy-minimization problem

on a grid-like graph and solved efficiently by dual decomposition. This method has been successfully applied to achieve target deformation profiles, $[66]$ mechanical cloaking, $[257]$ and frac-ture resistance.^{[\[318\]](#page-43-0)} Following a similar direction, Wang et al.^{[\[63\]](#page-39-0)} used a GAN model to learn the distribution of implicit-surfacebased geometries conditioned on given properties. By utilizing the continuity of the GAN-generated structures and a transition layer blending technique, compatible microstructures are inversely generated to achieve the designed properties in the as-sembled structure. Liu et al.^{[\[88\]](#page-39-0)} adopt a different strategy by devis-ing a stochastic growth rule, similar to a cellular automaton,^{[\[319\]](#page-43-0)} to blend different graph-based features into irregular microstructures (Figure [22d\)](#page-34-0). Good compatibility can be ensured by devising special local growth rules. With the relation between the homogenized properties and the growth rule, the full structure can achieve the target property distribution through an automatic growth process.

5.4. Discussion

Both data-driven multiscale frameworks have shown promise in achieving efficient multiscale design in various applications, but they still have limitations and are not yet ideal. In this section, we will compare these frameworks to illustrate their respective strengths and weaknesses, providing insights into their applicability. We will also highlight critical knowledge gaps and challenges in existing research.

5.4.1. Comparison of Bottom-Up and Top-Down Frameworks

- a) **Efficiency** The two types of frameworks both ignore the microscale details in multiscale optimization by using the unit cell parameters (bottom-up) or effective properties (top-down) as the design variables. This allows them to bypass nested optimization across different scales and numerous homogenization evaluations, enabling much higher computational efficiency than traditional multiscale designs. Among them, the bottom-up framework is generally more efficient than its topdown counterpart because the compatibility between neighboring unit cells is guaranteed by the parameterized unit cell or easily handled by adding local constraints. In the topdown framework, when diverse microstructures are considered, an extra tiling optimization is usually needed after the homogenization-based optimization to guarantee compatibility, which is relatively time-consuming.
- b) **Design Freedom** The bottom-up framework uses microscale parameters as the design variables and thus requires a parameterized model for unit cells. This restricts the change of unit cell topologies. In contrast, the top-down framework directly optimizes the effective properties, which can be adapted to any unit-cell geometries. Therefore, the top-down framework has higher design freedom than its bottom-up counterpart. The flexibility of the bottom-up framework can be improved by considering multiple classes of microstructures or the unit cell orientation, as previously illustrated. However, this will sacrifice some efficiency and may lead to a complex optimization problem with more local optima.
- c) **Manufacturability** The parameterization of unit cells in the bottom-up framework makes it easier to impose functionally graded constraints or filters on the geometries, which can benefit the manufacturability. The manufacturing restriction can also be considered in selecting the unit cell classes. In the top-down framework, manufacturing constraints of a single unit cell can be added to the construction of the database. However, the manufacturability of the assembled structure cannot be explicitly considered when designing the property distribution. Additional steps of compatibility optimization and post-processing are required in the assembly stage to obtain manufacturable full structures.[\[36,66,82\]](#page-39-0)
- d) **Generalizability** The top-down method assumes weak mechanical coupling between unit cells and has been confined to material designs in the realm of linear elasticity. For dynamic applications or nonlinear cases, this coupling might not be negligible.[\[21,23,320–322\]](#page-38-0) Since the unit cells in the bottom-up framework are parameterized, it is easier to consider the coupling between unit cells by modeling the interaction as a function of their parameters. The bottom-up framework can also be extended to consider strain-dependent properties in accommodating nonlinear cases.[\[286\]](#page-42-0)

5.4.2. Assumptions of Homogenization

The first-order homogenization method is the basis of most datadriven multiscale designs to obtain the effective properties of unit cells in the database.^{[\[60\]](#page-39-0)} It assumes that the stress of each point in the macroscale only depends on its local strain value and is not affected by neighboring unit cells.^{[\[323\]](#page-43-0)} This is only valid when the microscopic length scale is much smaller than that of the macrostructure, and when the microstructures are periodically distributed in deterministic cases or stationally ergodic in stochastic materials.[\[60\]](#page-39-0) For example, in a periodic design for compliance minimization, the ratio between the sizes of macro- and micro-structures should be 5–6 to ensure a relatively accurate evaluation of the performance with homogenization.^{[\[324\]](#page-43-0)} Most existing data-driven multiscale designs do not meet the first-order homogenization assumption due to the aperiodic tiling and the large unit cell size restricted by manufacturability. However, if neighboring unit cells have a smooth change or compatible connections within a large entire structure, homogenization can still provide relatively satisfying results as reported in some studies.[\[35,65\]](#page-39-0) Nevertheless, it is still advisable to present the performance metrics obtained via full-scale simulation as a validation, which was rarely reported in existing papers.

Meanwhile, reduced-order models combined with ML are becoming promising alternatives to replace homogenization in the multiscale deign which do not rely on first-order homogenization assumption and can thus simulate the response more accurately. For example, Wu and Fu et al.^{[\[285,325\]](#page-42-0)} condensed the fine-meshed finite element (FE) model of unit cells into a super element with only boundary nodes. Since no periodicity or scale separation is assumed, the macro-response remains accurate for heterogeneous design even with the size of the unit cell close to the macrostructure. By combining proper orthogonal decomposition and diffusion approximation methods, the relation

IDVANCED CIENCE NEWS www.advancedsciencenews.com www.advmat.de

between volume fraction and the condensed stiffness matrix can be directly obtained for the previous graded structural design framework.

Physics-based machine learning (PBML) model is emerging as another intriguing direction to accelerate the full-scale simulation without resorting to homogenization (see section [4.2.4\)](#page-20-0). This can be a potentially useful tool to bypass the first-order homogenization assumption. Currently, PBML mainly focuses on for-ward analysis instead of inverse optimization. Yao et al.^{[\[326\]](#page-43-0)} considered the finite element model as a special type of convolution layer to construct FEA-Net, predicting the mechanical response of metamaterials under external loading. Following similar ideas, Saha et al.^{[\[327\]](#page-43-0)} used neural networks to replace the interpolation functions in FEA models, which can provide fine-resolution results with lower computational expenses. While these physicsinformed methods require fewer data, they are either less efficient or restricted to a single type of microstructure, compared to the aforementioned fully data-driven methods. Although many existing studies claim that PBML is more efficient and easy to use than traditional FEA, most of these models can still be considered as special FE models or PDE solvers. They used a neural network to replace those classical approximation functions, which transforms the original linear weighted-form equation into a highly non-linear optimization problem. Stochastic gradient descents and their variances were then used to solve the problem, which does not guarantee convergence to the real solution. The universal applications and higher efficiency were obtained at the cost of rigorous theoretical foundations. Also, most existing studies compare PBML with naive FE models, instead of some more ad-vanced FEM, which might not be a fair comparison.^{[\[49,328,329\]](#page-39-0)} It is advisable for the researchers to explore both PBML and advanced FEM in accelerating the data-driven designs for multiscale systems.

5.4.3. Task Specificity of Data Acquisition for Multiscale Design

In Task-Related Metrics under Section [3.4.1,](#page-12-0) we have briefly discussed the task specificity of data assessment within metric-level assessment. Enlarging the scope to the system-level design optimization, we reiterate the task specificity based on some concrete results reproduced from the literature. **Figure [23](#page-37-0)** shows a set of target tasks at the system level defined by Wang et al.^{[\[66\]](#page-39-0)} Assuming linear elasticity deformation in 2D mechanical metamaterials, three design tasks are prepared to achieve different target displacements in Figure [23a.](#page-37-0) Required distributions of homogenized elasticity components $(C_{11}, C_{12}, C_{22}, C_{33})$ are precomputed for each target pattern (Figure [23b\)](#page-37-0). The 240k-size orthotropic mechanical metamaterial dataset, generated with Pixel/Voxel (Section [3.2.1\)](#page-4-0) and Perturbation (Section [3.2.2\)](#page-7-0), is used.

Figure [23c](#page-37-0) shows that the required data distributions vary significantly contingent upon tasks. Overall, the 240-k dataset can cover all three tasks, even if no information on them was given before the data acquisition. The on-demand property distribution for each task is quite disparate across the tasks. For example, the 50 \times 50 properties to achieve the smiley face (red) are relatively clustered. This implies that the wide coverage does not benefit this particular task as much as the other two cases. But

the smiley face task demands a large portion of anisotropic samples, i.e., those having large either C_{11}/C_{22} or C_{22}/C_{11} , as shown in the $C_{11} - C_{22}$ space. Thus, if known before the data acquisition, such samples can be prioritized with an associated metric during the data acquisition, as shown in Lee et al.^{[\[86\]](#page-39-0)} and Wang et al.^{[\[62\]](#page-39-0)} Meanwhile, the required 4×20 properties to produce the bridgelike deformation (blue) tend to widely spread in the associated property space. Among the tasks, the bridge-like deformation design benefits most from wide coverage, which is congruent with a generic goal of data acquisition. Even in this task, however, samples having negative Poisson's ratio are not used; this indicates such uniform coverage of property is not unconditionally favored, but depends on the intended tasks. We remark that herein we intentionally omit considering geometric/mechanical compatibility among building blocks to convey our point with a particular focus on data.

In summary, for multiscale design purposes, the data acquisition and assessment are intrinsically open to subjectivity, in part due to task specificity. Not all data holds equal utility for general design tasks. Given a target task(s) at the system level, the data acquisition and assessment should involve the specified task(s) and even intentionally introduce bias, in addition to data uniformity, to meet the task requirement and ensure an efficient data collection. Herein, the uniformity is meaningful only within the domains associated with the target tasks, as opposed to generic scenarios of data acquisition.

5.4.4. Other Challenges

Overall, the data-driven paradigm has shown its promise in multiscale metamaterial design, with a superior capability to excavate the underlying relations between properties and geometries. However, there is a tendency in existing studies to apply off-theshelf models as black-box tools for multiscale design, claiming that the universal fitting capability and high efficiency can benefit the design process. The underlying assumptions and constraints of the models have been constantly ignored, leaving the applicability questionable.

Before applying a specific model to multiscale design, it would be helpful to consider how it will influence the optimization solver, and whether its outputs and assumptions agree with the physics. For example, various training losses have been used to evaluate the performance of ML models, which mainly focus on the mean errors. It is not guaranteed that models trained with these loss functions will always produce physically feasible outputs in the whole input domain. In some cases, this could also lead to issues that are detrimental for optimization solvers, e.g., singular and negative-defined stiffness matrix, unit cells with disconnected components, the discontinuity between different cells, and non-differentiability and local fluctuations of the predicted responses. Some pre-trained deep NNs devised in computer graphics were transferred to the applications in multiscale design. However, hierarchical features extracted by the latent layers of these models might not be suitable for the structure designs. An important characteristic in multiscale meta-material designs is that a given macro-scale property can be achieved by multiple micro-structures.[\[35,66\]](#page-39-0) Nevertheless, in most existing methods, a one-to-one mapping is assumed between properties

 (a)

 (b)

 (c)

 C_{12}/C_{11}

Figure 23. Task specificity of data acquisition and assessment based on the case study in Wang et al.^{[\[66\]](#page-39-0)} a) Illustration of three system-level design tasks associated with different target displacements: smiley face (left), mouth (center), and bridge (right). b) On-demand distributions of homogenized elasticity components {*C*11, *C*12, *C*22, *C*33}. c) The distributions of required property for all the individual tasks plotted with respect to that of the orthotropic dataset (gray), created through Pixel/Voxel and Perturbation. All the results are from Wang et al.^{[\[66\]](#page-39-0)} a) and b) were reproduced with permission.[\[66\]](#page-39-0) Copyright 2020, Elsevier B.V.

 C_{11}

and geometries, which fails to accommodate the one-to-many nature. Most databases and ML models only consider constant property parameters of a single unit cell or periodically assembled unit cells. They do not take into account state-dependent and historydependent properties, e.g., strain-dependent stiffness tensor in nonlinear cases, and the interactions between different unit cells. Moreover, existing research mainly focuses on simple regression models to predict the homogenized properties and generative models to reduce the dimension of shape descriptors for the unitcell design. How to use data-driven methods to improve the optimization procedure itself, e.g., extraction of design rules and underlying physical knowledge, initial design recommendation, iterative optimization strategy, and combinatorial assembly, generally remains unexplored.

 C_{11}

6. Conclusion

We presented a comprehensive, critical review of the data-driven design of metamaterials and multiscale systems (generally referred to as DMD in this paper). Through our analysis, we categorized previous research endeavors into the distinct modules of a cohesive data-driven design framework, including data acquisition (Section [3\)](#page-4-0), unit-cell level learning and optimization (Section [4\)](#page-16-0), and multiscale system designs (Section [5\)](#page-26-0). In Section [3,](#page-4-0) we examined the common practices of data acquisition strategies, with special attention to the methodological components of shape generation and property-aware sampling, and to data assessment. In Section [4,](#page-16-0) we covered prior works that utilized datasets and ML to enable acceleration of, or higher

 C_{11}

CIENCE NEWS www.advancedsciencenews.com www.advmat.de

design freedom in, unit cell design. In Section [5,](#page-26-0) we reviewed data-driven multiscale designs that employed databases and ML at the unit-cell level to optimize structures across multiple scales, meeting heterogeneous properties requirements in a top-down or bottom-up manner. For these multiscale design efforts, the main focus was to replace homogenization with surrogate modeling, and increase design flexibility by accommodating diverse and oriented unit cells.

Based on our literature survey, we shared our perspectives on the current practices and suggested new avenues for future research efforts. In Section [3,](#page-4-0) we disclosed that the current research trend in DMD is arguably biased towards the final products at the downstream modules and lacks principled methods dedicated to data acquisition, which is critical to the successful and robust deployment of DMD frameworks. To address this, more benchmark datasets need to be made publicly available and standard dataset assessment protocols should be established so that the contributions of future works can be rigorously appreciated. In Section [4,](#page-16-0) we discussed the cost-benefit trade-off of the reviewed methods, which is usually neglected in prior works. In addition, we noted that the trustworthiness and creativity of ML models for unit cell designs are also under-studied. There are ample opportunities to develop interpretable or physics-informed ML methods, as well as to enhance their uncertainty quantification and generalization capabilities. In Section [5,](#page-26-0) we remarked that there are still many unexplored opportunities for data-driven multiscale system design, particularly in areas such as extracting design rules, discovering physical knowledge, providing initial design recommendations, facilitating iterative optimization strategies, and enabling combinatorial assembly.

Overall, ML has shown promise in metamaterial design with its superior capability to excavate the complex relations between properties and geometries. Despite the potential of the datadriven design approaches we reviewed, it is important to acknowledge that the field is still in its early stages and faces many grand challenges. Addressing these challenges means that the disconnect between the two primary approaches – data-driven and physics-based – needs to be resolved, which will require careful consideration of the cost-benefit and trustworthiness of ML methods, as well as collaboration between various disciplines. We believe that the tighter integration of physics and data-driven methods can unlock exciting possibilities for metamaterials design. Towards this future, we hope our review contributes to promoting interdisciplinary collaborations and bridging the gap between the two camps.

Supporting Information

Supporting Information is available from the Wiley Online Library or from the author.

Acknowledgements

D.L., W.W.C., and L.W. contributed equally to this work. This work was supported by the NSF BRITE Fellow program (Grant No. CMMI 2227641), the NSF CSSI program (Grant No. OAC 1835782), and the Northwestern Mc-Cormick Catalyst Award.

The authors declare no conflict of interest.

Keywords

data-driven design, machine learning, metamaterial, multiscale design, topology optimization

> Received: June 1, 2023 Revised: September 15, 2023 Published online: December 5, 2023

- [1] M. A. Skylar-Scott, J. Mueller, C. W. Visser, J. A. Lewis, *Nature* **2019**, *575*, 330.
- [2] M. Wehner, R. L. Truby, D. J. Fitzgerald, B. Mosadegh, G. M. Whitesides, J. A. Lewis, R. J. Wood, *Nature* **2016**, *536*, 451.
- [3] Y. Zhang, Z. Wang, Y. Yang, Q. Chen, X. Qian, Y. Wu, H. Liang, Y. Xu, Y. Wei, Y. Ji, *Sci. Adv.* **2020**, *6*, eaay8606.
- [4] Y. Kim, G. A. Parada, S. Liu, X. Zhao, *Sci. Rob.* **2019**, *4*, eaax7329.
- [5] X. Yu, J. Zhou, H. Liang, Z. Jiang, L. Wu, *Prog. Mater. Sci.* **2018**, *94*, 114.
- [6] N. I. Zheludev, *Science* **2010**, *328*, 582.
- [7] M. Kadic, G. W. Milton, M. van Hecke, M. Wegener, *Nat. Rev. Phys.* **2019**, *1*, 198.
- [8] T. S. Lumpe, T. Stankovic, *Proc. Natl. Acad. Sci. USA* **2021**, *118*, e2003504118.
- [9] U. Leonhardt, *Science* **2006**, *312*, 1777.
- [10] U. Leonhardt, T. Tyc, *Science* **2009**, *323*, 110.
- [11] N. Liu, H. Guo, L. Fu, S. Kaiser, H. Schweizer, H. Giessen, *Nat. Mater.* **2008**, *7*, 31.
- [12] Y. Ma, Y. Liu, L. Lan, T. Wu, W. Jiang, C. Ong, S. He, *Sci. Rep.* **2013**, *3*, 1.
- [13] J. B. Pendry, D. Schurig, D. R. Smith, *Science* **2006**, *312*, 1780.
- [14] C. Fan, Y. Gao, J. Huang, *Appl. Phys. Lett.* **2008**, *92*, 251907.
- [15] R. Schittny, M. Kadic, S. Guenneau, M. Wegener, *Phys. Rev. Lett.* **2013**, *110*, 195901.
- [16] T. Han, T. Yuan, B. Li, C.-W. Qiu, *Sci. Rep.* **2013**, *3*, 1.
- [17] R. Hu, S. Iwamoto, L. Feng, S. Ju, S. Hu, M. Ohnishi, N. Nagai, K. Hirakawa, J. Shiomi, *Phys. Rev. X* **2020**, *10*, 2.
- [18] Y. Chen, L. Wang, *Appl. Phys. Lett.* **2014**, *105*, 191907.
- [19] P. A. Deymier, *Acoustic metamaterials and phononic crystals*, vol. *173*, Springer Science & Business Media, Berlin **2013**.
- [20] A. S. Phani, M. I. Hussein, *Dynamics of lattice materials*, John Wiley & Sons, Hoboken, NJ **2017**.
- [21] J. Ma, K. Sun, S. Gonella, *Phys. Rev. Appl.* **2019**, *12*, 044015.
- [22] M. Wang, W. Zhou, L. Bi, C. Qiu, M. Ke, Z. Liu, *Nat. Commun.* **2020**, *11*, 1.
- [23] B. Florijn, C. Coulais, M. van Hecke, *Phys. Rev. Lett.* **2014**, *113*, 175503.
- [24] Y. Wang, L. Li, D. Hofmann, J. E. Andrade, C. Daraio, *Nature* **2021**, *596*, 238.
- [25] B. Jenett, C. Cameron, F. Tourlomousis, A. P. Rubio, M. Ochalek, N. Gershenfeld, *Sci. Adv.* **2020**, *6*, eabc9943.
- [26] D. M. Kochmann, K. Bertoldi, *Appl. Mech. Rev.* **2017**, *69*, 5.
- [27] T. Frenzel, M. Kadic, M. Wegener, *Science* **2017**, *358*, 1072.
- [28] L. Jin, R. Khajehtourian, J. Mueller, A. Rafsanjani, V. Tournat, K. Bertoldi, D. M. Kochmann, *Proc. Natl. Acad. Sci. USA* **2020**, *117*, 2319.
- [29] D. R. Reid, N. Pashine, J. M. Wozniak, H. M. Jaeger, A. J. Liu, S. R. Nagel, J. J. de Pablo, *Proc. Natl. Acad. Sci. USA* **2018**, *115*, E1384.

1DVANCI SCIENCE NEWS

- [30] M. Kadic, T. Bückmann, N. Stenger, M. Thiel, M. Wegener, *Appl. Phys. Lett.* **2012**, *100*, 191901.
- [31] M. C. Fernandes, J. Aizenberg, J. C. Weaver, K. Bertoldi, *Nat. Mater.* **2021**, *20*, 237.
- [32] M. A. Bessa, P. Glowacki, M. Houlder, *Adv. Mater.* **2019**, *31*, 1904845.
- [33] R. Hedayati, A. Güven, S. Van Der Zwaag, *Appl. Phys. Lett.* **2021**, *118*, 141904.
- [34] J. Panetta, Q. Zhou, L. Malomo, N. Pietroni, P. Cignoni, D. Zorin, *ACM Transactions on Graphics* **2015**, *34*, 1.
- [35] C. Schumacher, B. Bickel, J. Rys, S. Marschner, C. Daraio, M. Gross, *ACM Transactions on Graphics (TOG)* **2015**, *34*, 1.
- [36] D. Chen, M. Skouras, B. Zhu, W. Matusik, *Sci. Adv.* **2018**, *4*, 1.
- [37] G. X. Gu, C.-T. Chen, D. J. Richmond, M. J. Buehler, *Mater. Horiz.* **2018**, *5*, 939.
- [38] G. X. Gu, C.-T. Chen, M. J. Buehler, *Extreme Mechanics Letters* **2018**, *18*, 19.
- [39] D. Liu, Y. Tan, E. Khoram, Z. Yu, *Acs Photonics* **2018**, *5*, 1365.
- [40] S. Inampudi, H. Mosallaei, *Appl. Phys. Lett.* **2018**, *112*, 241102.
- [41] W. Ma, F. Cheng, Y. Liu, *ACS Nano* **2018**, *12*, 6326.
- [42] Z. Liu, D. Zhu, S. P. Rodrigues, K.-T. Lee, W. Cai, *Nano Lett.* **2018**, *18*, 6570.
- [43] X. Li, S. Ning, Z. Liu, Z. Yan, C. Luo, Z. Zhuang, *Computer Methods in Applied Mechanics and Engineering* **2020**, *361*, 112737.
- [44] B. Liu, L. Xu, J. Huang, *J. Appl. Phys.* **2021**, *130*, 045103.
- [45] C. Ma, Y. Chang, S. Wu, R. R. Zhao, *ACS Appl. Mater. Interfaces* **2022**, *14*, 33892.
- [46] J. Wu, O. Sigmund, J. P. Groen, *Structural and Multidisciplinary Optimization* **2021**, *63*, 1455.
- [47] L. Regenwetter, A. H. Nobari, F. Ahmed, *Journal of Mechanical Design* **2022**, *144*, 071704.
- [48] R. V. Woldseth, N. Aage, J. A. Bærentzen, O. Sigmund, *Structural and Multidisciplinary Optimization* **2022**, *65*, 1.
- [49] S. Mukherjee, D. Lu, B. Raghavan, P. Breitkopf, S. Dutta, M. Xiao, W. Zhang, *Archives of Computational Methods in Engineering* **2021**, *28*, 4549.
- [50] S. So, J. Mun, J. Park, J. Rho, *Adv. Mater.* **2022**, 2206399.
- [51] C. Wang, X. P. Tan, S. B. Tor, C. S. Lim, *Addit. Manuf.* **2020**, *36*, 101538.
- [52] M. Mozaffar, S. Liao, X. Xie, S. Saha, C. Park, J. Cao, W. K. Liu, Z. Gan, *J. Mater. Process. Technol.* **2022**, *302*, 2021.
- [53] S. So, T. Badloe, J. Noh, J. Rho, J. Bravo-Abad, *Nanophotonics* **2020**, *9*, 1041.
- [54] X. Liu, S. Tian, F. Tao, H. Du, W. Yu, *AIAA Scitech 2021 Forum* **2021**, 1.
- [55] S. Kumar, D. M. Kochmann, in *Current trends and open problems in computational mechanics*, Springer, New York **2022**, pp. 275–285.
- [56] Z. Liu, D. Zhu, L. Raju, W. Cai, *Adv. Sci.* **2021**, *8*, 2002923.
- [57] Y. Jin, L. He, Z. Wen, B. Mortazavi, H. Guo, D. Torrent, B. Djafari-Rouhani, T. Rabczuk, X. Zhuang, Y. Li, *Nanophotonics* **2022**, *11*, 439.
- [58] C.-X. Liu, G.-L. Yu, *Journal of Computational Design and Engineering* **2023**, *10*, 602.
- [59] X. Zheng, X. Zhang, T.-T. Chen, I. Watanabe, *Adv. Mater.* **2023**, 2302530.
- [60] S. Li, G. Wang, *Introduction to micromechanics and nanomechanics*, World Scientific Publishing Company, Singapore **2008**.
- [61] Y.-C. Chan, Yu-Chin Chan PhD Dissertation, Technical report, **2022**.
- [62] L. Wang, Y. C. Chan, Z. Liu, P. Zhu, W. Chen, *Structural and Multidisciplinary Optimization* **2020**, *61*, 2613.
- [63] J. Wang, W. W. Chen, D. Da, M. Fuge, R. Rai, *Computer Methods in Applied Mechanics and Engineering* **2022**, *396*, 115060.
- [64] Z. Liu, D. Zhu, S. P. Rodrigues, K. T. Lee, W. Cai, *Nano Lett.* **2018**, *18*, 6570.
- [65] Y. C. Chan, D. Da, L. Wang, W. Chen, *Structural and Multidisciplinary Optimization* **2022**, *65*, 1.
- [66] L. Wang, Y.-C. Chan, F. Ahmed, Z. Liu, P. Zhu, W. Chen, *Computer Methods in Applied Mechanics and Engineering* **2020**, *372*, 113377.
- [67] M. A. Kramer, *AIChE J.* **1991**, *37*, 233.
- [68] T. Qiu, X. Shi, J. Wang, Y. Li, S. Qu, Q. Cheng, T. Cui, S. Sui, *Adv. Sci.* **2019**, *6*, 1900128.
- [69] D. P. Kingma, M. Welling, *arXiv preprint arXiv:1312.6114* **2013**.
- [70] Z. Liu, L. Raju, D. Zhu, W. Cai, *IEEE Journal on Emerging and Selected Topics in Circuits and Systems* **2020**, *10*, 126.
- [71] I. Goodfellow, J. Pouget-Abadie, M. Mirza, B. Xu, D. Warde-Farley, S. Ozair, A. Courville, Y. Bengio, *Advances in Neural Information Processing Systems*, **2014**, *27*.
- [72] I. Goodfellow, J. Pouget-Abadie, M. Mirza, B. Xu, D. Warde-Farley, S. Ozair, A. Courville, Y. Bengio, *Commun. ACM* **2020**, *63*, 139.
- [73] A. Radford, L. Metz, S. Chintala, *arXiv preprint arXiv:1511.06434* **2015**.
- [74] C. Gurbuz, F. Kronowetter, C. Dietz, M. Eser, J. Schmid, S. Marburg, *The Journal of the Acoustical Society of America* **2021**, *149*, 1162.
- [75] W. Ma, F. Cheng, Y. Xu, Q. Wen, Y. Liu, *Adv. Mater.* **2019**, *31*, 1901111.
- [76] W. Ma, Y. Liu, *Science China Physics, Mechanics & Astronomy* **2020**, *63*, 1.
- [77] X. Zhu, Semi-supervised learning literature survey, Computer Sciences, University of Wisconsin-Madison, Madison, WI **2005**.
- [78] J. E. Van Engelen, H. H. Hoos, *Machine Learning* **2020**, *109*, 373.
- [79] R. Bellman, *Journal of Mathematics and Mechanics* **1957**, 679.
- [80] I. Sajedian, H. Lee, J. Rho, *Sci. Rep.* **2019**, *9*, 1.
- [81] F. Sui, R. Guo, Z. Zhang, G. X. Gu, L. Lin, *ACS Mater. Lett.* **2021**, *3*, 1433.
- [82] B. Zhu, M. Skouras, D. Chen, W. Matusik, *ACM Transactions on Graphics* **2017**.
- [83] I. Malkiel, M. Mrejen, A. Nagler, U. Arieli, L. Wolf, H. Suchowski, *Light: Science and Applications* **2018**, *7*, 1.
- [84] E. B. Whiting, S. D. Campbell, L. Kang, D. H. Werner, *Opt. Express* **2020**, *28*, 24229.
- [85] Y.-C. Chan, F. Ahmed, L. Wang, W. Chen, *Journal of Mechanical Design* **2021**, *143*, 3.
- [86] D. Lee, Y.-C. Chan, W. Chen, L. Wang, A. van Beek, W. Chen, *Journal of Mechanical Design* **2023**, *145*, 031704.
- [87] L. Wang, S. Tao, P. Zhu, W. Chen, *Journal of Mechanical Design* **2021**, *143*, 3.
- [88] K. Liu, R. Sun, C. Daraio, *Science* **2022**, *377*, 975.
- [89] D. Li, N. Dai, Y. Tang, G. Dong, Y. F. Zhao, *Journal of Mechanical Design, Transactions of the ASME* **2019**, *141*, 7.
- [90] S. Kumar, S. Tan, L. Zheng, D. M. Kochmann, *npj Comput. Mater.* **2020**, *6*, 1.
- [91] S. An, B. Zheng, H. Tang, M. Y. Shalaginov, L. Zhou, H. Li, M. Kang, K. A. Richardson, T. Gu, J. Hu, C. Fowler, H. Zhang, *Adv. Opt. Mater.* **2021**, *9*, 1.
- [92] S. An, C. Fowler, B. Zheng, M. Y. Shalaginov, H. Tang, H. Li, L. Zhou, J. Ding, A. M. Agarwal, C. Rivero-Baleine, K. A. Richardson, T. Gu, J. Hu, H. Zhang, *ACS Photonics* **2019**, *6*, 3196.
- [93] L. Wang, Z. Liu, D. Da, Y. C. Chan, W. Chen, P. Zhu, *Computer Methods in Applied Mechanics and Engineering* **2022**, *395*, 114967.
- [94] J. K. Wilt, C. Yang, G. X. Gu, *Adv. Eng. Mater.* **2020**, *22*, 1901266.
- [95] L. Wang, S. Yerramilli, A. Iyer, D. Apley, P. Zhu, W. Chen, *Journal of Mechanical Design* **2022**, *144*, 1.
- [96] L. Wang, Z. Liu, D. Da, Y. C. Chan, W. Chen, P. Zhu, *Computer Methods in Applied Mechanics and Engineering* **2022**, *395*, 114967.
- [97] H. Choi, A. M. Chung Baek, N. Kim, in *International Design Engineering Technical Conferences and Computers and Information in Engineering Conference*, vol. *59179*, American Society of Mechanical Engineers, New York **2019**, p. V001T02A037.
- [98] J. Boddapati, M. Flaschel, S. Kumar, L. De Lorenzis, C. Daraio, *arXiv preprint arXiv:2304.09112* **2023**.
- [99] Z. Liu, Z. Zhu, W. Cai, *Opt. Express* **2020**, *28*, 4825.

4DVANCED SCIENCE NEWS

- [100] W. Li, F. Wang, O. Sigmund, X. S. Zhang, *Proc. Natl. Acad. Sci. USA* **2022**, *119*, e2120563119.
- [101] Z. A. Kudyshev, A. V. Kildishev, V. M. Shalaev, A. Boltasseva, *Appl. Phys. Rev.* **2020**, *7*, 2.
- [102] Z. Chen, A. Ogren, C. Daraio, L. C. Brinson, C. Rudin, *Extreme Mechanics Letters* **2022**, 101895.
- [103] W. Ma, Y. Liu, *Science China: Physics, Mechanics and Astronomy* **2020**, *63*, 8.
- [104] S. So, J. Rho, *Nanophotonics* **2019**, *8*, 1255.
- [105] I. K. Kazmi, L. You, J. J. Zhang, in *2013 10th International Conference Computer Graphics, Imaging and Visualization*, IEEE, Piscataway, NJ **2013**, pp. 1–10.
- [106] I. Tanriover, D. Lee, W. Chen, K. Aydin, *ACS Photonics* **2022**.
- [107] H. Wang, S.-H. Xiao, C. Zhang, *Adv. Eng. Mater.* **2021**, *23*, 2100102.
- [108] A. A. Requicha, H. B. Voelcker, Constructive solid geometry, Technical report, **1977**.
- [109] X. Guo, W. Zhang, W. Zhong, *Journal of Applied Mechanics, Transactions ASME* **2014**, *81*, 1.
- [110] X. Lei, C. Liu, Z. Du, W. Zhang, X. Guo, *Journal of Applied Mechanics, Transactions ASME* **2019**, *86*, 1.
- [111] D. K. Lin, T. W. Simpson, W. Chen, *International Journal of Reliability and Applications* **2001**, *2*, 209.
- [112] R. Jin, W. Chen, A. Sudjianto, in *International design engineering technical conferences and computers and information in engineering conference*, vol. *36223*, **2002**, pp. 539–548.
- [113] S. So, J. Mun, J. Rho, *ACS Appl. Mater. Interfaces* **2019**, *11*, 24264.
- [114] C. M. Bishop, N. M. Nasrabadi, *Pattern recognition and machine learning*, vol. *4*, Springer, New York **2006**.
- [115] N. Sambasivan, S. Kapania, H. Highfill, D. Akrong, P. Paritosh, L. M. Aroyo, in *Proceedings of the 2021 CHI Conference on Human Factors in Computing Systems*, **2021**, pp. 1–15.
- [116] K. Fu, D. Moreno, M. Yang, K. L. Wood, *Journal of Mechanical Design* **2014**, *136*, 111102.
- [117] S. Jiang, J. Hu, K. L. Wood, J. Luo, *Journal of Mechanical Design* **2022**, *144*, 2.
- [118] M. Miniaci, A. Krushynska, A. B. Movchan, F. Bosia, N. M. Pugno, *Appl. Phys. Lett.* **2016**, *109*, 071905.
- [119] J. Liu, L. Li, B. Xia, X. Man, *Int. J. Solids Struct.* **2018**, *132*, 20.
- [120] R. Hamzehei, A. Zolfagharian, S. Dariushi, M. Bodaghi, *Smart Mater. Struct.* **2022**, *31*, 035001.
- [121] B. Li, C. Zhang, F. Peng, W. Wang, B. D. Vogt, K. Tan, *J. Mater. Chem. C* **2021**, *9*, 1164.
- [122] J. K. Nagel, R. L. Nagel, R. B. Stone, D. A. McAdams, *Ai Edam* **2010**, *24*, 521.
- [123] A. K. Goel, D. A. McAdams, R. B. Stone, *Biologically inspired design*, Springer, New York **2015**.
- [124] U. Faroog Ghumman, A. Iyer, R. Dulal, J. Munshi, A. Wang, T. Chien, G. Balasubramanian, W. Chen, *Journal of Mechanical Design* **2018**, *140*, 11.
- [125] A. Iyer, R. Dulal, Y. Zhang, U. F. Ghumman, T. Chien, G. Balasubramanian, W. Chen, *Comput. Mater. Sci.* **2020**, *179*, 109559.
- [126] W.-L. Loh, *The Annals of Statistics* **1996**, *24*, 2058.
- [127] B. Settles, in *Active learning and experimental design workshop in conjunction with AISTATS 2010*, JMLR Workshop and Conference Proceedings, **2011**, pp. 1–18.
- [128] Y. Gal, R. Islam, Z. Ghahramani, in *International conference on machine learning*, PMLR, **2017**, pp. 1183–1192.
- [129] P. Ren, Y. Xiao, X. Chang, P.-Y. Huang, Z. Li, B. B. Gupta, X. Chen, X. Wang, *ACM Computing Surveys (CSUR)* **2021**, *54*, 1.
- [130] R. M. Monarch, *Human-in-the-Loop Machine Learning: Active learning and annotation for human-centered AI*, Simon and Schuster, New York **2021**.
- [131] A. Farahani, S. Voghoei, K. Rasheed, H. R. Arabnia, *Advances in Data Science and Information Engineering: Proceedings from ICDATA 2020 and IKE 2020* **2021**, pp. 877–894.
- [132] H. Zhang, W. W. Chen, J. M. Rondinelli, W. Chen, *Appl. Phys. Rev.* **2023**, *10*, 021403.
- [133] K. Choudhary, K. F. Garrity, A. C. Reid, B. DeCost, A. J. Biacchi, A. R. Hight Walker, Z. Trautt, J. Hattrick-Simpers, A. G. Kusne, A. Centrone, A. Davydov, J. Jiang, R. Pachter, G. Cheon, E. Reed, A. Agrawal, X. Qian, V. Sharma, H. Zhuang, S. V. Kalinin, B. G. Sumpter, G. Pilania, P. Acar, S. Mandal, K. Haule, D. Vanderbilt, K. Rabe, F. Tavazza, *npj Comput. Mater.* **2020**, *6*, 173.
- [134] P. Branco, L. Torgo, R. P. Ribeiro, *ACM Computing Surveys* **2016**, *49*, 1.
- [135] A. Kulesza and, B. Taskar, *Foundations and Trends® in Machine Learning* **2012**, *5*, 123.
- [136] C. E. Rasmussen, in *Summer school on machine learning*, Springer, New York **2004**, pp. 63–71.
- [137] A. Rahimi, B. Recht, *Advances in Neural Information Processing Systems* **2007**, *20*.
- [138] J. Gillenwater, A. Kulesza, B. Taskar, *Advances in Neural Information Processing Systems* **2012**, *25*.
- [139] R. H. Affandi, E. Fox, B. Taskar, *Advances in Neural Information Processing Systems* **2013**, *26*.
- [140] D. Lee, S. Jiang, O. Balogun, W. Chen, *ACS Photonics* **2021**, *9*, 351.
- [141] H. Zhang, W. W. Chen, A. Iyer, D. W. Apley, W. Chen, *Sci. Rep.* **2022**, *12*, 19760.
- [142] R. Jin, W. Chen, A. Sudijanto, in *International design engineering technical conferences and computers and information in engineering conference*, vol. *37009*, **2003**, pp. 545–554.
- [143] X. Du, W. Chen, *J. Mech. Des.* **2004**, *126*, 225.
- [144] L. Pronzato, W. G. Müller, *Statistics and Computing* **2012**, *22*, 681.
- [145] B. Shang, D. W. Apley, *Journal of Quality Technology* **2021**, *53*, 173.
- [146] J. E. Saal, S. Kirklin, M. Aykol, B. Meredig, C. Wolverton, *Jom* **2013**, *65*, 1501.
- [147] T. M. Cover, J. A. Thomas, *Internet Access* **2006**, 19.
- [148] M. Ashby, *Scr. Mater.* **2013**, *68*, 4.
- [149] L. Wang, A. van Beek, D. Da, Y.-C. Chan, P. Zhu, W. Chen, *Compos. Struct.* **2022**, *280*, 114949.
- [150] L. Xia, P. Breitkopf, *Structural and Multidisciplinary Optimization* **2015**, *52*, 1229.
- [151] M. Zandehshahvar, Y. Kiarashinejad, M. Zhu, H. Maleki, T. Brown, A. Adibi, *ACS Photonics* **2022**, *9*, 714.
- [152] H. Abdi, L. J. Williams, *Wiley Interdisciplinary Reviews: Computational Statistics* **2010**, *2*, 433.
- [153] L. Van der Maaten, G. Hinton, *Journal of Machine Learning Research* **2008**, *9*, 11.
- [154] B. Schölkopf, J. C. Platt, J. Shawe-Taylor, A. J. Smola, R. C. Williamson, *Neural Computation* **2001**, *13*, 1443.
- [155] L. McInnes, J. Healy, J. Melville, *arXiv preprint arXiv:1802.03426* **2018**.
- [156] C. Picard, J. Schiffmann, F. Ahmed, *arXiv preprint arXiv:2305.09018* **2023**.
- [157] L. Regenwetter, B. Curry, F. Ahmed, *Journal of Mechanical Design* **2022**, *144*, 031706.
- [158] S. Ning, F. Yang, C. Luo, Z. Liu, Z. Zhuang, *Extreme Mechanics Letters* **2020**, *35*, 100623.
- [159] M. V. Zhelyeznyakov, S. Brunton, A. Majumdar, *ACS Photonics* **2021**, *8*, 481.
- [160] S. An, B. Zheng, M. Y. Shalaginov, H. Tang, H. Li, L. Zhou, Y. Dong, M. Haerinia, A. M. Agarwal, C. Rivero-Baleine, M. Kang, K. A. Richardson, T. Gu, J. Hu, C. Fowler, H. Zhang, *Adv. Opt. Mater.* **2022**, *10*, 2102113.
- [161] E. Ghavanloo, S. A. Fazelzadeh, *Acta Mech.* **2019**, *230*, 4453.

4DVANCED SCIENCE NEWS

- [162] H. Zhu, S. Patnaik, T. F. Walsh, B. H. Jared, F. Semperlotti, *Proc. Natl. Acad. Sci. USA* **2020**, *117*, 26099.
- [163] C. T. Ertsgaard, R. M. McKoskey, I. S. Rich, N. C. Lindquist, *ACS Nano* **2014**, *8*, 10941.
- [164] R. D. Buijs, T. A. Wolterink, G. Gerini, E. Verhagen, A. F. Koenderink, *Adv. Opt. Mater.* **2021**, *9*, 2100435.
- [165] C. Spägele, M. Tamagnone, D. Kazakov, M. Ossiander, M. Piccardo, F. Capasso, *Nat. Commun.* **2021**, *12*, 1.
- [166] S. L. Brunton, J. L. Proctor, J. N. Kutz, *Proc. Natl. Acad. Sci. USA* **2016**, *113*, 3932.
- [167] M. Raissi, P. Perdikaris, G. E. Karniadakis, *J. Comput. Phys.* **2019**, *378*, 686.
- [168] S. Goswami, M. Yin, Y. Yu, G. E. Karniadakis, *Computer Methods in Applied Mechanics and Engineering* **2022**, *391*, 114587.
- [169] S. A. Faroughi, N. Pawar, C. Fernandes, S. Das, N. K. Kalantari, S. K. Mahjour, *arXiv preprint arXiv:2211.07377* **2022**.
- [170] S. Cuomo, V. S. Di Cola, F. Giampaolo, G. Rozza, M. Raissi, F. Piccialli, *Journal of Scientific Computing* **2022**, *92*, 88.
- [171] Z. Fang, J. Zhan, *Ieee Access* **2019**, *8*, 24506.
- [172] Y. Chen, L. Lu, G. E. Karniadakis, L. Dal Negro, *Opt. Express* **2020**, *28*, 11618.
- [173] Y. Chen, L. Dal Negro, *APL Photonics* **2022**, *7*, 1.
- [174] J. Lim, D. Psaltis, *Apl. Photonics* **2022**, *7*, 1.
- [175] L. Lu, R. Pestourie, S. G. Johnson, G. Romano, *Phys. Rev. Res.* **2022**, *4*, 023210.
- [176] Y. Augenstein, T. Repan, C. Rockstuhl, *ACS Photonics* **2023**.
- [177] L. Zhang, H. You, Y. Yu, *arXiv preprint arXiv:2206.02040* **2022**.
- [178] M. Zhu, H. Zhang, A. Jiao, G. E. Karniadakis, L. Lu, *Computer Methods in Applied Mechanics and Engineering* **2023**, *412*, 116064.
- [179] J. Ho, A. Jain, P. Abbeel, *Advances in Neural Information Processing Systems* **2020**, *33*, 6840.
- [180] P. Dhariwal, A. Nichol, *Advances in Neural Information Processing Systems* **2021**, *34*, 8780.
- [181] N. N. Vlassis, W. Sun, *Computer Methods in Applied Mechanics and Engineering* **2023**, *413*, 116126.
- [182] J.-H. Bastek, D. M. Kochmann, *arXiv preprint arXiv:2305.19836* **2023**.
- [183] M. Rixner, P.-S. Koutsourelakis, *npj Comput. Mater.* **2022**, *8*, 46.
- [184] M. Motamedi, N. Sakharnykh, T. Kaldewey, *arXiv preprint arXiv:2110.03613* **2021**.
- [185] S. E. Whang, Y. Roh, H. Song, J.-G. Lee, *The VLDB Journal* **2023**, 1.
- [186] M. Mazumder, C. Banbury, X. Yao, B. Karlaš, W. G. Rojas, S. Diamos, G. Diamos, L. He, A. Parrish, H. R. Kirk, J. Quaye, C. Rastogi, D. Kiela, D. Jurado, D. Kanter, R. Mosquera, J. Ciro, L. Aroyo, B. Acun, L. Chen, M. S. Raje, M. Bartolo, S. Eyuboglu, A. Ghorbani, E. Goodman, O. Inel, T. Kane, C. R. Kirkpatrick, T.-S. Kuo, J. Mueller, *arXiv preprint arXiv:2207.10062* **2022**.
- [187] E. Strickland, *IEEE Spectrum* **2022**, *59*, 22.
- [188] H. Zhao, X. Li, Y. Zhang, L. S. Schadler, W. Chen, L. C. Brinson, *APL Mater.* **2016**, *4*, 053204.
- [189] J. P. McCusker, N. Keshan, S. Rashid, M. Deagen, C. Brinson, D. L. McGuinness, in *The Semantic Web–ISWC 2020: 19th International Semantic Web Conference, Athens, Greece, November 2–6, 2020, Proceedings, Part II*, Springer, New York **2020**, pp. 144–159.
- [190] L. C. Brinson, M. Deagen, W. Chen, J. McCusker, D. L. McGuinness, L. S. Schadler, M. Palmeri, U. Ghumman, A. Lin, B. Hu, *ACS Macro Lett.* **2020**, *9*, 1086.
- [191] M. D. Wilkinson, M. Dumontier, I. J. Aalbersberg, G. Appleton, M. Axton, A. Baak, N. Blomberg, J.-W. Boiten, L. B. da Silva Santos, P. E. Bourne, J. Bouwman, A. J. Brookes, T. Clark, M. Crosas, I. Dillo, O. Dumon, S. Edmunds, C. T. Evelo, R. Finkers, A. Gonzalez-Beltran, A. J. G. Gray, P. Groth, C. Goble, J. S. Grethe, J. Heringa, P. A. C. Hoen, R. Hooft, T. Kuhn, R. Kok, J. Kok, *Scientific Data* **2016**, *3*, 1.
- [192] M. Boeckhout, G. A. Zielhuis, A. L. Bredenoord, *European Journal of Human Genetics* **2018**, *26*, 931.
- [193] J. Smajic, C. Hafner, L. Raguin, K. Tavzarashvili, M. Mishrikey, *J. Comput. Theor. Nanosci.* **2009**, *6*, 763.
- [194] D. J. Poole, C. B. Allen, T. C. Rendall, *AIAA J.* **2015**, *53*, 1349.
- [195] C. B. Allen, D. J. Poole, T. Rendall, *Optimization and Engineering* **2018**, *19*, 453.
- [196] J. Li, J. Cai, K. Qu, *Structural and Multidisciplinary Optimization* **2019**, *59*, 403.
- [197] W. Chen, K. Chiu, M. D. Fuge, *AIAA J.* **2020**, *58*, 4723.
- [198] J. Li, M. Zhang, J. R. Martins, C. Shu, *AIAA J.* **2020**, *58*, 4243.
- [199] V. Sekar, M. Zhang, C. Shu, B. C. Khoo, *AIAA J.* **2019**, *57*, 993.
- [200] Q. Chen, J. Wang, P. Pope, W. W. Chen, M. Fuge, *J. Mech. Des.* **2022**, *144*.
- [201] A. Glaws, R. N. King, G. Vijayakumar, S. Ananthan, *AIAA J.* **2022**, *60*, 3035.
- [202] A. K. Azad, W. J. Kort-Kamp, M. Sykora, N. R. Weisse-Bernstein, T. S. Luk, A. J. Taylor, D. A. Dalvit, H.-T. Chen, *Sci. Rep.* **2016**, *6*, 20347.
- [203] W. Jiang, Y. Zhu, G. Yin, H. Lu, L. Xie, M. Yin, *Mater. Today Phys.* **2022**, *22*, 100616.
- [204] I. Sajedian, J. Kim, J. Rho, *Microsyst. Nanoeng.* **2019**, *5*, 1.
- [205] P. R. Wiecha, O. L. Muskens, *Nano Lett.* **2019**, *20*, 329.
- [206] S. An, B. Zheng, M. Y. Shalaginov, H. Tang, H. Li, L. Zhou, J. Ding, A. M. Agarwal, C. Rivero-Baleine, M. Kang, K. A. Richardson, T. Gu, J. Hu, C. Fowler, H. Zhang, *Opt. Express* **2020**, *28*, 31932.
- [207] K. Donda, Y. Zhu, A. Merkel, S.-W. Fan, L. Cao, S. Wan, B. Assouar, *Smart Mater. Struct.* **2021**, *30*, 085003.
- [208] A. P. Garland, B. C. White, S. C. Jensen, B. L. Boyce, *Mater. Des.* **2021**, *203*, 109632.
- [209] J. Zhang, G. Wang, T. Wang, F. Li, *ACS Appl. Mater. Interfaces* **2021**, *13*, 7792.
- [210] Q. Ji, Y. Qi, C. Liu, S. Meng, J. Liang, M. Kadic, G. Fang, *Int. J. Heat Mass Transfer* **2022**, *189*, 122716.
- [211] H. Pahlavani, M. Amani, M. C. Saldívar, J. Zhou, M. J. Mirzaali, A. A. Zadpoor, *Commun. Mater.* **2022**, *3*, 1.
- [212] S. Lee, Z. Zhang, G. X. Gu, *Mater. Horiz.* **2022**, *9*, 952.
- [213] Z. Wang, W. Xian, M. R. Baccouche, H. Lanzerath, Y. Li, H. Xu, *Journal of Mechanical Design* **2022**, *144*, 041705.
- [214] J. Hensman, N. Fusi, N. D. Lawrence, *arXiv preprint arXiv:1309.6835* **2013**.
- [215] T. Evans, P. Nair, in *International Conference on Machine Learning*, PMLR, **2018**, pp. 1417–1426.
- [216] R. Bostanabad, Y.-C. Chan, L. Wang, P. Zhu, W. Chen, *Journal of Mechanical Design* **2019**, *141*, 11.
- [217] K. Wang, G. Pleiss, J. Gardner, S. Tyree, K. Q. Weinberger, A. G. Wilson, *Advances in Neural Information Processing Systems* **2019**, *32*.
- [218] H. Liu, Y.-S. Ong, X. Shen, J. Cai, *IEEE Transactions on Neural Networks and Learning Systems* **2020**, *31*, 4405.
- [219] M. Elzouka, C. Yang, A. Albert, R. S. Prasher, S. D. Lubner, *Cell Reports Physical Science* **2020**, *1*, 100259.
- [220] J. Lin, C. Zhong, D. Hu, C. Rudin, M. Seltzer, in *International Conference on Machine Learning*. PMLR, **2020**, pp. 6150 –6160.
- [221] Z. Yang, M. J. Buehler, *Small Methods* **2022**, *6*, 2200537.
- [222] W. W. Chen, D. Lee, O. Balogun, W. Chen, *Journal of Mechanical Design* **2022**, *145*, 011703.
- [223] S. C.-y. Shen, M. J. Buehler, *Commun. Eng.* **2022**, *1*, 1.
- [224] T. Karras, S. Laine, T. Aila, in *Proceedings of the IEEE/CVF conference on computer vision and pattern recognition*, **2019**, pp. 4401–4410.
- [225] B. Liu, L. Xu, J. Huang, *J. Appl. Phys.* **2021**, *129*, 065101. [226] H. Van Hasselt, A. Guez, D. Silver, in *Proceedings of the AAAI conference on artificial intelligence*, vol. *30*, **2016** .
- [227] A. Raina, J. Cagan, C. Mccomb, *Journal of Mechanical Design* **2022**, 1.
- [228] J. Jiang, J. A. Fan, *Nano Lett.* **2019**, *19*, 5366.
- [229] J. Jiang, J. A. Fan, *Nanophotonics* **2020**, *9*, 1059.

4DVANCED SCIENCE NEWS

- [230] L. Lu, R. Pestourie, W. Yao, Z. Wang, F. Verdugo, S. G. Johnson, *SIAM Journal on Scientific Computing* **2021**, *43*, B1105.
- [231] G. E. Karniadakis, I. G. Kevrekidis, L. Lu, P. Perdikaris, S. Wang, L. Yang, *Nat. Rev. Phys.* **2021**, *3*, 422.
- [232] H. T. Kollmann, D. W. Abueidda, S. Koric, E. Guleryuz, N. A. Sobh, *Mater. Des.* **2020**, *196*, 109098.
- [233] T. Zhao, Y. Li, L. Zuo, K. Zhang, *Extreme Mechanics Letters* **2021**, *45*, 101297.
- [234] C. Liu, W. M. Yu, Q. Ma, L. Li, T. J. Cui, *Photonics Res.* **2021**, *9*, B159.
- [235] S. So, Y. Yang, T. Lee, J. Rho, *Photonics Res.* **2021**, *9*, B153.
- [236] C. Yeung, J. M. Tsai, B. King, B. Pham, D. Ho, J. Liang, M. W. Knight, A. P. Raman, *Nanophotonics* **2021**, *10*, 1133.
- [237] Z. Zhen, C. Qian, Y. Jia, Z. Fan, R. Hao, T. Cai, B. Zheng, H. Chen, E. Li, *Photonics Res.* **2021**, *9*, B229.
- [238] M. Mirza, S. Osindero, *arXiv preprint arXiv:1411.1784* **2014**.
- [239] K. Sohn, H. Lee, X. Yan, *Advances in Neural Information Processing Systems* **2015**, *28*.
- [240] J. Jiang, D. Sell, S. Hoyer, J. Hickey, J. Yang, J. A. Fan, *ACS Nano* **2019**, *13*, 8872.
- [241] H. P. Wang, Y. B. Li, H. Li, S. Y. Dong, C. Liu, S. Jin, T. J. Cui, *Advanced Intelligent Systems* **2020**, *2*, 2000068.
- [242] F. Wen, J. Jiang, J. A. Fan, *ACS Photonics* **2020**, *7*, 2098.
- [243] Y.-T. Luo, P.-Q. Li, D.-T. Li, Y.-G. Peng, Z.-G. Geng, S.-H. Xie, Y. Li, A. Alù, J. Zhu, X.-F. Zhu, *Research* **2020**, *2020*.
- [244] Y. Mao, Q. He, X. Zhao, *Sci. Adv.* **2020**, *6*, eaaz4169.
- [245] This work does not appear on Figure [15](#page-24-0) since the design is not limited to fixed dimensionality.
- [246] K. Barri, P. Jiao, Q. Zhang, J. Chen, Z. L. Wang, A. H. Alavi, *Nano Energy* **2021**, *86*, 106074.
- [247] E. W. Wang, D. Sell, T. Phan, J. A. Fan, *Opt. Mater. Express* **2019**, *9*, 469.
- [248] B. J. Roxworthy, A. M. Bhuiya, X. Yu, E. K. Chow, K. C. Toussaint Jr, *Nat. Commun.* **2014**, *5*, 4427.
- [249] C. Chen, O. Li, D. Tao, A. Barnett, C. Rudin, J. K. Su, *Advances in Neural Information Processing Systems* **2019**, *32*.
- [250] Z. Yang, A. Zhang, A. Sudjianto, *Pattern Recognition* **2021**, *120*, 108192.
- [251] G. W. Milton, M. Briane, J. R. Willis, *New J. Phys.* **2006**, *8*, 248.
- [252] W. Zuo, K. Saitou, *Structural and Multidisciplinary Optimization* **2017**, *55*, 477.
- [253] D. Gu, X. Shi, R. Poprawe, D. L. Bourell, R. Setchi, J. Zhu, *Science* **2021**, *372*, eabg1487.
- [254] E. Garner, H. M. Kolken, C. C. Wang, A. A. Zadpoor, J. Wu, *Addit. Manuf.* **2019**, *26*, 65.
- [255] A. Sydney Gladman, E. A. Matsumoto, R. G. Nuzzo, L. Mahadevan, J. A. Lewis, *Nat. Mater.* **2016**, *15*, 413.
- [256] Z. Li, R. Pestourie, Z. Lin, S. G. Johnson, F. Capasso, *ACS Photonics* **2022**, *9*, 2178.
- [257] L. Wang, J. Boddapati, K. Liu, P. Zhu, C. Daraio, W. Chen, *Proc. Natl. Acad. Sci. USA* **2022**, *119*, e2122185119.
- [258] T. Bückmann, M. Thiel, M. Kadic, R. Schittny, M. Wegener, *Nat. Commun.* **2014**, *5*, 1.
- [259] J. W. Boley, W. M. Van Rees, C. Lissandrello, M. N. Horenstein, R. L. Truby, A. Kotikian, J. A. Lewis, L. Mahadevan, *Proc. Natl. Acad. Sci. USA* **2019**, *116*, 20856.
- [260] T. Wu, A. Tovar, *Rapid Prototyping J.* **2018**.
- [261] J. Deng, J. Yan, G. Cheng, *Structural and Multidisciplinary Optimization* **2013**, *47*, 583.
- [262] J. Yan, X. Guo, G. Cheng, *Computational Mechanics* **2016**, *57*, 437.
- [263] M. Kadic, M. Wegener, A. Nicolet, F. Zolla, S. Guenneau, A. Diatta, *Wave Motion* **2020**, *92*, 102419.
- [264] Y. Zhang, M. Xiao, L. Gao, J. Gao, H. Li, *Mechanical Systems and Signal Processing* **2020**, *135*, 106369.
- [265] J. Zhao, H. Yoon, B. D. Youn, *Computer Methods in Applied Mechanics and Engineering* **2019**, *347*, 700.
- [266] C. R. Thomsen, F. Wang, O. Sigmund, *Computer Methods in Applied Mechanics and Engineering* **2018**, *339*, 115.
- [267] F. Wang, O. Sigmund, *Journal of the Mechanics and Physics of Solids* **2021**, *152*, 104415.
- [268] M.-S. Pham, C. Liu, I. Todd, J. Lertthanasarn, *Nature* **2019**, *565*, 305.
- [269] S. Yin, W. Guo, H. Wang, Y. Huang, R. Yang, Z. Hu, D. Chen, J. Xu, R. O. Ritchie, *Journal of the Mechanics and Physics of Solids* **2021**, *149*, 104341.
- [270] H. Jiang, A. Coomes, Z. Zhang, H. Ziegler, Y. Chen, *Composites, Part B* **2021**, *224*, 109183.
- [271] J. Wu, O. Sigmund, J. P. Groen, Structural and Multidisciplinary Op*timization* **2021**, *63*, 1455.
- [272] L. Xia, P. Breitkopf, *Archives of Computational Methods in Engineering* **2017**, *24*, 227.
- [273] R. Sivapuram, P. D. Dunning, H. A. Kim, *Structural and Multidisciplinary Optimization* **2016**, *54*, 1267.
- [274] M. P. Bendsøe, N. Kikuchi, *Computer Methods in Applied Mechanics and Engineering* **1988**, *71*, 197.
- [275] O. Sigmund, K. Maute, *Structural and Multidisciplinary Optimization* **2013**, *48*, 1031.
- [276] H. A. Eschenauer, N. Olhoff, *Appl. Mech. Rev.* **2001**, *54*, 331.
- [277] J. D. Deaton, R. V. Grandhi, *Structural and Multidisciplinary Optimization* **2014**, *49*, 1.
- [278] M. Zhou, G. I. Rozvany, *Computer Methods in Applied Mechanics and Engineering* **1991**, *89*, 309.
- [279] H. Mlejnek, *Structural Optimization* **1992**, *5*, 64.
- [280] M. P. Bendsøe, O. Sigmund, *Archive of Applied Mechanics* **1999**, *69*, 635.
- [281] L. Cheng, J. Bai, A. C. To, *Computer Methods in Applied Mechanics and Engineering* **2019**, *344*, 334.
- [282] X. Liu, L. Gao, M. Xiao, Y. Zhang, *Computer Methods in Applied Mechanics and Engineering* **2022**, *390*, 114466.
- [283] D. Li, N. Dai, Y. Tang, G. Dong, Y. F. Zhao, *Journal of Mechanical Design* **2019**, *141*, 7.
- [284] Y. Wang, H. Xu, D. Pasini, *Computer Methods in Applied Mechanics and Engineering* **2017**, *316*, 568.
- [285] Z. Wu, L. Xia, S. Wang, T. Shi, *Computer Methods in Applied Mechanics and Engineering* **2019**, *345*, 602.
- [286] D. A. White, W. J. Arrighi, J. Kudo, S. E. Watts, *Computer Methods in Applied Mechanics and Engineering* **2019**, *346*, 1118.
- [287] J. P. Groen, *Multi-scale design methods for topology optimization*, DTU Mechanical Engineering, **2018**.
- [288] M. Avellaneda, *SIAM J. Appl. Math.* **1987**, *47*, 1216.
- [289] Z. Wu, F. Fan, R. Xiao, L. Yu, *International Journal for Numerical Methods in Engineering* **2020**, *121*, 2964.
- [290] L. Jiang, X. D. Gu, S. Chen, *Journal of Mechanical Design* **2021**, *143*, 1.
- [291] J. P. Groen, O. Sigmund, *International Journal for Numerical Methods in Engineering* **2018**, *113*, 1148.
- [292] P. D. L. Jensen, O. Sigmund, J. P. Groen, *Computer Methods in Applied Mechanics and Engineering* **2022**, *399*, 115426.
- [293] A. Y. Tamijani, S. P. Velasco, L. Alacoque, *Mater. Des.* **2020**, *196*, 109155.
- [294] P. Vogiatzis, M. Ma, S. Chen, X. D. Gu, *Computer Methods in Applied Mechanics and Engineering* **2018**, *328*, 477.
- [295] C. Ma, D. Xue, S. Li, Z. Zhou, Y. Zhu, X. Guo, *Computer Methods in Applied Mechanics and Engineering* **2022**, *395*, 114861.
- [296] O. Pantz, K. Trabelsi, *SIAM Journal on Control and Optimization* **2008**, *47*, 1380.
- [297] J. P. Groen, F. C. Stutz, N. Aage, J. A. Bærentzen, O. Sigmund, *Computer Methods in Applied Mechanics and Engineering* **2020**, *364*, 112979.

CIENCE NEWS

www.advancedsciencenews.com www.advmat.de

- [298] M. O. Elingaard, N. Aage, J. A. Bærentzen, O. Sigmund, *Computer Methods in Applied Mechanics and Engineering* **2022**, *388*,
- 114197. [299] P. Geoffroy-Donders, G. Allaire, G. Michailidis, O. Pantz, *Interna-*
- *tional Journal for Numerical Methods in Engineering* **2020**. [300] P. Geoffroy-Donders, G. Allaire, O. Pantz, *J. Comput. Phys.* **2020**, *401*, 108994.
- [301] T. Kumar, K. Suresh, *Structural and Multidisciplinary Optimization* **2019**, *61*, 1399.
- [302] Z. Liu, L. Xia, Q. Xia, T. Shi, *Structural and Multidisciplinary Optimization* **2020**, *61*, 2227.
- [303] C. Wang, J. H. Zhu, W. H. Zhang, S. Y. Li, J. Kong, *Structural and Multidisciplinary Optimization* **2018**, *58*, 35.
- [304] Y.-C. Chan, D. Da, L. Wang, W. Chen, *Structural and Multidisciplinary Optimization* **2021**, [http://arxiv.org/abs/2112.00648.](http://arxiv.org/abs/2112.00648)
- [305] L. Zheng, S. Kumar, D. M. Kochmann, *Computer Methods in Applied Mechanics and Engineering* **2021**, *383*, 113894.
- [306] F. V. Senhora, E. D. Sanders, G. H. Paulino, *Adv. Mater.* **2022**, 2109304.
- [307] C. Wang, X. Gu, J. Zhu, H. Zhou, S. Li, W. Zhang, *Structural and Multidisciplinary Optimization* **2020**, *61*, 869.
- [308] Y. Zhang, H. Li, M. Xiao, L. Gao, S. Chu, J. Zhang, *Structural and Multidisciplinary Optimization* **2019**, *59*, 1273.
- [309] Y. Zhang, L. Gao, M. Xiao, *Computers \& Structures* **2020**, *230*, 106197.
- [310] C. Soyarslan, S. Bargmann, M. Pradas, J. Weissmüller, *Acta Mater.* **2018**, *149*, 326.
- [311] T. Bückmann, M. Kadic, R. Schittny, M. Wegener, *Proc. Natl. Acad. Sci. USA* **2015**, *112*, 4930.
- [312] B. Bickel, M. Bächer, M. Otaduy, H. Lee, H. Pfister, M. Gross, W. Matusik, *ACM Transactions on Graphics* **2010**, *29*, 1.
- [313] C. Chu, G. Graf, D. W. Rosen, *Computer-Aided Design and Applications* **2008**, *5*, 686.
- [314] A. D. Cramer, V. J. Challis, A. P. Roberts, *Structural and Multidisciplinary Optimization* **2015**, *53*, 489.
- [315] Y. Han, W. F. Lu, *Journal of Mechanical Design (1990)* **2018**, *140*, 9.
- [316] V. Mironov, R. P. Visconti, V. Kasyanov, G. Forgacs, C. J. Drake, R. R. Markwald, *Biomaterials* **2009**, *30*, 2164.
- [317] J.-H. Bastek, S. Kumar, B. Telgen, R. N. Glaesener, D. M. Kochmann, *Proc. Natl. Acad. Sci. USA* **2022**, *119*, e2111505119.
- [318] D. Da, Y.-C. Chan, L. Wang, W. Chen, *Extreme Mechanics Letters* **2022**, *50*, 101528.
- [319] S. Wolfram, *Nature* **1984**, *311*, 419.
- [320] V. P. Nguyen, M. Stroeven, L. J. Sluys, *Journal of Multiscale Modelling* **2011**, *3*, 229.
- [321] M. Wallin, D. A. Tortorelli, *Mech. Mater.* **2020**, *145*, 103324.
- [322] P. B. Nakshatrala, D. Tortorelli, K. Nakshatrala, *Computer Methods in Applied Mechanics and Engineering* **2013**, *261*, 167.
- [323] B. Hassani, E. Hinton, *Computers & Structures* **1998**, *69*, 707.
- [324] D. Da, L. Xia, *International Journal for Numerical Methods in Engineering* **2021**, *122*, 1333.
- [325] J. Fu, L. Xia, L. Gao, M. Xiao, H. Li, *Journal of Mechanical Design* **2019**, *141*, 7.
- [326] H. Yao, Y. Gao, Y. Liu, *Computer Methods in Applied Mechanics and Engineering* **2020**, *363*, 112892.
- [327] S. Saha, Z. Gan, L. Cheng, J. Gao, O. L. Kafka, X. Xie, H. Li, M. Tajdari, H. A. Kim, W. K. Liu, *Computer Methods in Applied Mechanics and Engineering* **2021**, *373*, 113452.
- [328] Y. Efendiev, T. Y. Hou, *Multiscale finite element methods: theory and applications*, vol. *4*, Springer Science & Business Media, Berlin **2009**.
- [329] H. Liu, Y. Hu, B. Zhu, W. Matusik, E. Sifakis, *ACM Transactions on Graphics (TOG)* **2018**, *37*, 1.

Doksoo Lee is a Ph.D. candidate at the Integrated DEsign Automation Laboratory (IDEAL) at Northwestern University. His current research involves developing next-generation devices through datadriven modeling and design of multiscale systems. Key topics of interest include data-driven design, scientific machine learning, multifunctional devices, and topology optimization. His core career quest is to create and deliver a suite of tools to support engineering-aware individual creation in both the academic and public spheres. He received his B.S. and M.S. in mechanical engineering at Yonsei University. He was a summer intern of Department of Materials Durability at Ford Motors Company.

Wei (Wayne) Chen is an assistant professor in mechanical engineering at Texas A&M University. His research focuses on how artificial intelligence and machine learning can assist humans in solving challenging design problems including high-dimensional problems, high-throughput inverse design, design under realistic uncertainty, and novel design synthesis. After he received his Ph.D. from the University of Maryland at College Park in 2019, he worked as a Research Scientist at Siemens Technology (Princeton, NJ) and a Postdoctoral Scholar at Northwestern University.

Liwei Wang is a postdoctoral researcher at Northwestern University in the Integrated DEsign Automation Laboratory. His research interests centered around the development of machine learning methods and computational design frameworks in the context of multiscale and programmable metamaterial systems to achieve physical intelligence. Dr. Wang holds a B.S. and a Ph.D. degree in mechanical engineering at Shanghai Jiao Tong University. Prior to his current role, he also studied as a visiting scholar at Northwestern University, where his research interests focused on the data-driven design of heterogeneous metamaterial systems.

Yu-Chin Chan is a research scientist at Siemens Corporation, Technology. Her interest is in merging deep learning, optimization, simulation, and graphics methods to advance industrial design and process automation. She holds a Ph.D. in mechanical engineering from Northwestern University, where she was a NSF Graduate Research Fellow, and a B.S., also in mechanical engineering, from North Carolina State University.

Wei Chen is the Wilson-Cook Professor in Engineering Design and Chair of Department of Mechanical Engineering at Northwestern University. Directing the Integrated DEsign Automation Laboratory (IDEAL-), her current research involves the use of statistical inference, machine learning, and uncertainty quantification techniques for design of emerging materials systems including microstructural materials, metamaterials and programmable materials. Dr. Chen is an elected member of the National Academy of Engineering (NAE). She served as the Editor-in-chief of the ASME Journal of Mechanical Design, the Chair of the ASME Design Engineering Division (DED), and the President of the International Society of Structural and Multidisciplinary Optimization (ISSMO). She received her Ph.D. from the Georgia Institute of Technology in 1995.